



Trapping of Magnetic Flux in Bi-2223 Ceramic Superconductors Doped with α -Al₂O₃ Nanoparticles

M. Hernández-Wolpez¹ · P. R. Gallart-Tauler² · I. García-Fornaris³ · E. Govea-Alcaide⁴ · E. Pérez-Tijerina⁵ · R. F. Jardim⁶ · P. Muné²

Received: 8 June 2019 / Accepted: 14 August 2019 / Published online: 5 September 2019
© Springer Science+Business Media, LLC, part of Springer Nature 2019

Abstract

By combining experimental results and a simple model, we offer here an explanation for the role played by the low-angle grain boundaries and the doping with α -Al₂O₃ nanoparticles in the trapping of the magnetic flux in Bi_{1.65}Pb_{0.35}Sr₂Ca₂Cu₃O_{10+ δ} (Bi-2223) ceramic samples. Our model correlates the size of the nanoparticles, properties of the superconducting matrix, and the magnetic flux trapped at the intragranular planar defects, the so-called Abrikosov-Josephson vortices. The results indicate that the role played by the doping with α -Al₂O₃ nanoparticles is to pin the AJ vortices at the grain boundaries with misorientation angles $\theta \sim 10^\circ$. We also argue that a similar procedure for identifying conditions for the trapping of the magnetic flux may be applied to other superconducting materials doped with non-magnetic nanoparticles.

Keywords Bi-based superconductors · Trapped flux · Doping with nanoparticles · Low-angle grain boundaries

1 Introduction

It is well established that three types of vortices can be found within polycrystalline samples of high T_c superconductors: Abrikosov (A), Abrikosov-Josephson (AJ), and Josephson (J) [1, 2]. It is also well established that, in the limit of low misorientation angle ($\theta < 5^\circ$) between neighboring crystallites, vortices behave as A vortices: they have

core sizes comparable to the coherence length, ξ , and are essentially pinned by defects [2, 3]. As θ increases, the superconducting critical current density, J_c , across grain boundaries (GBs) decreases exponentially [3, 4] and, as a consequence, vortices become anisotropic. Such an anisotropy is related to an increase in the size l of the core of the vortices along GBs, which is not observed along the perpendicular direction. These vortices, different from the A vortices, are called AJ vortices [1, 2]. For much higher values of θ , the condition $l > \lambda$, where λ is the London penetration depth, is fulfilled and a transition from AJ vortices to J vortices is expected to occur [2]. Gurevich and co-workers have calculated the flux flow resistivity of the AJ vortices, identified in transport data by taken $\theta \sim 7^\circ$ in unirradiated and irradiated YBCO bicrystals [2]. From these results, the core size and the intrinsic depairing current density on nanoscale of few GB dislocations were measured for the first time.

More recently, clusters of superconducting oxides have been associated with intragranular defects as colonies of low-angle c-axis boundaries [5]. Thus, the magnetic trapped flux within the clusters was found to be due to AJ vortices [6]. In order to identify such kind of vortices, *flux-trapping curves* [7] as a function of the average minimum boundary angle, θ_{min} , were constructed. It was possible because the maximum applied magnetic field, H_{am} , was written

✉ P. Muné
mune@uo.edu.cu

- ¹ Departamento de Física Universidad de Camagüey, Ctra. Circunvalación Norte, Km 5 1/2, Camagüey, Cuba
- ² Departamento de Física, Universidad de Oriente, Patricio Lumumba s/n, P. O. Box 90500, Santiago de Cuba, Cuba
- ³ Departamento de Matemática-Física, Facultad de Ciencias Informáticas, Naturales y Exactas, Universidad de Granma, Bayamo, Cuba
- ⁴ Departamento de Física, Universidade Federal do Amazonas, Av. General Rodrigo Octávio, 6200 Coroado I Manaus, 69077-000 Brazil
- ⁵ Centro de Investigación en Ciencias Físico-Matemáticas, FCFM, UANL, Monterrey, Nuevo León, México
- ⁶ Instituto de Física, Universidade de São Paulo, Rua do Matão, 1371, São Paulo, SP, 05508-090 Brazil

as a function of θ_{min} . Within this scenario, one is able to determine graphically, from the flux-trapping curves, values of the GB angles for which the superconducting critical current density exhibits appreciable decrease due to the trapped flux within the specimen. Moreover, different types of vortices present in the material, according to the value of H_{am} , could be determined. The model also offers the possibility of studying separately the dynamics of J, AJ, and A vortices located at grain boundaries of a polycrystalline sample in anisotropic superconducting materials. The discussion was focused on the region of the flux-trapping curves in which the maximum applied magnetic field, H_{am} , is less than the lower critical field of the crystallites, H_{c1c} , because the main objective there was the study of the intragranular medium of the materials.

In this paper, we clear up the role of the low-angle grain boundaries and the doping with α -Al₂O₃ nanoparticles in the intragranular flux trapping of Bi-2223 ceramic samples. We first discuss certain microstructural characteristics of the samples used in this study that are important support to confirm the theoretical results. This has been done by a generalization of the relationship between H_{am} and θ_{min} published elsewhere [6]. The next step was to compare: the required energy to create an AJ vortex in a low-angle boundary and the one required for the creation of a vortex in a pore, an extra phase region, or even a nanoparticle located in the vicinity of a low-angle boundary. From the results of this comparison, we have found an inequality, which correlates the following parameters: the angle of the grain boundary, its transparency, the size of the nanoparticles, and the coherence length of the superconducting material. Finally, based on these theoretical tools, the general features of the experimental flux-trapping curves are analyzed. Our results clearly indicate that the role played by the doping with α -Al₂O₃ nanoparticles is to pin the AJ vortices in grain boundaries with misorientation angles $\theta \sim 10^\circ$. As far as the authors know, the combination of these experimental and theoretical results is reported for the first time.

2 Experimental Details

Details of the preparation of the samples are described elsewhere [8–10]. In this work, three samples were studied: two of them have been used for other studies and their general physical properties are described elsewhere [10]. A third sample of Bi_{1.65}Pb_{0.35}Sr₂Ca₂Cu₃O_{10+ δ} (Bi-2223), without α -Al₂O₃ nanoparticles (0.0% (**B00**)), was synthesized, under the same conditions, in order to compare its behavior with those of the doped samples. We have also employed for the **B00** sample the same two experimental techniques applied in other investigations [10]: X-ray diffraction analysis (XRD) and the dependence

of $J_c(0, H_{am})$. This type of measurement corresponds to the so-called flux-trapping curves which were measured at 77 K [7].

To estimate the mean size of the α -Al₂O₃ NPs located within the Bi-2223 matrix, two techniques were employed: high-resolution transmission electron microscopy (HRTEM) and atomic force microscopy (AFM). By using a field-emission transmission electron microscope, JEM-2200FS (200 kV), several micrographies were obtained with 0.19-nm resolution in the TEM mode. For this purpose, a dry powder sample was dispersed in isopropanol and deposited onto a Formvar carbon copper grid. The AFM technique was applied by means of an CPII (SPM) model of a scanning probe microscope in scanning mode. Several images were obtained and size profiles of α -Al₂O₃ nanoparticles were extracted.

3 Results and Discussion

3.1 Microstructural Characterizations

The studied samples are composed by a mixture of phases in which the Bi-2223 phase coexists, along with α -Al₂O₃, with three other phases: Bi-2212, Bi-2201, and Ca₂PbO₄ (see Fig. 1) [10]. The determination of the phase composition was performed similar to Li and Han [11] and the results, displayed in Table 1, are slightly different from those reported for a similar material studied elsewhere [12]. Differences between both sample preparation methods [9] provoked these differences in their phase compositions.

The unit-cell parameters of the Bi-2223 phase were calculated with respect to an orthorhombic unit cell and the obtained values are displayed in Table 2.

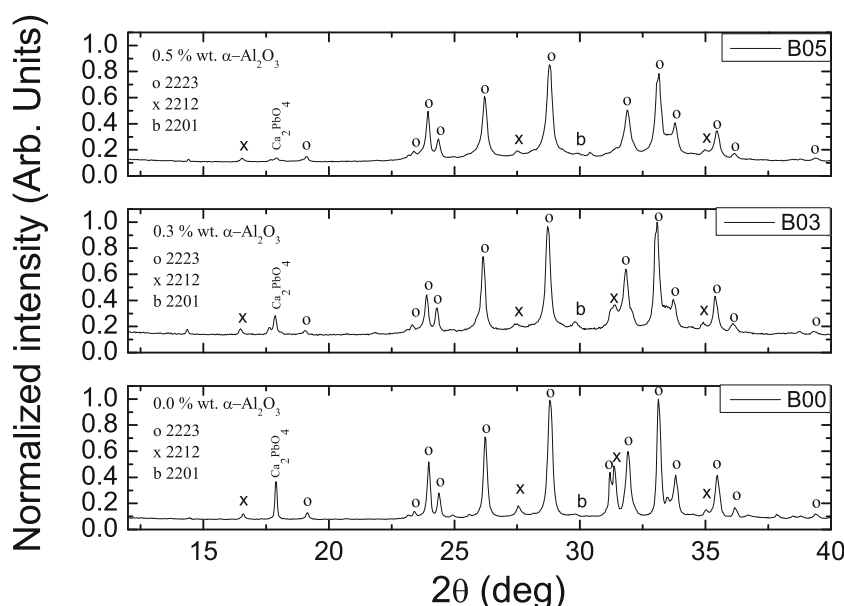
By using several images similar to those shown in Figs. 2 and 3, the mean size of the α -Al₂O₃ nanoparticles has been estimated. In order to obtain this important result, the images from HRTEM and AFM were analyzed by the softwares Gatan Digital Micrograph and WSxM 5.0 Develop 8.1 [13], respectively.

From the analysis of the HRTEM images, the mean size of the nanoparticles measured inside the superconducting matrix was 42 nm. In the case of the AFM images, the explored area was approximately 10.4 μm^2 and the obtained mean size was 35.6 nm. Both values (42 nm and 35.6 nm) are close to 40 nm, the mean size of the α -Al₂O₃ nanoparticles.

3.2 Flux-Trapping Curves and Doping Effects

Figure 4 displays the different behaviors of the magnetic trapping flux starting from the results of the critical current density dependence as a function of the maximum

Fig. 1 X-ray diffraction patterns of the samples **B00**, **B03**, and **B05**



applied magnetic field for the three studied samples. These differences can be summarized in the following way [10]:

1. Samples **B03** and **B05** begin to trap the magnetic flux in H_{am} lower than that of the pristine sample **B00**.
2. Sample **B03** exhibits the minor relative values of the critical current density in the H_{am} region $35 \text{ Oe} \leq H_{am} \leq 65 \text{ Oe}$.
3. The flux-trapping phenomenon occurs more abruptly in sample **B03**.

Although our samples have percentages of extra phases between 4 and 16% according to Table 1, an appreciable change in the magnetic flux trapping is mainly observed for the sample **B03**. It confirms that the trapping of the magnetic flux not only depends on the percentage of the extra-phases but also depends on its distribution, at nanometric scale, inside the superconducting matrix. Figure 3 is a good example for the case of $\alpha\text{-Al}_2\text{O}_3$ nanoparticles. A similar result for the flux pinning has been reported elsewhere [12].

In the case of Fig. 5, the model proposed in reference [6] has been applied. The main features of these curves are as follows: (i) the two $\alpha\text{-Al}_2\text{O}_3$ -doped samples start to trap flux at $\theta_{\min} \sim 15^\circ$, a feature not observed in sample **B00**, in which magnetic flux starts to trap in the vicinity of 12° ;

(ii) sample **B03** displays a sudden drop in the normalized critical current density close to 12° ; (iii) the normalized critical current density increases with increasing θ_{\min} in all samples; (iv) sample **B03** displays an almost constant value of the normalized critical current density for $6^\circ < \theta_{\min} < 10^\circ$; (v) sample **B03** clearly displays the most pronounced change in its $J_c(0, H_{am})/J_c(0)$ vs. θ_{\min} dependence. Here, the negative values of the angles correspond to values of maximum applied magnetic field where the transition AJ to A vortices happens and the crystallites are penetrated by the magnetic flux as discussed in reference [6]. The different behaviors regarding the magnetic flux trapping will be discussed in detail later.

3.3 Generalizing a Previous Result

As already reported [6], the maximum applied magnetic field H_{am} may be expressed as a function of the minimum penetrated misorientation angle by the following equation:

$$\theta_1 = -\theta_0 \left[\ln \left(\frac{3\sqrt{3}}{4\kappa} \right) + a \ln \kappa + a\gamma_1 - \gamma_2 \right]. \quad (1)$$

Here, θ_0 , κ , γ_1 , and γ_2 are the misorientation angles, 5° (Ref. [14]), 83 (Ref. [15]), 0.497 (Ref. [16]), and 0.423 (Ref. [17]), respectively. As a is H_{c1d} normalized to the

Table 1 Phase composition of the Bi-2223 ceramic samples

Samples	Bi-2223 (%)	Extra phases (%)
B00	84	16
B03	90	10
B05	96	4

Table 2 Crystallographic data of each sample

Samples	a (Å)	b (Å)	c (Å)
B00	5.604	4.767	37.166
B03	5.618	4.564	37.898
B05	5.608	5.207	37.197

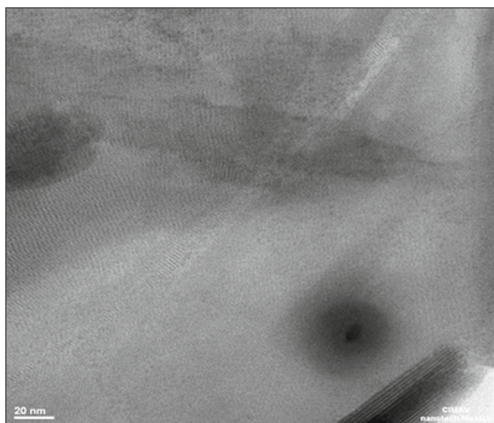


Fig. 2 HRTEM image of the sample **B03**. The black circles correspond to α - Al_2O_3 nanoparticles

lower critical field of the crystallites (known as intragranular region free of defects), then we have obtained the desired function.

As shown in a previous paper [6], the relative error in determining θ from (1) is approximately 6% and the average of geometric factors of the grains G is considered to be approximately 0.

To express (1) in a more general form, we introduce a factor β ($\beta < 1$) in (2) as follows:

$$J_s = \beta J_d \exp\left(-\frac{\theta}{\theta_0}\right). \quad (2)$$

This factor represents the effect of the transparency of the junctions (defects) and also models the decrease of J_d when the junction thickness is greater than the London penetration depth. Both conditions may be presented in any junction of a ceramic superconducting material. Thus, taking into

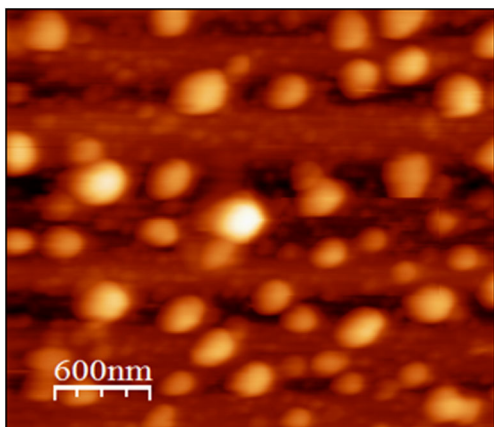


Fig. 3 AFM image of the sample **B03**. The white particles correspond to α - Al_2O_3 nanoparticles

account the results of Ref. [18], one obtains (3) more general than (1):

$$\theta_2 = -\theta_0 \left[\ln\left(\frac{3\sqrt{3}}{4\kappa\beta}\right) + a \ln \kappa + a\gamma_1 - \gamma_2 \right]. \quad (3)$$

Figure 6 displays the theoretical dependence of the misorientation angle versus the parameter β in the magnetic field window where AJ vortices are majority ($40 \text{ Oe} < H_{c1c} < 80 \text{ Oe}$). It is the result of plotting (3) to different values of β from 0.05 to 1 with a step size of 0.05, for five different values of the parameter a . The interpretation of Fig. 6 allows us to determine the permissible values of β . As the misorientation angle should be positive, hence for $a = 0.5$, β may take values inside the interval $0.15 \leq \beta < 1$. For $a = 1$, no allowed values for β are expected.

3.4 On The Role of Doping with Nanoparticles in the Flux-Trapping Mechanism

In order to clarify the role of doping with α - Al_2O_3 nanoparticles in the flux-trapping mechanism of Bi-2223, we should compare both: the required energy to create an AJ vortex in a low-angle boundary and that required to create a similar one, but in the vicinity of a pore (nanoparticle with certain dimensions). We will first focus on determining the lower critical field of the pore, H_{c1p} .

Regarding cylindrical pores, as a first approximation, and taking into account the Helmholtz equation [19], where α is the radius of cylinder, one obtains:

$$B_z = \frac{\phi_0}{2\pi\lambda^2} \left[\frac{K_0(\rho/\lambda)}{1/2 \frac{\alpha^2}{\lambda^2} K_0(\alpha/\lambda) + \frac{\alpha}{\lambda} K_1(\alpha/\lambda)} \right]. \quad (4)$$

On the other hand, the generalized (anisotropic) expression for the energy per unit length, (W_l), equation 6.226 in Ref. [19], is:

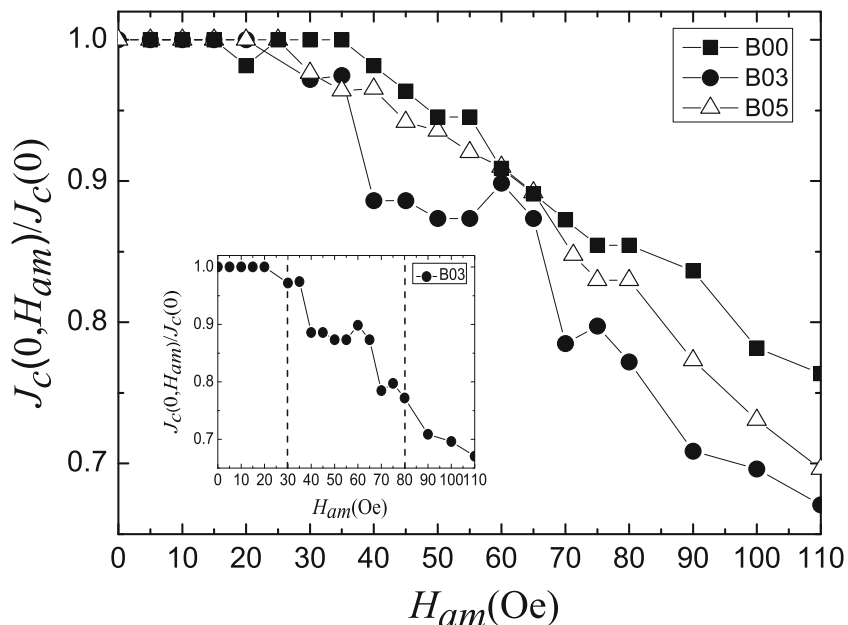
$$W_l = \frac{1}{2\mu_0} \int \mathbf{B} \cdot \mathbf{V} dx dy, \quad (5)$$

where $\vec{V} = \phi_0 \delta(x) \delta(y) \hat{k}$ is the vorticity and, dividing by ϕ_0 , the lower critical field for the vortex in the vicinity of the pore is given by the expression:

$$H_{c1p} = \frac{\phi_0}{4\pi\mu_0\lambda^2} \left[\frac{K_0(\alpha/\lambda)}{1/2 \frac{\alpha^2}{\lambda^2} K_0(\alpha/\lambda) + \frac{\alpha}{\lambda} K_1(\alpha/\lambda)} \right]. \quad (6)$$

In order to compare the lower critical fields of pores and defects, we can obtain those expressions normalized with respect to the lower critical field of the crystallites by using equations to the lower critical fields of the crystallites and

Fig. 4 Normalized critical current density as a function of H_{am} of the three samples studied: **B00**, **B03**, and **B05**. The inset displays the same dependence of the sample **B03**



the defects, respectively, as they are written in reference [18]:

$$H_{c1c} = \frac{\phi_0 \left[\ln \left(\frac{\lambda}{\xi} \right) + \gamma_1 \right]}{4\pi \mu_0 \lambda^2} \tag{7}$$

and:

$$H_{c1d} = \frac{\phi_0 \left[\ln \left(\frac{\lambda}{\xi} \right) + \gamma_2 \right]}{4\pi \mu_0 \lambda^2} \tag{8}$$

Thus, we may divide (6) and (8) by (7) and obtain:

$$\frac{H_{c1p}}{H_{c1c}} = \frac{1}{\ln \kappa + \gamma_1} \left[\frac{K_0(\alpha/\lambda)}{1/2 \frac{\alpha^2}{\lambda^2} K_0(\alpha/\lambda) + \frac{\alpha}{\lambda} K_1(\alpha/\lambda)} \right] \tag{9}$$

and

$$\frac{H_{c1d}}{H_{c1c}} = \frac{\ln \left(\frac{\lambda}{\xi} \right) + \gamma_2}{\ln(\kappa) + \gamma_1} \tag{10}$$

Fig. 5 Flux-trapping curves of samples **B00**, **B03**, and **B05**. The independent variable H_{am} has been replaced by θ_{min} following (1). Continuous lines divide the graphs into four different regions according to the flux-trapping mechanism

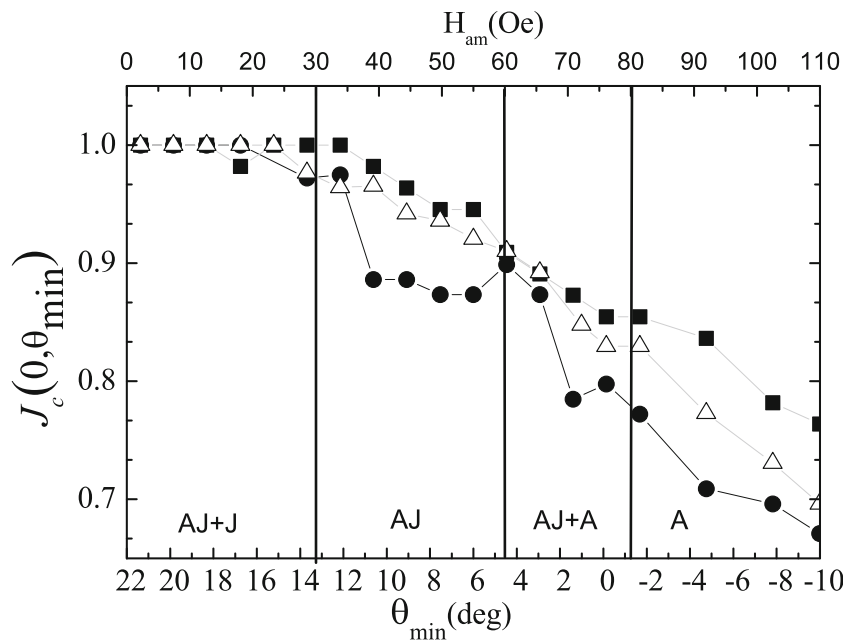
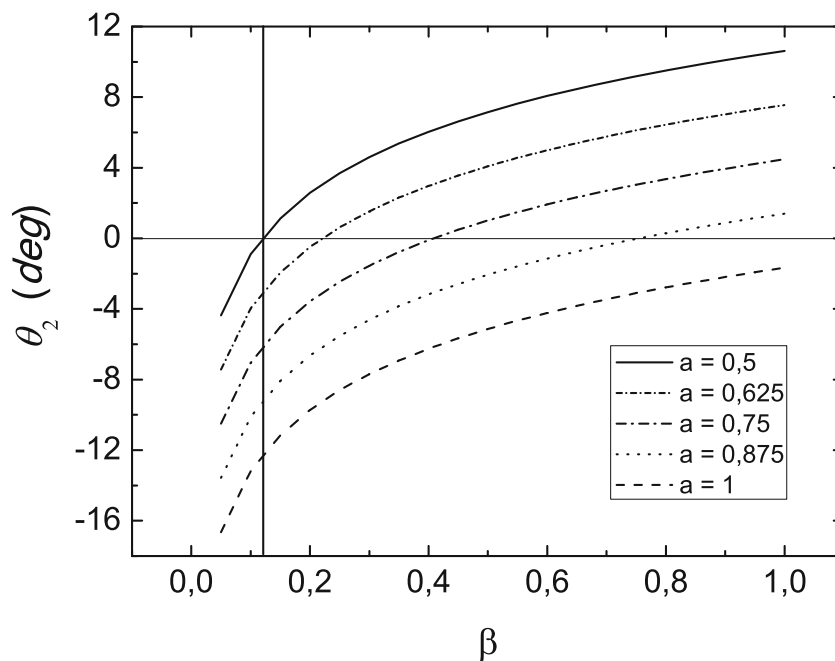


Fig. 6 Misorientation angle versus β dependence for different values of a



Applying the curl to (4), the critical current density as a function of the radial coordinate is obtained:

$$J_p = \frac{\phi_0}{2\pi\mu_0\lambda^3} \left[\frac{K_1(\rho/\lambda)}{1/2\frac{\alpha^2}{\lambda^2} K_0(\alpha/\lambda) + \frac{\alpha}{\lambda} K_1(\alpha/\lambda)} \right], \quad (11)$$

the denominator of the second factor approaches 1 for small values of the Bessel function parameter [19], so one can obtain:

$$J_p = \frac{\phi_0}{2\pi\mu_0\lambda^3} K_1(\rho/\lambda). \quad (12)$$

The previous expression evaluated at the radius of the pore $\rho = \alpha$ yields the maximum current density of the vortex:

$$J_{pmax} = \frac{\phi_0}{2\pi\mu_0\lambda^3} K_1(\alpha/\lambda). \quad (13)$$

Moreover, the critical current density of the defects as a function of the grain boundary angle is:

$$J_{cd} = \frac{\phi_0}{2\pi\mu_0\lambda^3} K_1(\xi/\lambda)\beta \exp\left(-\frac{\theta}{\theta_0}\right). \quad (14)$$

As $J_{pmax} \leq J_{cd}$, then one obtains:

$$K_1(\alpha/\lambda) \leq K_1(\xi/\lambda)\beta \exp\left(-\frac{\theta}{\theta_0}\right). \quad (15)$$

For small arguments $K_1(x) = (1/x)$ and:

$$\alpha\beta \geq \xi \exp\left(\frac{\theta}{\theta_0}\right) \quad (16)$$

On the other hand, taking into account that the lower critical field is proportional to the energy per unit length as follows:

$$H_{c1d} = \frac{W_{c1d}}{\phi_0}. \quad (17)$$

One can write this equation for the pores and crystallites. After that, we may normalize the equations for the clusters and pores with respect to the equation of crystallites:

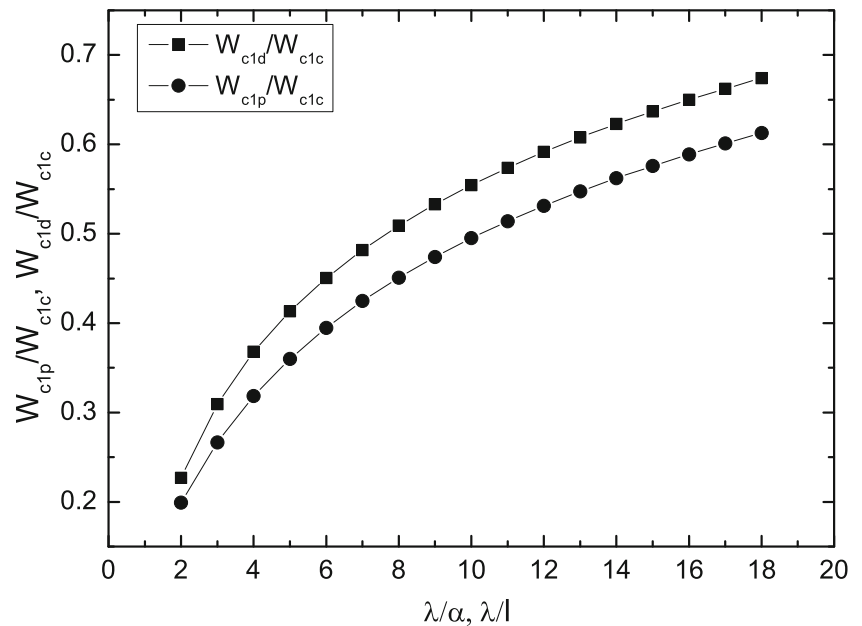
$$\frac{H_{c1d}}{H_{c1c}} = \frac{W_{c1d}}{W_{c1c}}, \quad (18)$$

$$\frac{H_{c1p}}{H_{c1c}} = \frac{W_{c1p}}{W_{c1c}}, \quad (19)$$

Figure 7 displays both magnitudes as a function of the ratios $\frac{\lambda}{l}$ and $\frac{\lambda}{\alpha}$, respectively. The main result here is that the energy of the vortices in the vicinity of a pore is lower than that of the vortices in the low-angle boundary free of pores.

Based on the theoretical framework described above, the next step is to explain our experimental results displayed in the flux-trapping curves. First of all, the presence of α - Al_2O_3 nanoparticles in the matrix of Bi-2223 is responsible for a decrease in the energy to create an AJ vortex, which can be explained by using the results displayed in Fig. 7. Under this circumstance, the doped samples start trapping flux for lower applied magnetic fields than the pristine sample. Secondly, taking $\beta = 1$ in the expression (16), it is possible to determine if the inequality is fulfilled. As the model assumes pores located at GBs (planar defects), it is reasonable to assume that our samples have misorientation angles $\theta \sim 10^\circ$. Then, taking $\theta_0 = 5^\circ$ [14] and $\xi \sim 3$ nm [20], one obtains $\alpha \geq 22$ nm. This result is of interest

Fig. 7 Energy per unit length of an AJ vortex and for a vortex in a pore normalized to the energy per unit length of an A vortex vs λ/l and λ/α dependence, respectively



and explains well the sudden drop in the flux-trapping curve of sample **B03**. The presence of nanoparticles is responsible for an improvement of the flux trapping as a consequence of the increase of the pinning energy, but such an effect is critical starting from the fulfillment of the condition given by the expression (16). It demonstrates that the nanoparticles used for doping our samples, with mean diameter ~ 40 nm, favor the increase of the trapping of AJ vortices, which emerge at low-angle boundaries. Finally, sample **B05** does not exhibit the same behavior of the **B03** due to a decrease of the transparency of the junctions, a feature caused by the excess of α -Al₂O₃ nanoparticles throughout the material. Within this scenario, the apparent size of the nanoparticales ($\alpha\beta$) decreases and the condition given by expression (16) is not fulfilled in the range of the low-angle boundaries. For this reason, the general superconducting behavior of sample **B05** is similar to that observed in sample **B00**.

4 Conclusions

We successfully correlated the effects of the size of α -Al₂O₃ nanoparticles and the superconducting properties of Bi-2223 to get an effective pinning of the AJ vortices in the intragranular low-angle boundaries. This correlation was established by comparison of the energies required to create an AJ vortex in a planar defect free of nanoparticles with that required for the presence of α -Al₂O₃ nanoparticles. The α -Al₂O₃ nanoparticles embedded in the superconducting matrix have two main effects related to the parameters of the model described in this paper: (a) an increase of the pinning

centers, mainly for the AJ vortices due to the size of the nanoparticle, α , and the angle of the planar defects, $\theta \leq 10^\circ$; (b) a decrease of the junction transparency, β , which can be modified by the excess of α -Al₂O₃ nanoparticles within the Bi-2223 matrix. The model captures important features of the combined data and is sufficient to explain why the flux trapping and pinning of the AJ vortices first increase with the α -Al₂O₃ nanoparticle content, and then decreases for values above a certain percentage in weight in the samples, as has been observed by other authors [12]. Although the model described here has been applied to a specific doping and superconducting matrix, it may be useful in other superconducting systems.

Acknowledgments The authors are indebted to Prof. E. Martínez-Guerra and César Cutberto Leyva Porras, from Centro Investigación en Materiales Avanzados S. C., Unidad Monterrey-PIIT and Chihuahua, respectively, for the HRTEM images. We also thank Prof. F. Solís-Pomar, from Centro de Investigación en Ciencias Físico-Matemáticas, FCFM, UANL, Monterrey, Nuevo León, México, for the AFM images. Finally, the authors would like to express their deepest gratitude to Prof. A. Gurevich for several important suggestions.

Funding Information This work was supported by Brazil's agencies FAPESP (Grant nos. 13/07296-2 and 14/19245-6), CNPq, and CAPES under Grant CAPES/MES no. 104/10.

References

1. Gurevich, A.: Nonlocal Josephson electrodynamics and pinning in superconductors. *Phys. Rev. B* **46**, R3187 (1992)
2. Gurevich, A., Rzchowski, M.S., Daniels, G., Patnaik, S., Hinaus, B.M., Carrillo, F., Tafuri, F., Larbalestier, D.: Flux flow of Abrikosov-Josephson vortices along grain boundaries in high temperature superconductors. *Phys. Rev. Lett.* **88**, 97001 (2002)

3. Hänisch, J., Attenberger, A., Holzapfel, B., Shultz, L.: Electrical transport properties of $Bi_2Sr_2Ca_2Cu_3O_{10+\delta}$ thin film [001] tilt grain boundaries. *Phys. Rev. B* **65**, 052507 (2002)
4. Held, R., Schneider, C.W., Mannhart, J., Allard, L.F., More, K.L., Goyal, A.: Low-angle grain boundaries in $YBa_2Cu_3O_{7-\delta}$ with high critical current densities. *Phys. Rev. B* **79**, 014515 (2009)
5. Hernández-Wolpez, M., Cruz-García, A., Vázquez-Robaina, O., Jardim, R.F., Muné, P.: Penetration and trapping of the magnetic flux in planar defects of $Bi_{1.65}Pb_{0.35}Sr_2Ca_{2+x}Cu_{3+x}O_y$ superconductor. *Phys. C* **525–526**, 84 (2016)
6. Hernández-Wolpez, M., Cruz-García, A., Jardim, R.F., Muné, P.: Intragranular defects and Abrikosov-Josepson vortices in Bi-2223 bulk superconductors. *J. Mater Sci:Mater Electron*. **28**, 15246 (2017)
7. Altshuler, E., García, S., Barroso, J.: Flux trapping in transport measurements of $YBa_2Cu_3O_{7-x}$ superconductors A fingerprint of intragrain properties. *Phys. C* **177**, 61 (1991)
8. Muné, P., Govea-Alcaide, R.F.: Jardim. Influence of the compacting pressure on the dependence of the critical current with magnetic field in polycrystalline $(Bi - Pb)_2Sr_2Ca_2Cu_3O_x$. *Phys. C* **384**, 491 (2003)
9. Hernández-Wolpez, M., García-Fornaris, I., Govea-Alcaide, E., Jardim, R.F., Muné, P.: Some effects of the doping α - Al_2O_3 nanoparticles in the transport properties of $Bi_{1.65}Pb_{0.35}Sr_2Ca_{2+x}Cu_{3+x}O_y$ ceramic superconductors. *Rev. Mex. Fis.* **64**, 127 (2018)
10. Hernández-Wolpez, M., Fernández-Gamboa, J.R., García-Fornaris, I., Govea-Alcaide, E., Pérez-Tijerina, E., Jardim, R.F., Muné, P.: Voltage relaxation and Abrikosov-Josepson vortices in Bi-2223 superconductors doped with α - Al_2O_3 nanoparticles. *J. Mater Sci:Mater Electron*. **29**, 5926 (2018)
11. Li, M.Y., Han, Z.: The effect of lead-rich phases on the microstructure and properties of (Bi, Pb)-2223/Ag tapes. *Supercond. Sci. Technol.* **20**, 843 (2007)
12. Ghattas, A., Annabi, M., Zouaoui, M., Azzouz, F.B., Salem, M.B.: Flux pinning by Al-based nano particles embedded in polycrystalline (Bi, Pb)-2223 superconductors. *Phys. C* **468**, 31 (2008)
13. Horcas, I., Fernández, R., Gómez-Rodríguez, J.M., Colchero, J., Gómez-Herrero, J., Baroet, A.M.: WSxM: A software for scanning probe microscopy and a tool for nanotechnology. *Rev. Sci. Instrum.* **78**, 013705 (2007)
14. Hilgenkamp, H., Mannhart, J.: Grain boundaries in high- T_c , superconductors. *Rev. Mod. Phys.* **74**, 485 (2002)
15. Majoros, M., Martini, L., Zainella, S.: Lower critical magnetic field H_{c1} and intergranular coupling of Bi-2223/Ag concentric tapes. *Phys. C* **282**, 2205 (1997)
16. Hu, C.-R.: Numerical constants for isolated vortices in superconductors. *Phys. Rev. B* **6**, 1756 (1972)
17. This value of $\gamma_2 = 0.423$ differs from the incorrect value $\gamma_2 = 1.118$ of (Ref. [1])
18. Gurevich, A.: Nonlinear viscous motion of vortices in Josephson contacts. *Phys. Rev. B* **48**, R12857 (1993)
19. Orlando, T.P., Delin, K.A.: Foundations of applied superconductivity. Adisson- Wesley, New York (1991)
20. Yavuz, S., Bilgili, O., Kocabas, K.: Effects of superconducting parameters of SnO_2 nanoparticles addition on (Bi, Pb)-2223 phase. *J. Mater. Sci: Mater. Electron*. **27**, 4526 (2016)

Publisher's Note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.

Journal of Superconductivity & Novel Magnetism is a copyright of Springer, 2020. All Rights Reserved.