



Interface matters: Design of an efficient α -Ag₂WO₄/Ag₃PO₄ photocatalyst

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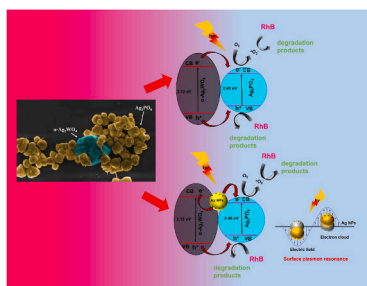
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HIGHLIGHTS

- α -Ag₂WO₄/Ag₃PO₄ heterojunctions were synthesized by the co-precipitation method.
- Heterojunction with 24-wt% α -Ag₂WO₄ degrades RhB (94.3%) in 5 min in visible light irradiation.
- Surface plasmon resonance effect by Ag nanoparticles promotes improvement in charge carrier separation.

GRAPHICAL ABSTRACT



ARTICLE INFO

Keywords:

α -Ag₂WO₄/Ag₃PO₄
Photocatalytic activity
Surface plasmon resonance
Type I heterojunction

ABSTRACT

Heterojunction engineering of complex metal oxides is an active area of research that addresses fundamental questions in solid-state systems with broad technological applications. In this work, α -Ag₂WO₄/Ag₃PO₄ heterojunctions with different amounts of α -Ag₂WO₄ (12, 24, and 36 wt%) were synthesized by the coprecipitation method and characterized by X-ray diffraction, X-ray photoelectron spectroscopy, field emission scanning electron microscopy, transmission electron microscopy, UV-vis diffuse reflectance spectroscopy, and photoluminescence. The α -Ag₂WO₄/Ag₃PO₄ heterojunction containing 24% wt of α -Ag₂WO₄ showed the most enhanced photocatalytic activity for the degradation of Rhodamine B, being much higher than Ag₃PO₄ and α -Ag₂WO₄. Trapping experiments revealed that the holes and superoxide radical, in minor extent, were the main active species in the photocatalytic degradation. Such enhanced photocatalytic performance was explained by the surface plasmon resonance effect associated with the presence of metallic Ag at the interface and the formation of a type I heterojunction between α -Ag₂WO₄ and Ag₃PO₄ semiconductors.

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<https://doi.org/10.1016/j.matchemphys.2022.125710>

Received 20 September 2021; Received in revised form 27 December 2021; Accepted 4 January 2022

Available online 10 February 2022

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1. Introduction

Over the past few years, semiconductor photocatalysis has attracted much interest due to the synthesis of innovative materials with excellent performance for organic pollutant degradation and waste water cleaning [1,2]. A highly efficient semiconductor needs to have a wide range of light absorption, a fast charge separation of photo-generated electron-hole (e^- - h^+) pairs and a strong redox ability [3-7]. However, it is difficult for a single-component photocatalyst to possess all these characteristics.

To overcome these drawbacks and improve the performance of a single semiconductor, in the last years two feasible methods emerged via a synergistic interaction/effect [8]. The first refers to the formation of a heterojunction involving the combination of two or more semiconductors to form new materials [9-17]. This strategy constitutes a hotspot in recent research [1,18-24] to obtain new materials with superior electrical, optical, and catalytic properties associated with the transfer of interfacial charge and a synergistic effect between coupled semiconductors [25]. The second concerns the deposition of active metal nanoparticles (NPs) (for example, noble metals, such as Ag, Au or/and Cu nanoparticles), displaying localized surface plasmon resonance (SPR) effect on the semiconductor's surface [26,27]. In the first case, the photocatalytic activity of the heterojunction depends on the semiconductor type (p or n), chemical composition, band alignment, and location of valence/conduction band potentials.

The combination of metal and semiconductor significantly enhances the activity because of two main features: the formation of a Schottky barrier between the metal NPs, resulting in the increased separation efficiency of photogenerated (e^- - h^+) pairs [28]. The high electron trapping ability of NPs help to effectively promote the conductivity of NPs, which can then be directly injected into the conduction band of the semiconductors, facilitating the charge separation at the interface between NPs and the semiconductor, and consequently improving its photocatalytic ability [29-42].

The interface causes induced e^- and h^+ in or near this area to move in opposite directions, promoting charge transfer and suppressing the recombination of e^- - h^+ pairs, thus increasing their stability [43] and maximizing the redox power for heterojunction photocatalysts. This quantum phenomenon occurs in a confined region and is capable of enhancing the simultaneous (successive or parallel) transformation of compounds. As it occurs in other Ag-based semiconductors, both Ag_3PO_4 and $\alpha-Ag_2WO_4$ are unstable and tend to segregate Ag metal nanoparticles (Ag NPs) on the surfaces of these complex oxides, hence forming Ag NPs/ Ag_3PO_4 and Ag NPs/ $\alpha-Ag_2WO_4$ composites, respectively. These composites present SPR effect associated with the easy formation of Ag NPs on the surface of these semiconductors [30,32,37,41,44-48].

Despite the practical importance of such composites and the fact that they have been observed for different types of heterojunction [49], so far there is a lack of studies in the literature on a consistent pattern demonstrating the synergistic influence of plasmons of Ag NPs on their photocatalytic activity. Inspired by the unique properties of both Ag_3PO_4 and $\alpha-Ag_2WO_4$, it was envisaged that the coupling of the heterojunction of both materials with presence of Ag NPs at the interface could provide additional active sites, significantly improving their photocatalytic property. Therefore, the aim of the present work was to obtain a $\alpha-Ag_2WO_4/Ag_3PO_4$ heterojunction via a facile coprecipitation method in order to contribute to the discussions on the topic. To this end, a wide range of $\alpha-Ag_2WO_4$ compositions (12, 24, and 36 wt%) was used. The as-synthesized heterojunctions were fully characterized by X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), transmission electron microscopy (TEM), field emission gun scanning electron microscopy (FE-SEM) and diffuse reflectance spectroscopy in the ultraviolet-visible region (UV-vis). This work can also provide important understanding on the role played by Ag NPs at the interface of these heterojunctions during the photocatalytic process of Rhodamine B

(RhB) dye degradation. After systematically characterizing and evaluating the properties of the photocatalyst, it was possible to observe that this heterojunction exhibited higher photoactivity than pure counterparts. Based on the scavenger results, the photocatalytic mechanism was proposed and discussed in detail. Finally, the stability of the material was studied via recycling and reusability experiments.

2. Experimental section

2.1. Synthesis

Ag_3PO_4 sample was synthesized by the coprecipitation method in aqueous medium at 30 °C. 50 mL of deionized water and $(NH_4)_2HPO_4$ salt (0.001 mol) (98.6%, JT Baker) were added to a beaker. This solution was stirred at 30 °C for 10 min for complete salt dissolution. In another beaker, the same procedure was performed using 50 mL of deionized water and $AgNO_3$ salt (0.003 mol) (99.8%, Vetec). After salt dissolution, the solution containing $AgNO_3$ was added to the $(NH_4)_2HPO_4$ solution and kept under stirring for 10 min at 30 °C. Subsequently, the formed precipitate was washed with deionized water to remove residual ions, and then dried in a conventional oven at 60 °C for 12 h.

The $\alpha-Ag_2WO_4$ crystals were also prepared by the coprecipitation method at room temperature. $Na_2WO_4 \cdot 2H_2O$ (0.007 M) (99.5%, Sigma-Aldrich) and $AgNO_3$ (0.0035 M) (99.8%, Sigma Aldrich) were stirred separately in a solution containing 50 mL of deionized water at room temperature until the complete dissolution of the salts. Afterwards, the $AgNO_3$ solution was added to the Na_2WO_4 solution and stirred for 10 min. The formed precipitate was washed with deionized water to remove residual ions, and then dried in a conventional oven at 60 °C for 12 h.

Stoichiometric contents of precursors reagents for $\alpha-Ag_2WO_4$ and Ag_3PO_4 were used to obtain $\alpha-Ag_2WO_4/Ag_3PO_4$ heterojunction with 12, 24 and 36 wt% of $\alpha-Ag_2WO_4$ in relation to Ag_3PO_4 , which were denominated as AWP 1, AWP 2 and AWP 3, respectively. For this purpose, $\alpha-Ag_2WO_4$ was dispersed in a beaker containing 20 mL of deionized water at 30 °C under constant agitation. In another beaker, $(NH_4)_2HPO_4$ salt was dissolved in 20 mL of deionized water at 30 °C under constant stirring. This solution was added to the beaker containing the dispersed $\alpha-Ag_2WO_4$ and kept under stirring at 30 °C for 10 min. Another solution was prepared using 20 mL of deionized water and $AgNO_3$, which was dissolved under stirring at 30 °C. This solution was dripped onto the suspension containing $\alpha-Ag_2WO_4$ and $(NH_4)_2HPO_4$, and the whole mixture was stirred for 10 min. The precipitate formed was washed with deionized water to remove residual ions, and then dried in a conventional oven at 60 °C for 12 h.

2.2. Characterization

The obtained materials were characterized by XRD diffraction using a D/Max-2500PC diffractometer (Rigaku, Japan) with $CuK\alpha$ radiation ($\lambda = 1.54056 \text{ \AA}$) in the 2θ range of 10° - 80° at a scanning speed of 1° min^{-1} and step size of 0.02° . Transmission electron microscopy (TEM) was performed using a FEI Tecnai G2F20 (Netherlands) microscope operating at 200 kV. Dark field (DF) image as well as local compositional analyses and mapping via energy-dispersive X-ray spectroscopy (EDS) were recorded in the scanning TEM (STEM) mode. The morphologies of the samples were characterized by field emission gun scanning electron microscopy (FE-SEM) in a FEI instrument (Inspection Model F50) operating at 10 kV.

Measurements of X-ray photoelectron spectroscopy (XPS) were performed on a Scientia Omicron ESCA + (Germany) spectrometer using monochromatic $Al K\alpha$ (1486.7 eV). The maximum deconvolution was performed using a Gaussian (70%)-Lorentzian (30%) line with Shirley nonlinear sigmoid-type baseline. For calibration of the binding energy of the elements, peak C 1s at 284.8 eV was used as a reference. To obtain ultraviolet-visible (UV-vis) absorption spectra, a Varian Cary 5G

(United States) spectrophotometer was used in diffuse reflection mode.

Photoluminescence (PL) measurements were performed by using a 500 M SPEX spectrometer coupled to a GaAs-PMT detector. A Kimmon He-Cd laser (325 nm line) was used as excitation source. The PL measurements were performed in the range of 380–750 nm with laser power of about 16 mW and at 300 K.

2.3. Photocatalytic measurements

The photocatalytic activity of Ag_3PO_4 , $\alpha\text{-Ag}_2\text{WO}_4$ and heterojunction samples AWP 1, AWP 2 and AWP 3 were tested for Rhodamine B (RhB) discoloration (95%, Aldrich) under visible light irradiation. For the photocatalytic experiments, 50 mg of each sample and 50 mL of RhB (10 mg L^{-1}) were placed in a beaker, and later in an ultrasonic bath (Branson, model 1510; frequency 42 kHz) for 15 min, followed by stirring for another 30 min, always keeping samples in the dark. An aliquot was collected at time 0, and the RhB solution containing the catalyst was exposed to the irradiation of 6 lamps (Philips TL-D, 15 W). The entire system was maintained at a constant temperature of 20°C . Aliquots were removed at certain times (0, 2, 5, and 10 min). All aliquots were centrifuged, and their degradation was monitored by measuring the peak of maximum RhB absorption ($\lambda_{\text{max}} = 554 \text{ nm}$) using a UV-visible spectrophotometer (V-660, JASCO). A control experiment was carried out under the same conditions, but without the presence of catalysts.

To elucidate the mechanism of $\alpha\text{-Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$ photocatalytic activity enhancement, the main active species that participated in the photocatalytic reaction were investigated. The free radicals and holes trapping experiments were carried out. The scavenger experiments were performed by the addition of *tert*-butyl alcohol (TBA, 0.012 mol/L) (Alfa Aesar), ammonium oxalate (AO, 0.012 mol/L) (Alfa Aesar) and benzoquinone (BQ, 0.012 mol/L) (Alfa Aesar) as scavengers of hydroxyl radical ($\bullet\text{OH}$), hole (h^+) and superoxide radical ($\bullet\text{O}_2^-$), respectively.

3. Results and discussion

3.1. Powders characterization

The Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ were synthesized by the coprecipitation method, as previously reported [50,51]. The heterojunctions containing 12, 24 and 36 wt% of $\alpha\text{-Ag}_2\text{WO}_4$ in relation to Ag_3PO_4 were synthesized as shown in Fig. 1, and denoted as AWP 1, AWP 2 and AWP 3, respectively.

Fig. S1, in Supplementary Information, shows the XRD patterns of the Ag_3PO_4 , AWP 1, AWP 2, AWP 3 and $\alpha\text{-Ag}_2\text{WO}_4$ samples. The XRD pattern of the Ag_3PO_4 sample are in good agreement with the Inorganic Crystal Structure Database (ICSD) No. 14000, with a cubic structure and $P43n$ space group [52]. On the other hand, the $\alpha\text{-Ag}_2\text{WO}_4$ sample has an orthorhombic structure with a $Pn2n$ space group, according to ICSD No. 4165 [53].

It was observed that as the amount of $\alpha\text{-Ag}_2\text{WO}_4$ increased, the XRD peaks referring to this phase intensified, which is well expected due to

the higher content of this material. No characteristic peaks related to any impurities or secondary phase were noted, suggesting that the samples presented only $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 phases.

TEM analysis was performed for the sample with an intermediate amount of $\alpha\text{-Ag}_2\text{WO}_4$ (AWP 2). Fig. 2(a) shows a low-magnified DF-STEM image, where two main arrangement of particles in this sample comprising faceted nanospheres and elongated irregular microparticles covered by these nanospheres can be seen. Fig. 2(b) shows a magnified view corresponding to the dotted square marked in Fig. 2(a). An elemental characterization by EDS was conducted in the microparticle and nanospheres, as shown in Fig. 2(b). The resulting EDS spectrum for the microparticle indicates the predominance of the elements Ag, P, W and O, whereas the irregular nanospheres are mainly composed of Ag, P and O. Fig. 2(c–f) show the EDS mapping of the elements Ag, P, W and O corresponding to the region in (b), again confirming the presence of the elements referring to the heterojunction and the homogeneity of the sample.

To complement this result, selected area electron diffraction (SAED) analyses (insets in Fig. 3) were conducted in the same region of Fig. 2 (b), and as indicated by the dotted arrows they reveal patterns typically observed in polycrystalline materials. The nanospheres exhibit concentric rings that can be indexed to the (220), (310), and (520) families of planes of the cubic Ag_3PO_4 , while the patterns on the elongated microparticle reveal the presence of both (402) and (633) families of planes of the $\alpha\text{-Ag}_2\text{WO}_4$ structure, and (233) family of planes associated with the Ag_3PO_4 structure. These results confirm the successful formation of Ag_3PO_4 nanospheres and their crystallization on the surface of $\alpha\text{-Ag}_2\text{WO}_4$ microcrystals, forming the heterojunction.

Fig. 4 shows the micrographs of the pure materials and the heterojunction formed in different proportions. The Ag_3PO_4 structure, represented by the orange color, exhibits a morphology with irregular nanospheres (Fig. 4(a)), while the $\alpha\text{-Ag}_2\text{WO}_4$ structure, represented by the blue color, displays a morphology with hexagonal microrods (Fig. 4 (b)). In the heterojunction (Fig. 4(c–e)), it can be seen that the interaction of the materials during the synthesis process causes changes in the morphology of both materials. As presented in the TEM analysis, the elongated irregular blue microparticle is composed of the $\alpha\text{-Ag}_2\text{WO}_4$ structure, whereas the smaller nanosphere orange particles are formed by the Ag_3PO_4 structure. The change in the morphology of the heterojunction is due to the dissolution and recrystallization of the rods in aqueous medium, which consequently formed larger faceted microparticles. Sample AWP1 showed an agglomerate of particles that coalesced to form a large microparticle represented by the blue color, which can be attributed to the $\alpha\text{-Ag}_2\text{WO}_4$ structure (Fig. 4(c)). Samples AWP2 and AWP3, on the other hand, were characterized by the presence of large microparticles with well-defined surfaces (Fig. 4(d–e)). EDS results revealed that the $\alpha\text{-Ag}_2\text{WO}_4$ microparticles were covered by the elements Ag, P and O, with the smaller nanoparticles corresponding to the Ag_3PO_4 structure. Moreover, the Ag_3PO_4 nanoparticles presented more defined surfaces in the heterojunction.

To investigate the chemical composition and surface structure of the as-prepared samples, XPS measurements were carried out. The analysis

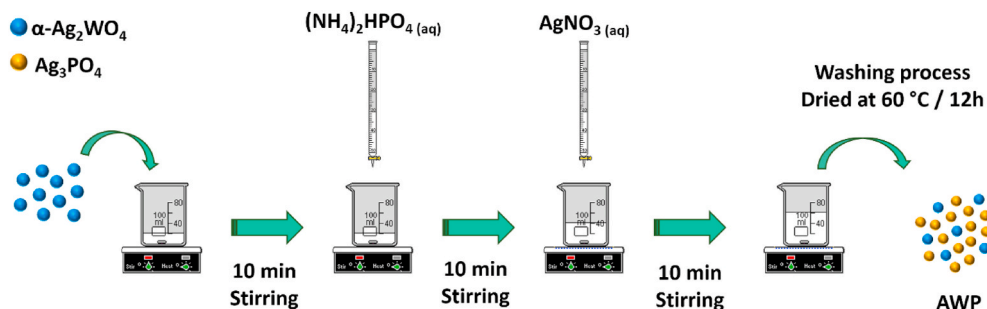


Fig. 1. Schematic representation of the synthesis procedure.

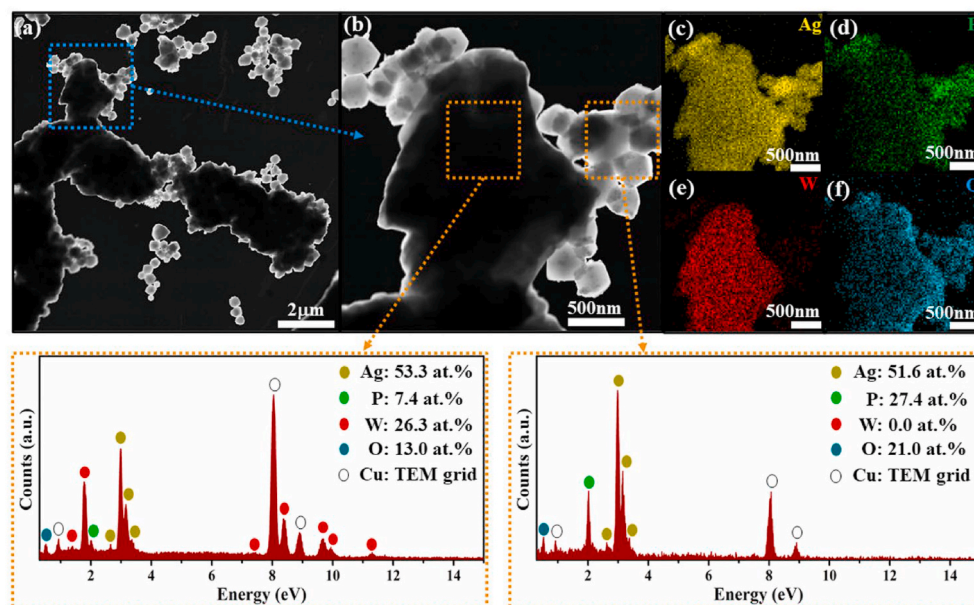


Fig. 2. TEM characterization of sample AWP 2: (a) DF-STEM image, (b) high-magnification DF-STEM image and EDS spectra obtained at Ag_3PO_4 crystallized on $\alpha\text{-Ag}_2\text{WO}_4$ surface, and (c–f) EDS mapping of the elements Ag, P, W, and O corresponding to the region in (b).

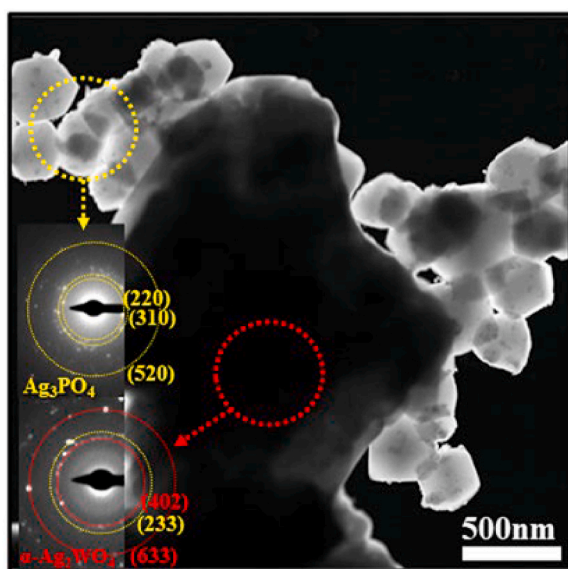


Fig. 3. DF-STEM image and its corresponding SAED patterns of sample AWP 2.

of the high-resolution spectra of the constituent elements of $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 samples is presented in Fig. S2, in Supplementary Information. The deconvolution analysis of the P 2p spectrum of Ag_3PO_4 as well as the W 4f spectrum of $\alpha\text{-Ag}_2\text{WO}_4$ indicates the presence of P^{5+} and W^{6+} oxidation states, as shown in Fig. S2(b) and Fig. S2(e), respectively [54,55]. The deconvolution analysis of the Ag 3d spectra for both Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ samples reveals the presence of Ag^+ and Ag^0 oxidation states according to their respective spin-orbit coupling peaks, as demonstrated in Fig. S2(a) and Fig. S2(d), respectively [56–58]. Fig. S2(c) and Fig. S2(f) show the O 1s spectra for the Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ samples, respectively, and their components for lattice oxygen (≈ 530 eV), oxygen vacancies (≈ 531.5 eV) and hydroxyl-bounded groups on particle surface (≈ 533 eV) [59,60].

The heterojunction with intermediate amount of $\alpha\text{-Ag}_2\text{WO}_4$ (AWP 2) was investigated by XPS. The Ag 3d spectrum of sample AWP 2 is shown in Fig. 5(a), where it is possible to see well-defined components of the

spin-orbit coupling related to both Ag^+ and Ag^0 oxidation states. The contribution of the Ag^0 oxidation state arises from the presence of metallic Ag nanoparticles on the surface of Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ particles that results in Ag metal/semiconductor interface. The deconvolution analysis of the P 2p spectrum for AWP 2 indicates the presence of P^{5+} oxidation state (Fig. 5(b)). The O 1s spectrum for AWP 2 together with its main components and the survey spectra for Ag_3PO_4 , $\alpha\text{-Ag}_2\text{WO}_4$ and AWP 2 are shown in Fig. 5(c) and (e), respectively. Variations in the broadening of the peaks of W 4f spectrum for AWP 2 in relation to pure $\alpha\text{-Ag}_2\text{WO}_4$ can be observed in Fig. 5(d) by the broadening of the peaks. The deconvolution results indicated the presence of two oxidation states for W in this sample: W^{6+} in major proportion and W^{5+} in minor proportion. The presented components were attributed to the spin-orbit coupling of $\text{W}^{6+} 4f_{7/2}$ (35.9 eV) and $\text{W}^{6+} 4f_{5/2}$ (38.0 eV), $\text{W}^{5+} 4f_{7/2}$ (35.0 eV) and $\text{W}^{5+} 4f_{5/2}$ (37.2 eV) [61], and a shoulder related to the W^{6+} loss feature (41.2 eV).

Several works have reported the effects of heterojunction formation in the surface structure of the materials, such as interfacial strain [62–65]. The presence of W^{5+} in sample AWP 2 can be attributed to interfacial strain due to lattice mismatch between $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 since this latter was grown upon $\alpha\text{-Ag}_2\text{WO}_4$ particles. Furthermore, the W^{6+} reduction can be assigned to the charge transfer effect between $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 caused by the effective formation of the heterojunction that leads to the energy level alignment [66].

It is well known that the photocatalytic activity of the photocatalysts is closely related to their light absorption ability. In this way, UV–vis diffuse reflectance spectroscopy was employed to determine the optical absorption properties of the pure Ag_3PO_4 , $\alpha\text{-Ag}_2\text{WO}_4$, and $\alpha\text{-Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$ heterojunction. The results are represented in Fig. 6. The Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ samples show an absorption edge in the visible region at 510 nm and 405 nm, respectively. It was observed in the heterojunction samples (AWP 1, AWP 2 and AWP 3) that the spectra have two absorption edge wavelengths near these absorption edges, proving the presence of both materials, and therefore the successful formation of the heterojunction. It was also noted a slight redshift in the absorption edge at 405 nm referring to $\alpha\text{-Ag}_2\text{WO}_4$, which favors the use of visible light in the photocatalytic process.

The band gap energy (E_g) of the $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 samples (Fig. S3) were calculated using the Tauc method, following Equation (1) [67]:

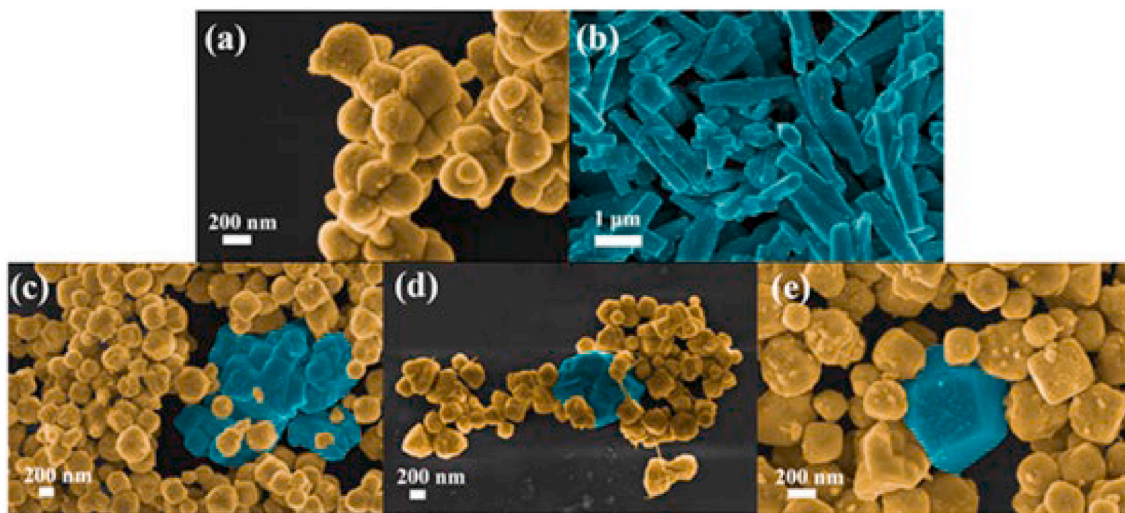


Fig. 4. FE-SEM images of (a) Ag_3PO_4 (orange color), (b) $\alpha\text{-Ag}_2\text{WO}_4$ (blue color), (c) AWP 1, (d) AWP 2 and (e) AWP 3. (For interpretation of the references to color in this figure legend, the reader is referred to the Web version of this article.)

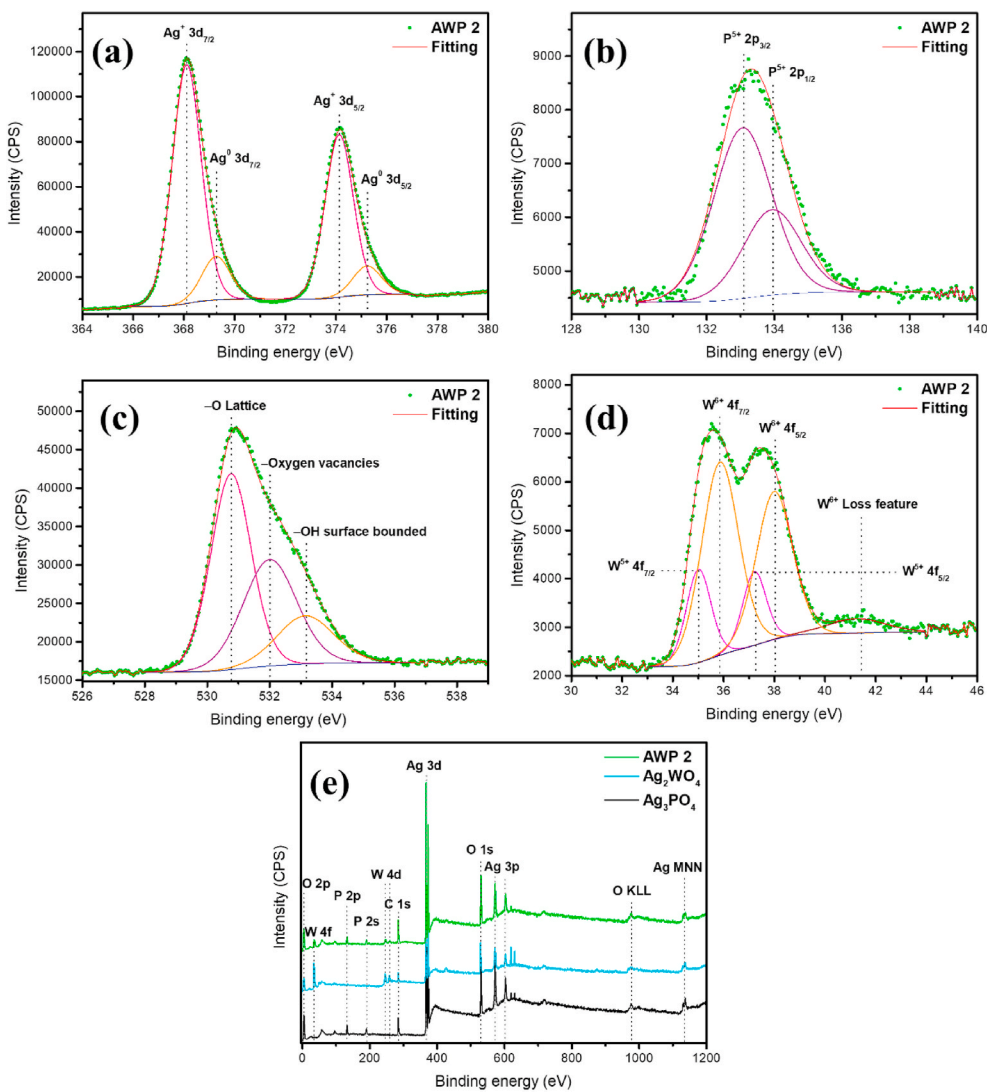


Fig. 5. High-resolution XPS spectra of (a) Ag 3d, (b) P 2p, (c) O 1s and (d) W 4f, and (e) survey of AWP 2, $\alpha\text{-Ag}_2\text{WO}_4$, and Ag_3PO_4 .

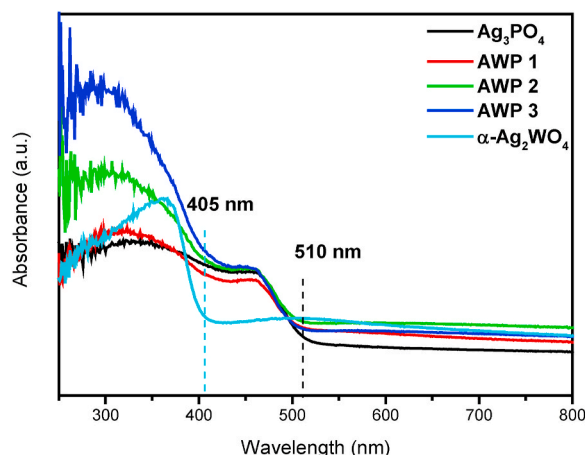


Fig. 6. UV-vis diffuse reflectance spectra of Ag_3PO_4 , AWP 1, AWP 2, AWP 3 and $\alpha\text{-Ag}_2\text{WO}_4$.

$$(\alpha h\nu)^{2/n} \sim (h\nu - E_g) \quad (1)$$

where $n = 1$ for a semiconductor with direct band gap, $n = 4$ for a semiconductor with indirect band gap, $\alpha =$ absorbance, and $h\nu$ is photon energy. Ag_3PO_4 has an indirect band gap ($n = 4$) [68] with an experimental E_g value of 2.40 eV, as shown in Fig. S3(a). The calculated E_g value is in agreement with those reported in the literature [69]. Fig. S3 (b) shows the experimental band gap of $\alpha\text{-Ag}_2\text{WO}_4$, which has direct band gap ($n = 1$) [70] with a value of 3.12 eV. This experimental E_g value is also in well agreement with those reported in the literature [71].

The photoluminescence (PL) spectrum is usually employed to investigate the separation and transfer efficiency of the photogenerated (e^-h^+) pairs during the photocatalytic process [72]. It is well known that the stronger PL intensity indicates the faster recombination speed of photogenerated charge carriers, implying that fewer photoinduced e^- and h^+ took part in the photocatalytic oxidation and reduction reactions, resulting in lower photocatalytic activity [73]. Fig. 7 exhibits the PL spectra of the as-prepared samples under the excitation wavelength of 325 nm. An analysis of the results renders that the PL spectrum of Ag_3PO_4 presents a broadband profile with the maximum emission intensity in the blue region of the visible electromagnetic spectrum at around 454 nm (2.731 eV), which is in agreement with the previous results reported in the literature [46,74]. This emission occurs mainly by the charge transfer (e^-h^+) process between the different clusters that compose the material, such as $[\text{PO}_4]$ tetrahedron clusters [46,68,75]. On the other hand, the $\alpha\text{-Ag}_2\text{WO}_4$ sample showed a PL intensity peak also in

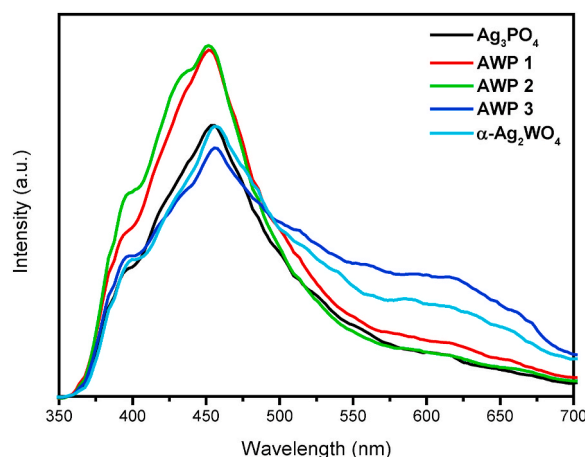


Fig. 7. PL spectra of Ag_3PO_4 , AWP 1, AWP 2, AWP 3 and $\alpha\text{-Ag}_2\text{WO}_4$ at 300 K.

the blue region centered at 456 nm (2.719 eV), and another emission in the red region with a maximum emission intensity between 570 and 650 nm (about 1.984 eV). The blue emission peak is usually related to distorted $[\text{WO}_6]$ octahedron, while the emission in the red region can be assigned to $[\text{AgO}_y]$ clusters with $y = 2, 4, 6$ and 7 , which arise from oxygen vacancies in the semiconductor [76,77]. The heterojunction samples also showed emissions in the blue region at approximately 451 nm, which are close to the blue emission from the isolated materials. Particularly, samples AWP 1 and AWP 2 exhibited a broadband profile similar to Ag_3PO_4 , while AWP 3 had a profile similar to $\alpha\text{-Ag}_2\text{WO}_4$. This result could be associated with the fact that AWP 3 has a higher amount of $\alpha\text{-Ag}_2\text{WO}_4$, resulting in a band profile similar to $\alpha\text{-Ag}_2\text{WO}_4$. It can then be concluded that the PL results are consistent with the charge transfer processes of carriers between Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ semiconductors in the heterojunction.

According to the interfacial strain resulting from synergetic effects, it was possible to observe that the $\alpha\text{-Ag}_2\text{WO}_4$ presented surface defects, such as a mix of W^{5+} and W^{6+} oxidation states (suggested by XPS experimental results), leading to a higher proportion of distorted $[\text{WO}_6]$ clusters in the heterojunction. This caused the charge recombination related to these defects to increase, resulting in a higher emission in the blue region (around 454 nm) for samples AWP 1 and AWP 2, which showed higher PL intensities than the other samples. Additionally, the e^-h^+ recombination rate in Ag_3PO_4 could be decreased, boosting its photocatalytic performance for oxidative reactions.

3.2. Photocatalytic activity

The photocatalytic performance of $\alpha\text{-Ag}_2\text{WO}_4$, Ag_3PO_4 and heterojunction samples AWP 1, AWP 2 and AWP 3 were investigated for the photodegradation of RhB dye under visible light irradiation. The degradation process was monitored by UV-vis absorption spectra at 554 nm, and the degradation curves are displayed in Fig. 8. Tests on the RhB adsorption process of all samples were performed and the results are presented Fig. S4. For this, after a period of 15 min in the ultrasound bath, the samples were under constant agitation (in the dark) for 30 min, with aliquots being removed every 10 min in order to evaluate the absorption profiles of the samples. An analysis of the results shows that the absorbance profiles are no dependent of the adsorption process. The Ag_3PO_4 sample showed a higher photocatalytic activity than $\alpha\text{-Ag}_2\text{WO}_4$, degrading approximately 70% of the dye in 10 min of reaction, while for the latter no dye degradation was observed in the same reaction time. This low photocatalytic activity of $\alpha\text{-Ag}_2\text{WO}_4$ can be attributed to the use of visible light since the material has a band gap energy that corresponds to excitation in the UV region (Fig. S3(a)). As a consequence, e^- cannot be excited from the valence band (VB) to the conduction band (CB) [78]. Conversely, Ag_3PO_4 has a band gap energy that corresponds to excitation in the visible region, generating e^-h^+ pairs under visible light irradiation, consequently degrading the dye. This makes Ag_3PO_4 a promising photocatalyst compared to $\alpha\text{-Ag}_2\text{WO}_4$, as it has the possibility of using sunlight irradiation for the degradation of organic compounds. The degradation curves for the heterojunction are also shown in Fig. 8. It can be noted that all samples presented a photocatalytic activity superior to that of the isolated materials, demonstrating that the synergistic effect was effective for the formation of a heterojunction interface, responsible for the charge carrier transfer.

To investigate the kinetics of the RhB photodegradation reaction, the Langmuir-Hinshelwood model [15] was used, considering a pseudo-first order reaction, according to equation $-\ln(C_N/C_0) = kt$, where C_N and C_0 represent the RhB concentration at different time intervals and in the initial stage, respectively, and k and t represent the rate constant and irradiation time correspondingly, respectively. Fig. 9(a) displays the variations in RhB concentration as C_N/C_0 versus irradiation time for the different samples. The photolysis experiment performed is illustrated in Fig. 8(a), where no degradation can be seen. Fig. 9(b) shows the values found for the rate constants (k) using the Langmuir-Hinshelwood model

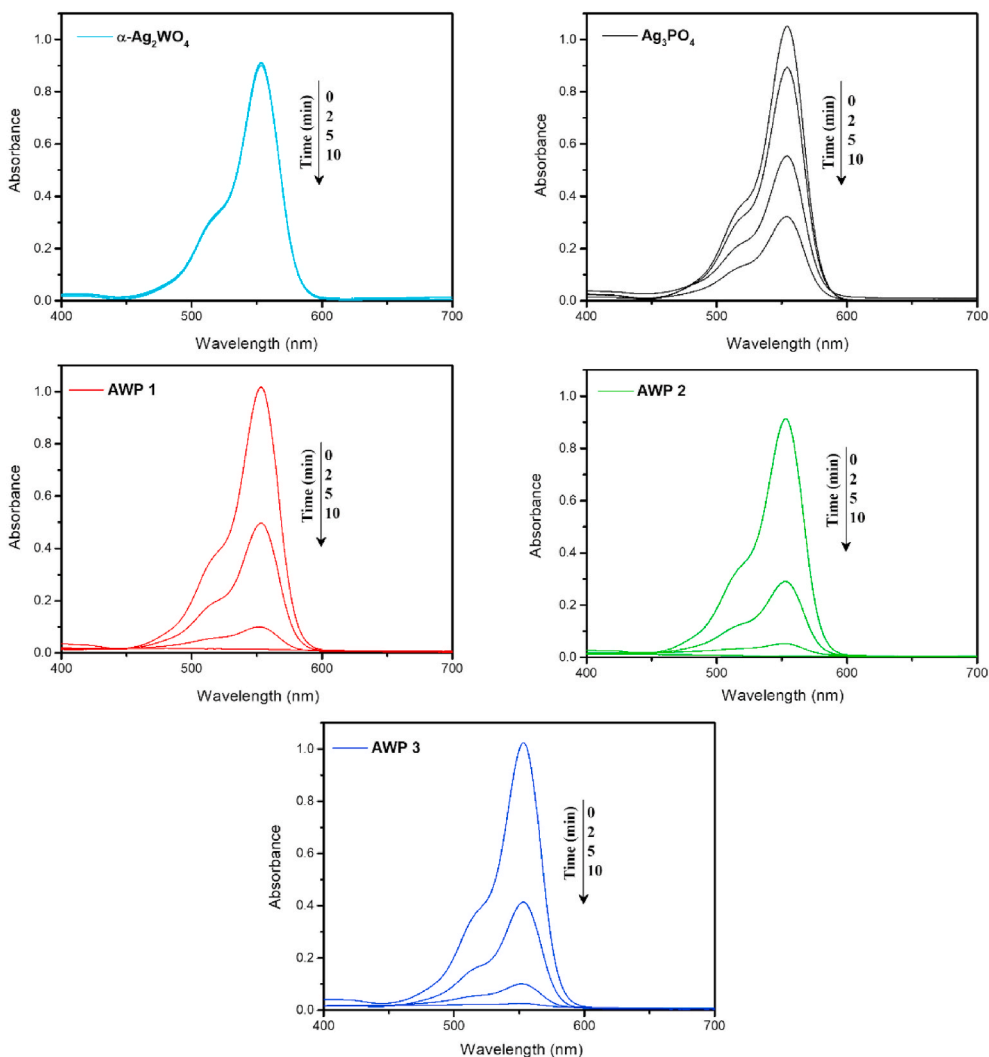


Fig. 8. UV-vis absorbance plots for RhB photocatalytic degradation in the presence of 50 mg of each catalyst and 50 mL of RhB (10 mg L^{-1}).

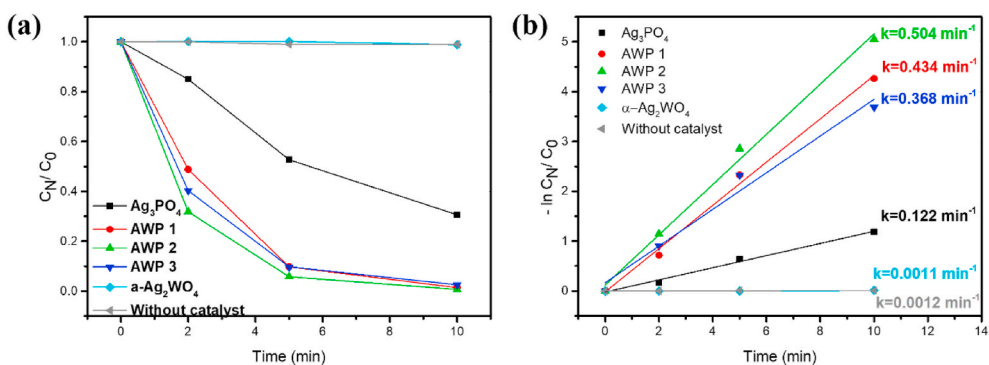


Fig. 9. Photocatalytic degradation (a) and kinetic fit curves for all samples under visible-light irradiation (b).

abovementioned. It was observed that the heterojunction presented k values higher than those of isolated materials, especially sample AWP 2, which reached a value 4.13 times higher than that of Ag_3PO_4 and 458 times higher than of $\alpha\text{-Ag}_2\text{WO}_4$, being the sample with the highest photocatalytic activity.

Note that the increased value of k in AWP 2 with respect to AWP 1 may have been due to the greater amount of $\alpha\text{-Ag}_2\text{WO}_4$ inserted into the sample, which increased the synergistic effect between $\alpha\text{-Ag}_2\text{WO}_4$ and Ag_3PO_4 , enhancing the transfer of charge carriers and improving the

photocatalytic activity of AWP 2. The value of k for AWP 3 possibly decreased in relation to AWP 2 as a result of the highest amount of $\alpha\text{-Ag}_2\text{WO}_4$ contained in this sample in relation to the other heterojunction, thus requiring a higher proportion of UV light for the separation of the charge carriers, lowering its photocatalytic activity – which was still higher than that of isolated materials since the synergistic effect between the Ag_3PO_4 and $\alpha\text{-Ag}_2\text{WO}_4$ samples prevailed. In addition, the excess of $\alpha\text{-Ag}_2\text{WO}_4$ in sample AWP 3 caused a competition between the active sites and Ag_3PO_4 for the adsorption of species.

Table 1 shows the photocatalytic efficiency of the α -Ag₂WO₄/Ag₃PO₄ heterojunction synthesized in this work in comparison with other studies that used modified Ag₃PO₄ or the composite coupled to another material for RhB photodegradation. The data presented in Table 1 indicate that α -Ag₂WO₄/Ag₃PO₄ heterojunction in this work has a remarkably higher efficiency for RhB degradation, being the material with the highest rate constant among those reported there. Although Li et al. [79] synthesized an α -Ag₂WO₄/Ag₃PO₄ heterostructure for the photodegradation of bisphenol A, however without mentioning the presence of metallic silver in its heterostructure, this is the first time that the α -Ag₂WO₄/Ag₃PO₄ heterojunction in the presence of Ag NPs is studied for RhB photodegradation under visible light irradiation.

To evaluate the stability and reuse of the photocatalyst, recycling experiments were carried out under identical conditions using the sample with the highest photocatalytic activity (AWP 2). Fig. 10 displays the five cycles performed for the RhB photodegradation. It can be noted that the first three cycles are relatively stable, with a catalyst deactivation of approximately 27% and 47% only in the fourth and fifth cycles, respectively. This decrease in the photocatalytic activity of AWP 2 may be related to the photocorrosion process that the sample is susceptible, which is a well-known phenomenon in Ag-based materials. To confirm the photocorrosion effect, the analysis of the XRD patterns of the AWP 2 sample was performed after the fifth cycle of the photocatalytic experiment and the results are presented in Fig. S5. Note that the formation of a peak located at $2\theta = 38.1^\circ$, which corresponds to the (111) diffraction plane of Ag⁰ (JCPDS Card n° 89-3722). It is believed that the loss in photocatalytic activity after the third cycle is related to the increase in the production of Ag⁰, since the excess of Ag⁰ formed in the material decreases the light absorption, and then its performance.

In this study, trapping experiments were performed in sample AWP 2 to investigate the main reactive species (Fig. 11). The scavenger experiments for $\bullet\text{O}_2^-$, $\bullet\text{OH}$ and h^+ were investigated by using the capture agents 1,4-benzoquinone (BQ), *t*-butyl alcohol (TBA) and ammonium oxalate (AO), respectively [87]. Meanwhile, the RhB degradation process was inhibited by the addition of AO; however, when BQ was added, this process was moderately suppressed, i.e., the photocatalytic rate was reduced to 48%. Furthermore, no inhibition was detected when using TBA, indicating that $\bullet\text{OH}$ does not contribute to the degradation of RhB. These results demonstrate that h^+ and $\bullet\text{O}_2^-$, in minor extent, are the main active species participating in the dye degradation ($h^+ > \bullet\text{O}_2^- \gg \bullet\text{OH}$). At this point, it is important to mention that the possible mechanism for the degradation process is very dependent not only on the antioxidant capacity of scavengers, but also on the nature of the radical chain reaction. The chain reactions involving $\bullet\text{O}_2^-$ radical participate in the degradation process, as occurred in the present case. The participation and generation of $\bullet\text{OH}$ along the degradation process does not rule, and the scavenging effect of BQ on $\bullet\text{O}_2^-$ radical tends to inhibit the formation of $\bullet\text{OH}$ radicals. In this case, the catalytic reaction rate will be greatly reduced by adding either BQ or TBA.

3.3. Possible photocatalytic mechanism

It is known that the mechanism for photocatalytic degradation is

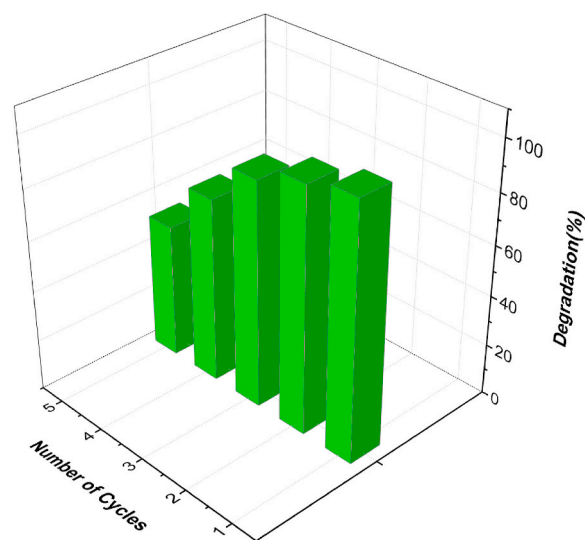


Fig. 10. The RhB photodegradation cycles using 50 mg of AWP 2 sample and 50 mL of RhB (10 mg L⁻¹) under visible light irradiation.

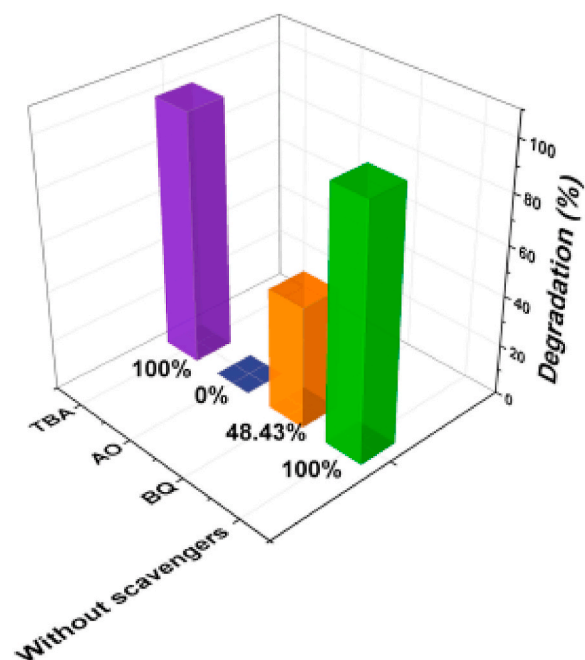


Fig. 11. Effect of scavengers on the degradation efficiency.

closely related to the position of the conduction (CB) and valence (VB) bands of a semiconductor. Thus, to fully understand the photocatalytic reaction mechanism occurring during the photodegradation of the as-

Table 1

Comparison of photodegradation of RhB using visible light by materials containing Ag₃PO₄ with the reported literature.

Catalyst	Mass of catalyst (mg)	RhB concentration (mg/L)	Rate constant (min ⁻¹)	Time degradation (min)	Reference
Ag ₃ PO ₄ /N-TiO ₂	20	10	0.019	120	[80]
g-C ₃ N ₄ @Ag@Ag ₃ PO ₄	6	20	0.030	60	[81]
Ag ₂ MoO ₄ /Ag ₃ PO ₄	50	10	0.359	12	[82]
Ag ₃ PO ₄ /NiO	30	5	0.224	30	[83]
Ag ₃ PO ₄ /BiNbO ₄	15	20	0.146	30	[84]
Ag ₃ PO ₄ @MgFe ₂ O ₄	20	10	0.1370	30	[85]
Ag ₃ PO ₄ :Mo	50	10	0.347	15	[50]
Ag ₃ PO ₄ :W	50	10	0.449	10	[86]
α -Ag ₂ WO ₄ /Ag ₃ PO ₄	50	10	0.504	10	Our work

prepared α -Ag₂WO₄/Ag₃PO₄ heterojunction, the energy band edge positions of the VB and the CB of both α -Ag₂WO₄ and Ag₃PO₄ were calculated. The energy band diagram for the heterojunction was constructed using Equations (2) and (3), which are based on the Mulliken's electronegativity, i.e., the geometric mean of the electronegativity of the constituent atoms in the material composition (Hill notation) [88,89]:

$$E_{VB} = \chi - E_g + 0.5E_g \quad (2)$$

$$E_{CB} = E_{VB} - E_g \quad (3)$$

where E_{VB} is the valence band potential, E_{CB} is the conduction band potential, E_e is the free electron energy on the hydrogen scale (~ 4.5 eV), E_g is the semiconductor band gap energy (Fig. S3), and χ is the absolute electronegativity (Mulliken's electronegativity) of the semiconductor. For Ag₃PO₄, the value of χ is 5.96 eV [83], and for α -Ag₂WO₄ it is 6.00 eV [90]. Therefore, by employing Equations (2) and (3) the E_{VB} and E_{CB} values of 2.66 eV and 0.26 eV for Ag₃PO₄ and 3.06 eV and -0.06 eV for α -Ag₂WO₄, respectively, were obtained and are in agreement with those reported in the literature [91,92].

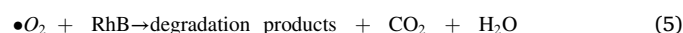
Fig. 12 illustrates two degradation mechanisms proposed for the α -Ag₂WO₄/Ag₃PO₄ heterojunction based on the obtained results. As shown in Fig. 12 (a), under visible light irradiation both Ag₃PO₄ and α -Ag₂WO₄ are excited, generating e^- - h^+ pairs. Due to the energy position of the CB and the VB of each material, a type I heterojunction is

formed, the photoexcited e^- in the CB of α -Ag₂WO₄ migrates to the CB of Ag₃PO₄, and the photogenerated h^+ in the VB of α -Ag₂WO₄ also migrates to the VB of Ag₃PO₄, thus generating an accumulation of charge carriers in Ag₃PO₄. In the VB of Ag₃PO₄, the h^+ that comes from α -Ag₂WO₄ and the photogenerated h^+ in the Ag₃PO₄ can directly degrade RhB [86,93], as shown in Equation (4):



The as-mentioned reaction is in accordance with the observed scavenger results displayed in Fig. 11, which indicate that h^+ is the main species acting in the RhB photodegradation.

As the CB of Ag₃PO₄ (0.26 eV) is more positive than the potential of O₂/ \bullet O₂⁻ ($E^0(\text{O}_2/\bullet\text{O}_2^-) = -0.046$ eV vs. NHE) [94], from a theoretical point of view it can be concluded that the photoexcited e^- on Ag₃PO₄ cannot react with dissolved O₂ in the reaction solution to yield $\bullet\text{O}_2^-$. However, this reaction between the accumulated e^- on the Ag₃PO₄ surface with the adsorbed O₂ to produce $\bullet\text{O}_2^-$ have been reported in several recent studies [83,95,96]. As seen in Fig. 11, the $\bullet\text{O}_2^-$ species act, to a lesser degree, in the photodegradation of RhB, according to Equation (5):



As seen in the scavenger experiments (Fig. 11), the $\bullet\text{OH}$ species do not participate in the degradation mechanism, and a possible explanation is that $\bullet\text{OH}$ would be rapidly self-consumed, forming H₂O and O₂

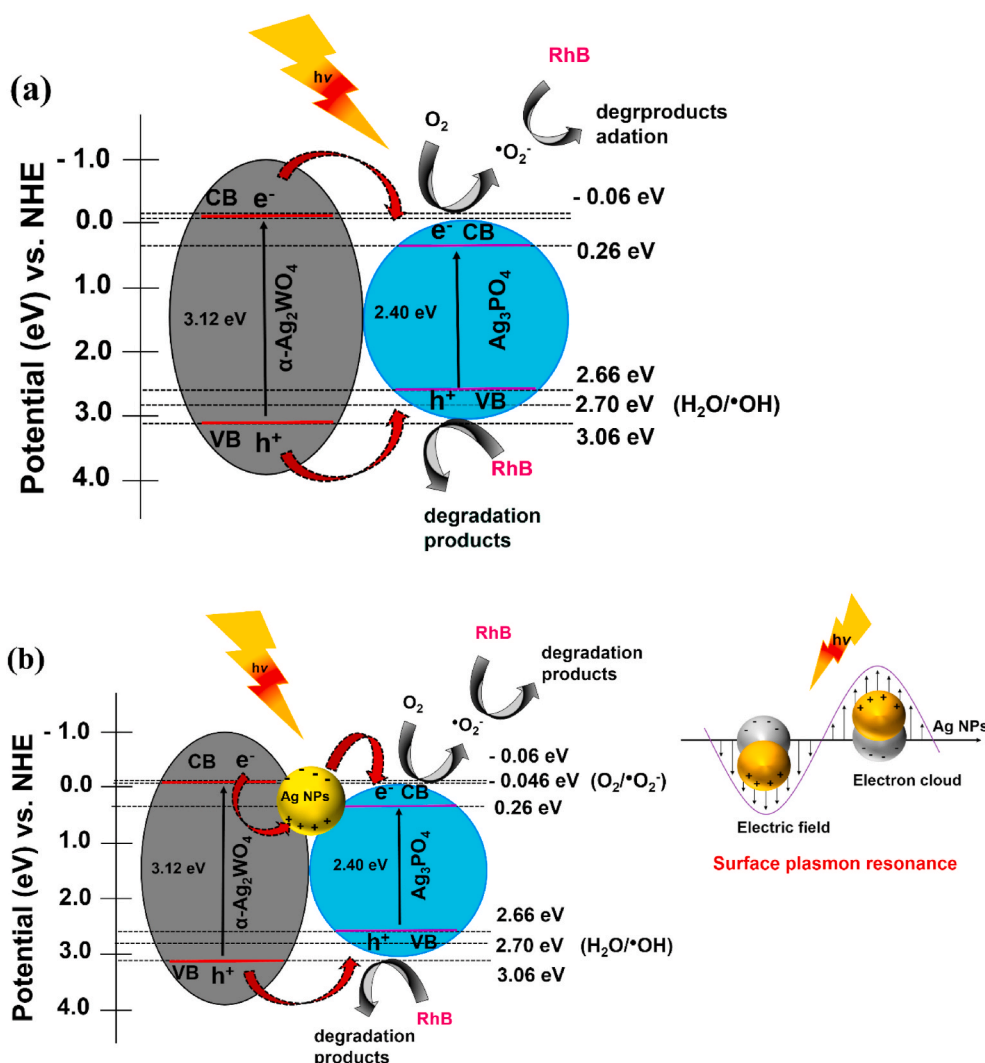


Fig. 12. Possible photocatalytic mechanism diagram.

[50], according to Equation (6):



As seen in XPS spectra results (Fig. 5), the Ag NPs formed may be present at the interface of the two semiconductors and could contribute to the separation process of e^- - h^+ pairs due to the surface plasmon resonance (SPR) effect, which creates a cross-linking bridge for both semiconductors [16]. The surface plasmon excitations are generated under visible light irradiation and partially converted into energetic electrons on the surface of Ag NPs. Fig. 12 (b) shows a proposed RhB degradation mechanism in the presence of Ag NPs, where it can be seen that the heterojunction remains type I and the same oxidizing species act. However, e^- photoexcited in α - Ag_2WO_4 can be transferred more quickly to Ag NPs. Thus, the e^- that are photoexcited to the CB of α - Ag_2WO_4 are able to migrate to the CB of Ag_3PO_4 more efficiently, as they use Ag NPs as a bridge. Therefore, the formation of metallic Ag NPs leads to an SPR effect, which is advantageous for the effective separation of charge carriers, resulting in improved photoactivity.

4. Conclusion

In conclusion, α - $\text{Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$ heterojunctions with different weight ratios of α - Ag_2WO_4 were fabricated by a facile chemical precipitation method. Under visible light irradiation, the as-prepared α - $\text{Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$ heterojunction displayed enhanced photocatalytic activity for the photocatalytic degradation of RhB. In particular, the sample with weight content of 24% of α - Ag_2WO_4 (named AWP 2) showed the highest photocatalytic performance for RhB photodegradation, degrading 94.3% in only 5 min of exposure to visible light, which is a very promising result when compared to pure Ag_3PO_4 and α - Ag_2WO_4 materials, which degraded 45 and 10%, respectively, under the same conditions. Our results demonstrate that this heterojunction can significantly enhance the absorption of visible light and successfully separate photogenerated electrons and holes after excitation. Such enhanced photocatalytic performance was explained by the surface plasmon resonance effect associated with the presence of metallic Ag at the metallic interface, as well as the formation of type I heterojunction, which served as a load transfer bridge, avoiding e^- - h^+ recombination, and consequently improving the photocatalytic activity of the heterojunction since charge carriers are available to react with the adsorbed species for a longer time.

CRedit authorship contribution statement

Aline B. Trench: Conceptualization, Methodology, Validation, Formal analysis, Writing – original draft. **Roman Alvarez:** Conceptualization, Methodology, Validation, Formal analysis, Writing – original draft. **Vinícius Teodoro:** Investigation, Writing – original draft. **Letícia G. da Trindade:** Investigation, Writing – original draft. **Thales R. Machado:** Investigation, Writing – original draft. **Mayara M. Teixeira:** Investigation, Writing – original draft. **Daniele de Souza:** Investigation, Writing – original draft. **Ivo M. Pinatti:** Investigation, Writing – original draft. **Alexandre Z. Simões:** Conceptualization, Writing – review & editing, Supervision, Resources, Funding acquisition. **Yara Galvão Gobato:** Conceptualization, Writing – review & editing, Supervision, Resources, Funding acquisition. **Juan Andrés:** Conceptualization, Writing – review & editing, Supervision, Resources, Funding acquisition. **Elson Longo:** Conceptualization, Writing – review & editing, Supervision, Resources, Funding acquisition.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgments

The authors acknowledge the financial support of the Brazilian research financing institutions: Fundação de Amparo à Pesquisa do Estado de São Paulo (FAPESP; Grant Nos. 2013/07296-2, 19/23488-5, 2019/03722-3, 2019/25944-8, and 2019/23488-5), Coordenação de Aperfeiçoamento de Pessoal de Nível Superior - Brasil (CAPES) - Finance Code 001 and Conselho Nacional de Desenvolvimento Científico e Tecnológico (CNPq, 142035/2017-3). J.A. acknowledges Universitat Jaume I (project UJI-B2019-30), and the Ministerio de Ciencia, Innovación y Universidades (Spain) (project PGC2018094417-B-I00) for financially supporting this research.

Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.matchemphys.2022.125710>.

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