

One-step preparation of the BiVO₄ film photoelectrode

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Abstract A one-step method of preparing photoelectrochemically active nanostructured BiVO₄ films is reported based on thermolysis (500 °C in air) of a polyethylene glycol (PEG300) “paint-on” precursor solution containing Bi³⁺ (as nitrate) and VO₄³⁻ (as the metavanadate ammonium salt). Films are formed directly on tin-doped indium oxide (ITO) substrates and characterised by electron microscopy (scanning electron microscopy (SEM), energy-dispersive X-ray spectroscopy (EDS)), X-ray diffraction, Raman spectroscopy, and photoelectrochemistry. The nanocrystalline film exhibited typically up to 52 % incident photon to current efficiency (IPCE) at 1.0 V vs. saturated calomel electrode (SCE) in aqueous 0.5 M Na₂SO₄ with oxalate, strongly enhancing photocurrents.

Keywords Water splitting · Photoanode · Oxygen evolution · Solar energy · Vanadium oxide · Nanostructure

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Introduction

Bismuth vanadate (BiVO₄) is an n-type semiconductor, which is usually used as a yellow pigment, but which has recently also attracted attention for use in photoelectrochemical water splitting [1–4] and in the photolytic destruction of organic pollutants [5, 6]. The advantages offered by BiVO₄ are (i) a suitable band gap (usually 2.4 to 2.5 eV for monoclinic BiVO₄; however, a wider range of values has been reported), (ii) inexpensive components, and (iii) excellent stability in aqueous media, making this material a promising photoanode for visible light harvesting. More recent studies have focused on the preparation of BiVO₄, seeking to improve photocatalytic activity by using various strategies such as morphology control, construction of nanocomposite structures, and doping. BiVO₄ can be fabricated by solution-based methods including aqueous, hydrothermal, and solvothermal processes [7–9] or by spray pyrolysis [10]. In most of the synthesis methods proposed in the literature, several hours of reflux and use of aggressive chemicals such as concentrated nitric acid are required. In addition, often BiVO₄ is obtained as powder material, and when it is employed in photoelectrodes, an extra preparation step (conversion of powder to thin film) is necessary. For example, an elegant solvothermal method using an ethylene glycol–water–sodium oleate system was proposed by Wang et al., where the precursor solution was treated in a Teflon-lined autoclave at 180 °C for 24 h and BiVO₄ powder was produced [11]. Sayama et al. used a modified method of metal-organic decomposition and a polyethylene glycol solvent followed by evaporation under vacuum to obtain BiVO₄ films on tin-doped indium oxide (ITO) [12]. In this paper, we propose a relatively simple and direct one-step route to photoactive BiVO₄ nanostructured films based on combining Bi(NO₃)₃ and NH₄VO₃ directly in polyethylene glycol (PEG300) as solvent and applying this

precursor solution to ITO-coated glass, followed by thermal treatment at 500 °C in air.

Experimental

Reagents used were $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$, NH_4VO_3 , polyethylene glycol (PEG300), and Na_2SO_4 from Sigma-Aldrich (UK). Aqueous solutions were prepared with deionised and filtered water taken from a Thermo Scientific water purification system (Barnstead Nanopure, UK). Electrochemical measurements were performed with a $\mu\text{Autolab}$ Type II potentiostat system (Ecochemie, Utrecht, Netherlands) with Autolab GPES software.

A solar simulator Xe lamp with 100 mW cm^{-2} (AM1.5) output was employed in cyclic voltammetry photoelectrochemical experiments. In some experiments, a high-power blue LED ($\lambda=405 \text{ nm}$, M405L2 UV LED, Thorlabs, Ely, UK) together with LED driver (T-Cube LED driver, Thorlabs, Ely, UK) and a DDS Function/Arbitrary TG4001 Generator (TTi, Huntingdon, UK) were employed. The electrode/electrolyte contact area was 1 cm^2 , front-lit, and the electrolyte was $0.5 \text{ M Na}_2\text{SO}_4$. Field emission scanning electron microscopy (FE-SEM) images were obtained using a JEOL JSM6301F microscope. Photocurrent spectra (incident photon to current efficiency (IPCE) data) were plotted as a function of incident light with λ ranging from 350 to 550 nm in steps of 10 nm. Light intensity of this optical window was calibrated using a standard silicon photodiode. The light chopping frequency was set at 0.5 Hz to minimise the attenuation affects. The apparatus was based on a Stanford Research Systems SR830 (Stanford, USA) lock-in amplifier, SR540 chopper controller, a home-made potentiostat, a 75-W xenon lamp powered by a Bentham 650 (Reading, UK) power supply, and a monochromator controller PMC3B of Bentham Instruments Ltd. (Reading, UK). Raman spectroscopy studies were carried out with an inVia Renishaw Raman microscope system (Gloucestershire, UK). The crystal phase of the BiVO_4 thin film was confirmed by X-ray diffraction (XRD, Bruker AXS D8 Advance diffractometer with a θ - 2θ configuration and using $\text{CuK}\alpha$ radiation $\lambda=1.5418 \text{ \AA}$). Diffuse reflectance spectra were obtained using a UV-vis-NIR spectrometer (PerkinElmer Lambda 35).

The electrodes used were a platinum wire counter electrode and a potassium chloride saturated calomel (SCE) reference electrode. The working electrodes were made from glass coated with ITO (Image Optics Components, Basildon, Essex, UK), cut to give $1 \text{ cm} \times 3 \text{ cm}$ electrodes (sonicated at 30 min in acetone, washed with distilled water, and heated for 1 h at 500 °C before use).

For BiVO_4 film production, initially two solutions were prepared. The first solution was prepared by dissolving 1.036 g of $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ in 5 mL of PEG300 and the second

by dissolving 0.25 g of NH_4VO_3 in 5 mL of PEG300. The solutions were then mixed to form a dark orange solution, which was agitated in an ultrasonic bath for 30 min. This mixture was then used as the precursor for deposition of BiVO_4 films. The deposition was carried out by employing a drawing roller (hard rubber lino printing roller) [13]. The precursor solution was distributed evenly onto the ITO substrate (ca. $20 \mu\text{L cm}^{-2}$ to give an average thickness of 0.2 to $0.5 \mu\text{m}$), and it was then pre-dried with warm air for 15 min. The process was repeated to increase the thickness of the film. After this deposition, the coated ITO was heated at 500 °C (this temperature was chosen to give high-quality BiVO_4 films without deformation of the underlying glass slide) for 1 h in a tube furnace in the presence of air. The ITO glass film substrate was fully covered with a yellow adherent film after the preparation process.

Results and discussion

The Raman spectrum and XRD data for BiVO_4 thin films are shown in panels a and b of Fig. 1, respectively. From the Raman data, it is possible to observe characteristic peaks at around 210, 324, 368, 720, and 828 cm^{-1} , corresponding to the Raman active vibrational modes consistent with a pure monoclinic phase film of BiVO_4 [14]. The appearance of V-O band at 828 cm^{-1} indicates good crystallinity [15, 16]. The broad peak at $1,100 \text{ cm}^{-1}$ corresponds to the ITO substrate. Figure 1b shows the XRD pattern for the BiVO_4 film sample. The indexed peaks are associated mainly with the planes of the monoclinic BiVO_4 structure (PDF No. 01-083-1699). The characteristic strong peak of tetragonal zircon phase at 24° was not observed, and therefore, there is no tetragonal zircon-type BiVO_4 phase present in the samples [17]. It is also possible to observe further diffraction peaks from the underlying ITO substrate, identified with an asterisk symbol in Fig. 1b. A scanning electron micrograph (SEM) of a BiVO_4 film is shown in Fig. 1c, where agglomeration of individual crystallites is observed. The SEM image shows that the film is dense, crystalline, and with particles typically around 1 to $2 \mu\text{m}$ in diameter. The energy-dispersive X-ray spectroscopy (EDS) data analysis in Fig. 1d revealed the presence of the elements Bi, V, and O in the film with approximately 1:1 atomic ratio for Bi:V.

Typical cyclic voltammograms for the BiVO_4 film on ITO in aqueous $0.5 \text{ M Na}_2\text{SO}_4$ solution in light and in the dark are shown in Fig. 2a. Two characteristic redox systems centred at approximately -0.1 V (weak) and at -0.4 V vs. SCE (strong) are indicative of reduction and re-oxidation of two different types of V^{5+} to V^{4+} surface states (tentatively assigned; or possibly also accumulation of charges in the conduction band, which has been reported to have mixed V, Bi, and O character [18, 7]). An anodic photocurrent is observed commencing

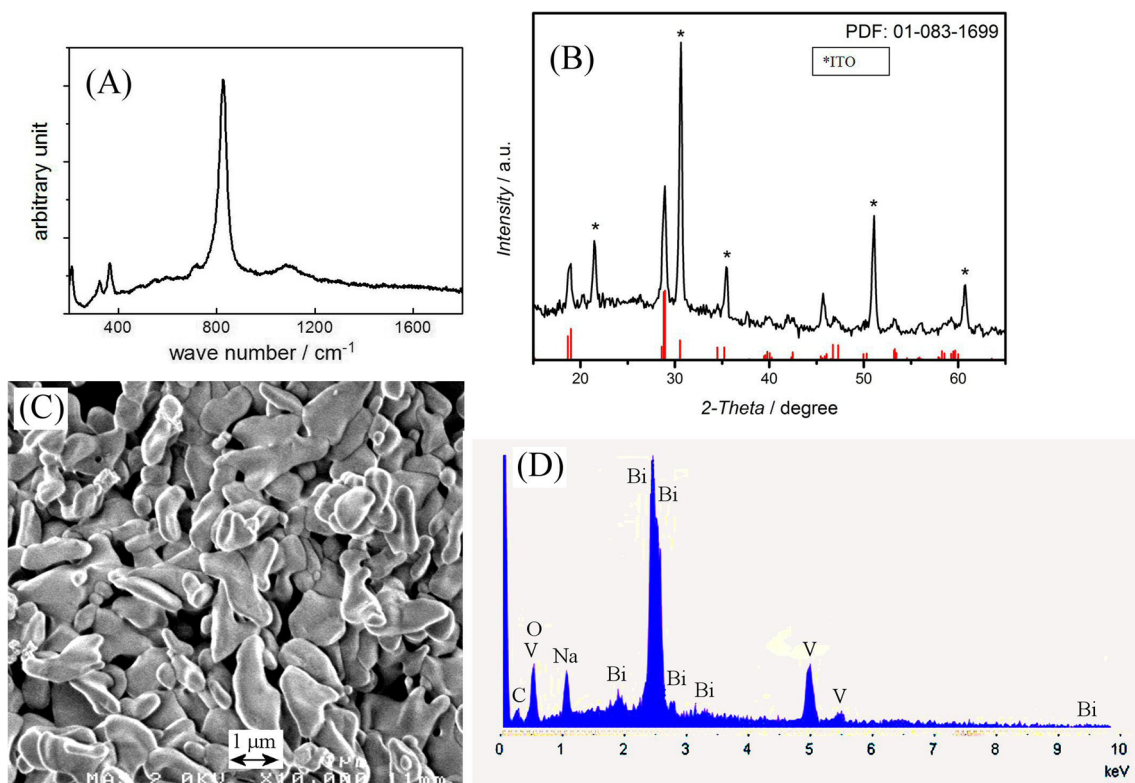


Fig. 1 a Raman spectrum, b X-ray diffraction pattern, and c FE-SEM image with d EDS analysis for the BiVO₄ thin film electrode formed on ITO

from ca. 0.0 V vs. SCE, which increases with increased bias potential when under illumination. The anodic photocurrent is associated with photooxidation of water. Hysteresis in the photocurrent profile is likely to be linked to local pH changes within the porous film structure. At 1.0 V vs. SCE, a typical value for the photocurrent is 300 $\mu\text{A cm}^{-2}$ under solar simulator conditions (AM1.5). Substantially lower photocurrents were reported for example by Chatchai et al. using FTO/SnO₂/BiVO₄ and FTO/BiVO₄ film photoanodes [19]. The photocurrent observed in this work is larger than most results reported in the literature for pure BiVO₄ films [19–21] and consistent with high-quality films with good electrical contact to the underlying substrate.

The diffuse reflectance spectra of the samples were measured (see Fig. 2b). The band gap can be estimated as 2.52 eV (see red line), which is close to the values reported previously [10, 11, 13, 17]. In contrast, tetragonal zircon-type BiVO₄ has been reported to exhibit a wider less beneficial band gap (2.9 eV) [18]. Therefore, the results are consistent with Raman and XRD data, which indicate the presence of only the monoclinic BiVO₄ phase. In Fig. 2c, the IPCE values for BiVO₄/ITO electrodes are presented as a function of the irradiation wavelength measured at +1.0 V vs. SCE anodic bias in 0.5 M Na₂SO₄ solution. The electrodes show photocurrent responses in the visible light region up to about 490 nm (see inset), in accordance with the literature. Again, a band gap of approximately 2.52 eV (see red line),

characteristic of BiVO₄, is observed. The ITO/BiVO₄ electrode shows a high efficiency for solar light conversion in both regions of UV and visible light. The maximum value of IPCE was 52 %, which is considerable when compared to typical results reported in the literature [19–23] but lower than the 73 % IPCE data (at 420 nm) reported by Jie et al. [24].

Both this study and that by Jie et al. [24] are based on a thermolysis process involving a precursor solution, which may have beneficial effects on the resulting substrate to BiVO₄ film interface. This interface is crucial in allowing effective flow of electrons into the underlying ITO substrate as well as in minimising recombination by oxygen reacting with V (IV) or directly at the ITO surface. The nature of this interface is likely to be affected by the gradual evaporation of PEG and residual components forming during thermolysis.

The presence of recombination processes can be demonstrated by addition of an oxalate hole quencher into the aqueous phase and study of photocurrents under pulsed blue light conditions. Figure 3 shows data for (a) a BiVO₄ electrode immersed in 0.5 M Na₂SO₄ and (b) the same experiment in the presence of 8 mM oxalate and (c) in the presence of 50 mM oxalate (all at pH 5). In the absence of oxalate, a typical light pulse transient response is observed with crossover from anodic to cathodic photocurrent pulse depending on the scan direction. This behaviour suggests non-steady-state behaviour possibly due to pH gradients developing in the solution close to the electrode surface. With 8 mM oxalate present (Fig. 3b),

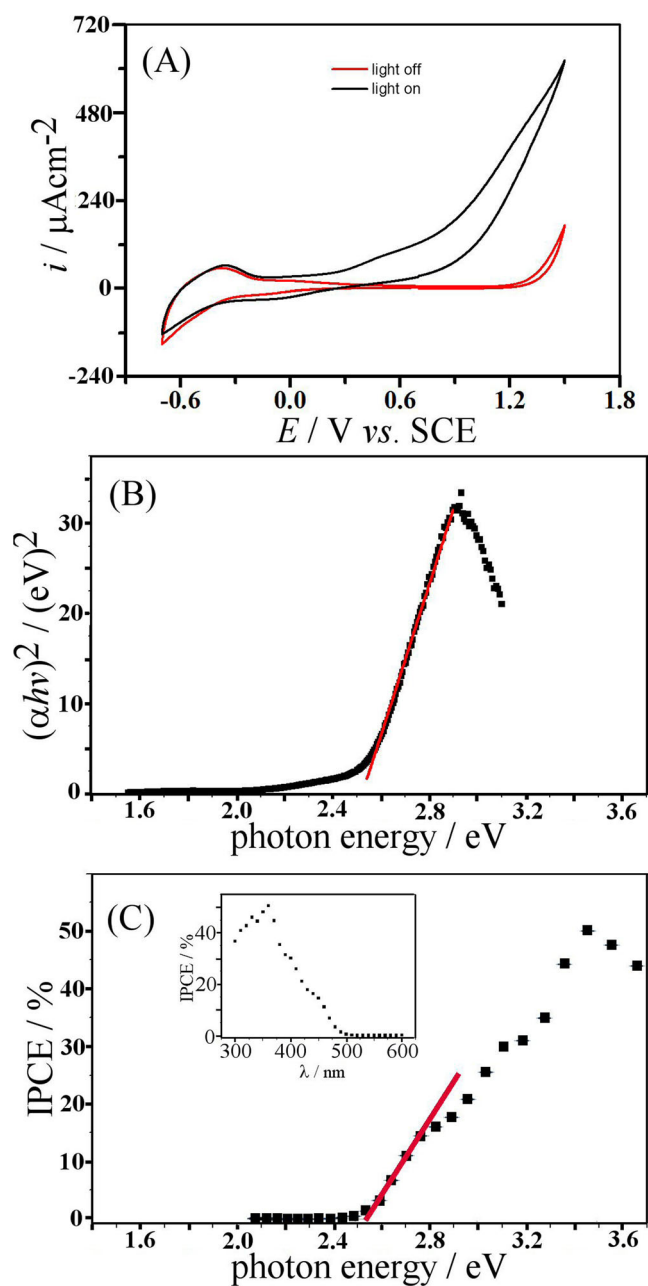


Fig. 2 **a** Cyclic voltammety (scan rate 20 mVs^{-1}) for a BiVO_4 film electrode in $0.5 \text{ M Na}_2\text{SO}_4$ in the dark and light (AM1.5 solar simulator, 100 mW cm^{-2} Xe lamp). **b** Diffuse reflectance data plot $(\alpha h\nu)^2$ vs. photon energy for a BiVO_4 thin film on ITO. **c** IPCE plot for a BiVO_4 thin film at 1.0 V vs. SCE in $0.5 \text{ M Na}_2\text{SO}_4$. *Inset*: IPCE plot vs. wavelength

features are similar but now significantly higher photocurrents are detected. Oxalate is able to react with two holes from the photoexcited BiVO_4 to give CO_2 [25] (this is possible even on the ITO substrate [26], but the process is believed to be dominated here by BiVO_4). Therefore, recombination via solution processes can be suppressed. Further increase in oxalate concentration to 50 mM (see Fig. 3c) causes only minor further enhancement. There are two distinct steps in

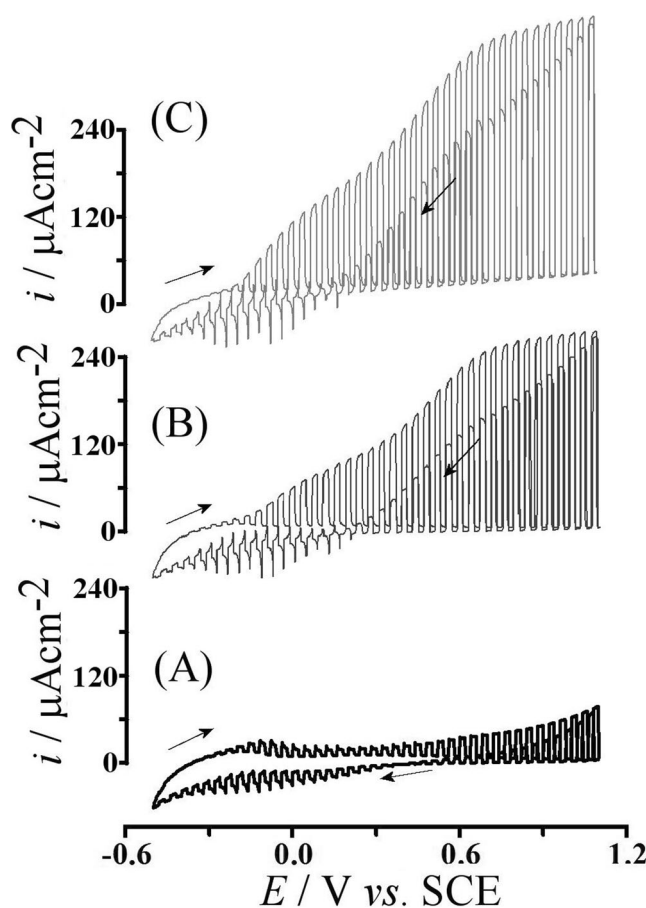


Fig. 3 **a–c** Cyclic voltammety (scan rate 20 mVs^{-1}) for a BiVO_4 film electrode in $0.5 \text{ M Na}_2\text{SO}_4$ **a** with 8 mM oxalate **b** and with 50 mM oxalate **c** in the presence of pulsed light (blue LED, 0.5 Hz)

the photocurrent possibly due to the increase in applied bias potential causing a change in reaction pathway. Even higher concentrations of oxalate were observed to gradually corrode BiVO_4 (presumably by extraction of Bi^{3+} cations) associated with an increase in the dark current.

Conclusions

A crystalline, homogeneous, and adherent BiVO_4 thin film with monoclinic structure was produced on ITO substrate electrodes using a one-step method based on a PEG300 precursor solution. The method is experimentally simple, fast, and inexpensive and leads to an effective ITO– BiVO_4 interface. The photoanode material obtained exhibits photocurrents over a large potential range and with good IPCE values. It can be concluded that the “paint-on” synthesis of BiVO_4 films presented in this work is promising for use as photoelectrodes in water splitting, in photocatalytic degradation of organics, or in dye-sensitised solar cells [27]. Work on a wider range of substrates and applications is in progress.

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