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# **Ag<sub>2</sub>WO<sub>4</sub> as a multifunctional material: fundamentals and progress of an extraordinarily versatile semiconductor**

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## **Abstract**

Silver tungstate (Ag<sub>2</sub>WO<sub>4</sub>) based materials are intensively studied because of their unique chemical and physical properties. Ag<sub>2</sub>WO<sub>4</sub> polymorphs present discrete and intrinsically complex architectures due to a multifaceted framework derived from the different local coordination of Ag and W cations. This review starts with an overview of the synthesis of the three existing polymorphs ( $\alpha$ ,  $\beta$ , and  $\gamma$ ) and their structures. Then, the synthesis methods, surface modification, doping process, formation of solid solutions and heterojunctions, involving Ag<sub>2</sub>WO<sub>4</sub>, and the effect of electron beam/femtosecond laser irradiation have been exemplified and discussed in detail. Finally, existing challenges and future development are also provided to shed light on the application of highly efficient Ag<sub>2</sub>WO<sub>4</sub>-based materials

**Keywords:**  $\alpha$ -,  $\beta$ -,  $\gamma$ -Ag<sub>2</sub>WO<sub>4</sub> crystals; Semiconductor; Multifunctional material; morphology, Technological applications.

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## 1. Introduction

A major challenge for 21<sup>st</sup>-century scientists is to find novel strategies, concepts, and materials and to guarantee continued and steady technological progress in the following decades. Semiconductor materials, based on metal oxides, have become ubiquitous in scientific research due to their potential to overcome the challenges of availability, high-cost processing, and toxicity, which have been insurmountable to be applied in new technologies. These benefits justify that the US government and EU are investing US\$52 and US\$48 billion, respectively, in technological development [1, 2]. The family of metallic tungstates is a particularly appealing hunting ground, and a representative member is the silver tungstate ( $\text{Ag}_2\text{WO}_4$ ), which offers great structural variability and, consequently, broad technical applicability.

The multifunctional properties of  $\text{Ag}_2\text{WO}_4$ -based materials have been intensively studied, such as high adsorption capacity for cationic dye removal, antibacterial, photoluminescent, antifungal, antiviral, electrochemical sensing, violet-visible photocatalytic performance, lubrication, light-emitting diodes, gate dielectrics, ozone-ethanol-acetone-ammonia gas sensors, electro-catalyst, and so on [3-43]. Moreover, interesting biological properties have been demonstrated for  $\alpha$ - $\text{Ag}_2\text{WO}_4$ , and it has been used as an antimicrobial and anticancer agent [8, 19, 24, 44-47]. Therefore, it is very important to give a complete overview of this oxide material based on its various electronic properties and applications reported in the literature.

The review is subdivided as follows: we begin with the progress made in the different synthesis methods of  $\text{Ag}_2\text{WO}_4$  polymorphs, and then we continue studying their

structures and morphologies. Subsequently, the solid solutions, doping, and composites are outlined. Afterward, the effect and practical applications of  $\text{Ag}_2\text{WO}_4$ -based materials, induced by an electron beam (EB) and femtosecond (fs) laser irradiation, are summarized, and several applications that are (or can be) positively affected by the development and use of the materials formed by these interactions are discussed. We conclude by suggesting directions for future work to progress in the materials science field. We believe that this review of the above knowledge will help give us a proper perspective to assess the scientific merit regarding the reliability of  $\text{Ag}_2\text{WO}_4$  as a multifunctional semiconductor for new discoveries and practical applications in the future.

## 2. Structures for $\text{Ag}_2\text{WO}_4$ polymorphs

From a structural point of view,  $\text{Ag}_2\text{WO}_4$  polymorphs have a complicated network structure with remarkable crystallographic inhomogeneity and intrinsic anisotropy associated with their building blocks, i.e., constituent  $[\text{AgO}_x]$  and  $[\text{WO}_y]$  clusters. These clusters present distinct bonding environments of the statistically disordered  $\text{Ag}^+$  and  $\text{W}^{6+}$  cations coordinated with  $\text{O}^{2-}$  anions. The  $\text{W}^{6+}$  cation shows three  $[\text{WO}_y]$  ( $y = 4, 5, \text{ and } 6$ ) clusters, while the  $\text{Ag}^+$  cation can display different  $[\text{AgO}_x]$  clusters with low and high coordination numbers,  $[\text{AgO}_x]$  ( $x = 2, 4, 5, 6, \text{ and } 7$ ) (see **Fig. 1(a-c)**). The piling up of these clusters builds the three-dimensional structures.

$\text{Ag}_2\text{WO}_4$  crystals exhibit polymorphism, forming either the  $\alpha$ -,  $\beta$ - and  $\gamma$ - $\text{Ag}_2\text{WO}_4$  [48]: the thermodynamically stable  $\alpha$ -polymorph, with orthorhombic structure and space group ( $Pn2n$ ) and point-group symmetry ( $C_{2v}^{10}$ ); the metastable  $\beta$ -polymorph, with a hexagonal structure and space group ( $P6_3/m$ ) and point-group symmetry ( $C_{6h}^2$ ); and also metastable, the cubic  $\gamma$ -polymorph with space group ( $Fd\bar{3}m$ ) and point-group symmetry ( $O_h^7$ ) [14, 23, 26, 48-50].

<Insert Fig. 1(a-c)>

The  $\gamma$ - $\text{Ag}_2\text{WO}_4$  structure is a well-ordered crystal composed of two clusters,  $[\text{AgO}_6]$  and  $[\text{WO}_4]$ . When this symmetry is perturbed, the electron density between both types of clusters is modified, generating a change from the distorted octahedral  $[\text{AgO}_6]$  and tetrahedral  $[\text{WO}_4]$  clusters into distorted bipyramid  $[\text{AgO}_5]$  and trigonal bipyramid  $[\text{WO}_5]$  clusters, respectively, to render the hexagonal  $\beta$ - $\text{Ag}_2\text{WO}_4$  polymorph. These structural rearrangements involving the clusters reach the maximum crystalline adaptation when the hexagonal structure becomes orthorhombic, giving rise to the  $\alpha$ - $\text{Ag}_2\text{WO}_4$  structure. In this structure, three different tetrahedral  $[\text{WO}_4]$  clusters appear, while two different deltahedral  $[\text{AgO}_7]$  clusters, one distorted octahedral  $[\text{AgO}_6]$  cluster, two distorted tetrahedral  $[\text{AgO}_4]$  clusters, and one angular  $[\text{AgO}_2]$  cluster is present. Therefore, the local coordination of the metal cation in the bulk corresponds to the specific number of surrounding oxygen anions. However, these clusters present undercoordination at the surfaces, i.e., a minor number of oxygen anions around Ag, which is responsible for most of the semiconductor properties.

From an electronic point of view, these clusters, in turn, display low or high electron density, rendering different band gap values for each exposed surface. When

performing an experimental measurement of the band gap, an average value over all exposed surfaces at the morphology is obtained. On the other hand, the electron transfer process depends directly on the electron distribution at each surface and the electronic structure of the undercoordinated/distorted clusters at the surface. Therefore, the semiconductor can be considered a dynamic system in which the clusters are quantum clusters to carry out the absorption/emission of an electron in the ground and excited state, respectively. The different types of clusters on the same surface may have different properties and functions depending on the number and nature of defects. In the  $\text{Ag}_2\text{WO}_4$  semiconductor, the top of the valence band (VB) is composed mainly of hybridized Ag  $4d$  and O  $2p$  orbitals, which can lift the top position of the VB and narrow down the band gap. The bottom of the conduction band (CB) consisting of delocalized  $s$  and/or  $p$  orbitals displays significant dispersity so that it possesses a high migration efficiency of photogenerated electrons [51, 52]. The electronic properties and applications can be associated with the nature and rearrangement of the Ag–O and W–O chemical bonds, which range from ionic, covalent, to metallic, as well as unique electronic properties [53]. Recently, Moura et al. have presented a temperature-dependent Raman study on the structural changes induced by the temperature in the  $\beta\text{-Ag}_2\text{WO}_4$  crystal. In these transitions, the changes that occurred in the Raman spectra were characterized by the appearance and disappearance of new Raman modes. The results lead to a complex sequence of phase transitions involving  $\beta$ -hexagonal to  $\alpha$ -orthorhombic,  $\alpha$ -orthorhombic to hexagonal, and  $\beta$ -hexagonal to  $\alpha$ -orthorhombic again [54].

It is well known that many properties of solid materials, including semiconductors, are controlled not only by their geometry and electronic structure but also by the nature and defects in their crystal structure [55-58]. In this sense, it is essential to understand the influence of defects on the activity and performance for further optimization of the semiconductor system. In transition metal oxides, defects can be associated with anionic vacancies (oxygen vacancies) and cationic vacancies (metal vacancies or complexes vacancies). These vacancies act not only as electron-hole recombination centers by capturing a free electron and hole but also break the electronic structure of an otherwise perfectly periodic crystal. Our group is engaged in a research project devoted to understanding the nature of the defects in semiconductors. Positron annihilation lifetime spectroscopy measurements in conjunction with first-principles calculations, at the density functional theory level, X-ray diffraction (XRD), Raman spectroscopy, photoluminescence emissions, and field emission gun scanning electron microscopy techniques render that main defects are composed by Ag vacancies and that the concentration of this type of defect increases with the irradiation time [59].

### **3. Synthesis of the different polymorphs of $\text{Ag}_2\text{WO}_4$ -based materials**

One of the most sought-after goals in materials science is the acquisition of knowledge and the complete understanding of the different phases of a given material, which is essential to identifying which polymorph is best suited for a specific technological application. Changes in the concentration and preparation of the reactants, the nature of the

solvents, and/or external conditions (such as pressure and temperature) have been traditionally employed to control the synthesis of the different polymorphs. Therefore, this section provides a brief analysis and survey of the literature about  $\text{Ag}_2\text{WO}_4$  preparation.

Since it was first reported in the literature in 1949 by Rabes and Schenck [60], through classical experiments, metallic silver ( $\text{Ag}^0$ ) is oxidized to silver ion ( $\text{Ag}^+$ ) at high temperatures and reacts with tungsten oxide ( $\text{WO}_3$ ) to form  $\text{Ag}_2\text{WO}_4$  and another type of tungsten oxide ( $\text{W}_3\text{O}_8$ ) following the chemical reaction:  $2\text{Ag} + 4\text{WO}_3 \rightarrow \text{Ag}_2\text{WO}_4 + \text{W}_3\text{O}_8$ .

**Fig. 2(a-c)** displays the main steps employed in the initial and traditional synthesis method related to solid-state reaction or oxides powder mixtures to obtain  $\text{Ag}_2\text{WO}_4$ .

<Insert Fig. 2(a-c)>

As can be observed in **Fig. 2(a)**, the initial stage corresponds to traditional solid-state reactions of oxides powder mixtures, 1 mol of silver (I) oxide ( $\text{Ag}_2\text{O}$ ) and 1 mol of tungsten (VI) oxide ( $\text{WO}_3$ ), with black and canary yellow color, respectively. In the second stage, after the homogeneous mixing of the two oxides, both are placed in a rotating mill with balls of  $\text{ZrO}_2\text{:Y}$  with high hardness for better homogenization of this mixture; ethyl alcohol is employed to facilitate the interaction between the contact surface of the  $\text{Ag}_2\text{O}$  and  $\text{WO}_3$  powders for several grinding cycles for 4 to 6 hours and then placed in an oven for dry power [61], as shown in **Fig. 2(b)**. Finally, this more homogeneous and drier mixture was heated in a muffle furnace between 350 and 950 °C for 2-24 h to obtain  $\alpha$ - $\text{Ag}_2\text{WO}_4$  purple color powder by means of solid-state reaction:  $1\text{Ag}_2\text{O}$  and  $1\text{WO}_3 \rightarrow \alpha$ - $\text{Ag}_2\text{WO}_4$ , as displayed in **Fig. 2(c)**.



Later, some papers reported on thermochemistry and conductivity studies on this material. In 1973, Takahashi *et al.* [62] reported the study of a novel high ionic conductivity solid electrolyte, AgI-Ag<sub>2</sub>WO<sub>4</sub>, at ambient temperature. Two years later, Skarstad and Geller [49] determined data exclusively by XRD patterns, associated with the orthorhombic structure of the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> phase. Bottlebergs *et al.* [63] studied the phase diagram and ionic conductivity of the Na<sub>2</sub>WO<sub>4</sub>-Ag<sub>2</sub>WO<sub>4</sub> system. Additionally, several groups examined the thermal capacity, kinetics, and solid-state reaction mechanisms of Ag<sub>2</sub>WO<sub>4</sub> with bromide and mercury chloride [64-66]. In addition,  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> has been prepared as a “composite thin film” (polyvinyl chloride-C-Ag<sub>2</sub>WO<sub>4</sub>) and used as an electrode (cathode) for possible application in solid-state lithium batteries. Also,  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> was prepared as a composite coating to be used as a high-temperature solid lubricating material for wear applications [28, 67]. The synthesis of  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> was first reported by Mckechnie *et al.* in 1979 [68]. Schiraldi and Pezzati observed the application of Ag<sub>2</sub>WO<sub>4</sub> crystals as super-ionic conducting glasses in 1981 [69]. The following year, van den Berg and Juffermans [48] discovered the presence of polymorphism for Ag<sub>2</sub>WO<sub>4</sub> crystals: the metastable phases beta ( $\beta$ -Ag<sub>2</sub>WO<sub>4</sub>) related to the hexagonal structure and the gamma ( $\gamma$ -Ag<sub>2</sub>WO<sub>4</sub>) with a cubic structure.

The Ag<sub>2</sub>WO<sub>4</sub> crystals were prepared by different methods, such as solid-state reactions, oxide mixtures, mechanochemical [14, 70, 71], precipitation with calcination at elevated temperatures [72], and crystal growth utilizing a Czochralski process [73]. However, these preparation methods require high temperatures, long processing times, and sophisticated equipment with high maintenance costs. In recent years, improved synthesis

methods have been developed to overcome some problems of the older approaches to obtain crystals with different sizes, shapes, and single phases. The most frequently employed synthesis is controlled precipitation [74-76] *via* a simple ion-exchange method at room temperature [77], conventional hydrothermal (CH) [10, 78-80], and microwave-assisted hydrothermal (MAH) [81].

In particular, the CH method has been used recently in the preparation of different tungstates with different sizes and shapes [82-84]. Therefore, this method has received much attention from the scientific community due to its numerous advantages over other methods, such as the use of an environmentally friendly solvent (water) and a low processing temperature ( $\approx 200$  °C) [85]. By using the CH method, crystalline materials are obtained involving the dissolution of aqueous soluble reagents or precursors with or without the presence of a mineralizer (chemical bases) under temperature and pressure [86]. Generally, the materials obtained by this method are very fine, crystalline, and easily dispersed [87]. Recently, this method has been presented in the literature [10, 78] to prepare nanorods and nanofibers of  $\text{Ag}_2\text{WO}_4$ . However, obtaining  $\text{Ag}_2\text{WO}_4$  nanocrystals by this method requires long processing times (12-24 h) due to the low reaction kinetics [10].

The use of microwave energy in CH systems allows the development of a new synthesis method with enhanced processing capabilities for accelerating the reaction rate and promoting crystallization due to the increase in the number of effective shocks and the high heating rates [88]. In typical MAH processing, the “chemical reaction acceleration” factor is favored by the high frequency of electromagnetic radiation (2.45 GHz) that interacts with the permanent dipoles of the liquid phase (water) [89]. The movement of the

permanent or induced dipoles in the dispersed phase results in the rapid heating of the particles and consequently favors an increase in the reaction temperature. By this mechanism, coupling occurs between the reagents involved in the chemical reaction, which results in the formation of the products [90, 91]. Furthermore, processing in a MAH system may offer other advantages, such as different forms, highly pure phases, shorter processing time, and low energy consumption [92].

More recent research has performed methods of synthesis and stabilization of such metastable phases. Wang et al. [26] obtained superfine  $\beta$ - $\text{Ag}_2\text{WO}_4$  antibacterial powders by supersonic-assisted homogeneous precipitation from a chemical reaction between  $\text{AgNO}_3$  and sodium tungstate dihydrate ( $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ ). Another example is the fabrication of hierarchically porous metastable hollow nanospheres of  $\beta$ - $\text{Ag}_2\text{WO}_4$  by Wang et al. [93]. They employed the polymer poly(methyl methacrylate) (PMMA) as a template and inhibitor of the transformation to the  $\alpha$ - $\text{Ag}_2\text{WO}_4$ . PMMA added to the  $\text{AgNO}_3$  solution before the addition of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  solution acted as a template by arranging into cross-linked micelles with an inner hydrophobic backbone and outer hydrophilic carboxylic acid groups ( $-\text{COOH}$ ). A core-shell structure was formed composed of inorganic  $\text{Ag}_2\text{WO}_4$  nanoparticles (NP) in the shell and a PMAA at the core. The PMAA core could be easily removed by washing with water and alcohol, forming  $\text{Ag}_2\text{WO}_4$  hollow nanospheres. The obtained product was composed of hollow nanospheres (200-500 nm in size with a wall thickness of approximately 20 nm) consisting of an assembly of many small NPs with a diameter of less than 10 nm.

Sonochemical (SC) synthesis was used by Zinatloo-Ajabshir et al. [94] to obtain  $\alpha$ - $\text{Ag}_2\text{WO}_4$  nanostructures as visible-light-driven photocatalysts for wastewater treatment. The authors reported that the sample synthesized using sodium o-dodecylbenzenesulfonate as a surfactant improved the photocatalytic activity of the material. The  $\alpha$ - $\text{Ag}_2\text{WO}_4$  rod-like microcrystals were obtained by Nobre et al. [47] using sonochemistry and sonochemistry followed by CH methods in which the antimicrobial properties of the samples were investigated. Recently, the sonochemical method was employed to achieve the control size and morphology of the  $\alpha$ - $\text{Ag}_2\text{WO}_4$  materials by using three carboxylic acids (tartaric, benzoic, and citric). The degradation process of Rhodamine B (RhB) dye was excellent, reaching 100% degradation in 60 minutes for the as-synthesized samples with citric acid, which have nanoscale dimensions [95].

In 2014, our research group performed two different synthesis methods that led to the formation of two  $\text{Ag}_2\text{WO}_4$  phases. One was a coprecipitation (CP) reaction followed by a MAH method that resulted in micro-sized irregular polyhedral crystals of the  $\alpha$ - $\text{Ag}_2\text{WO}_4$  phase, and the other was a simple CP at 70 °C and pH 4, which resulted in nanowires and polyhedral crystals of the  $\beta$ - and  $\alpha$ - $\text{Ag}_2\text{WO}_4$ , respectively [9]. Another synthesis of nanorods was performed by the CP method followed by an SC treatment [13]. In 2014 and 2016, our research group employed a MAH treatment after a CP reaction in the presence of polyvinylpyrrolidone (PVP) surfactant, obtaining nanorods with gas sensor applications [6, 7]. Recently, Rakshitha et al. used the CP method to obtain the  $\text{Ag}_2\text{WO}_4$  with nanoflower-like morphology and enhanced photocatalytic activity [41].

Our research group synthesized multifaceted rod-like microcrystals of  $\beta$ - $\text{Ag}_2\text{WO}_4$  by a simple CP reaction at room temperature, which was further analyzed by field emission scanning electron microscopy (FE-SEM) and transmission electron microscopy (TEM) to observe the *in-situ* growth of Ag-NPs on the surface of the material [31]. The same synthesis method was further employed and combined experimental and theoretical studies were performed by Lemos et al. [96] and Roca et al. [31] to investigate the  $\beta$ - $\text{Ag}_2\text{WO}_4$  metastable polymorph.

Chen and Xu [5] synthesized  $\alpha$ - and  $\beta$ - $\text{Ag}_2\text{WO}_4$  phases by the same method (CP reaction with the dropwise addition of  $\text{AgNO}_3$ ) but at different concentrations. The first was the addition of 50 mL of 0.40 M  $\text{AgNO}_3$  to 50 mL of 0.20 M  $\text{Na}_2\text{WO}_4$ , which resulted in the  $\alpha$ - $\text{Ag}_2\text{WO}_4$ . The second method was similar, but in this case, 100 mL of 20 mM  $\text{AgNO}_3$  was added to 1 L of 1.0 mM  $\text{Na}_2\text{WO}_4$  related to the  $\beta$ - $\text{Ag}_2\text{WO}_4$ . Both phases have a rod-like morphology, but the  $\beta$ - $\text{Ag}_2\text{WO}_4$  crystals were larger (the average grain size of  $\alpha$ - $\text{Ag}_2\text{WO}_4$  was  $1.78 \times 0.56 \mu\text{m}$ , and that of  $\beta$ - $\text{Ag}_2\text{WO}_4$  was  $3.10 \times 0.79 \mu\text{m}$ ). Our group successfully used the dropwise precipitation method to synthesize the  $\beta$ - $\text{Ag}_2\text{WO}_4$  microcrystals using aqueous media at a low temperature (15 °C). The strategy of synthesis was: 4 mmol of  $\text{AgNO}_3$  in 50 mL of water was cooled to 15 °C and added dropwise at  $2 \text{ mL min}^{-1}$  to a solution of 2 mmol of  $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  dissolved in 50 mL of deionized water at 15 °C [96]. This practical and controllable synthesis method may inspire the synthesis of metastable phases of other complex metal oxides.

In 2015, Ramezani et al. [97] employed the electrochemical deposition method to prepare  $\beta$ - $\text{Ag}_2\text{WO}_4$  in different morphologies at the nanoscale by simply varying the

parameters of the conditions, such as the concentration of the  $\text{Na}_2\text{WO}_4$  electrolyte solution, voltage, stirring rate, and reaction temperature. As a result, they obtained nanosheets, nanorods, flower-like structures, and agglomerates of those morphology types. Other synthetic strategies, based on the use of  $\text{Ag}_2\text{WO}_4$  composites [98] and dopants [99] to stabilize the  $\beta$ - $\text{Ag}_2\text{WO}_4$  metastable have been reported.

Liu et al. [100] conducted a study by the CP synthesis at pH values varying from 1.5 to 6.8 and W concentrations to obtain  $\text{Ag}_2\text{WO}_4$  nanowires. For the  $\text{Ag}_2\text{WO}_4$  nanofibers, the diffraction peaks at  $27.5^\circ$  and  $41.6^\circ$  belong to the (121) and (132) facets of  $\beta$ - $\text{Ag}_2\text{WO}_4$ , while the other peaks are all attributable to  $\alpha$ - $\text{Ag}_2\text{WO}_4$  crystals, suggesting that the  $\text{Ag}_2\text{WO}_4$  sample is mainly the  $\alpha$ - $\text{Ag}_2\text{WO}_4$  phase and contains a small amount of the  $\beta$ - $\text{Ag}_2\text{WO}_4$ .

Our research group performed a comparative study among SC, CP, and CH methods. The results were similar, and eight-faceted rod-like microcrystals were obtained [11]. Our research group also attained these microcrystals in 2013 in a rapid stoichiometric CP reaction at  $90^\circ\text{C}$  without external agents and *via* MAH in 2014 and another study with PVP surfactant, but this time, the morphologies were accompanied by other shapes, such as cubic and triangular crystals [12,17,19]. In our recent article, we report the selective synthesis of each  $\alpha$ -,  $\beta$ -,  $\gamma$ - $\text{Ag}_2\text{WO}_4$  polymorphs through a simple precipitation route at ambient temperature by controlling the volumetric ratios of the  $\text{AgNO}_3/\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$  precursors in solution [101]. This selective synthesis allows us to control the formation of the desired polymorph. A schematic representation of the different synthetic procedures for  $\text{Ag}_2\text{WO}_4$  polymorphs is presented in **Fig. 3**.

<Insert Fig. 3>

#### 4. Morphology of $\text{Ag}_2\text{WO}_4$ polymorphs

The first article providing information about the microscopic aspects of  $\alpha\text{-Ag}_2\text{WO}_4$  was published by Yu et al. [78] in 2003, in which a facile and general low-temperature solution approach was employed to synthesize a family of single-crystalline tungstate nanorods and nanowires, such as  $\text{ZnWO}_4$ ,  $\text{FeWO}_4$ ,  $\text{MnWO}_4$ ,  $\text{Bi}_2\text{WO}_6$ ,  $\text{Ag}_2\text{WO}_4$ , and  $\text{Ag}_2\text{W}_2\text{O}_7$ . Their proposal was based on a hydrothermal crystallization method of an aqueous dispersion containing amorphous particles of each oxide, obtained by direct precipitation at room temperature.

Most studies related to  $\alpha\text{-Ag}_2\text{WO}_4$  crystals report the occurrence of a rod-like shape morphology with nano or micro dimensions. The HM is an appropriate route for the synthesis of fine, crystalline materials [87], while CP reactions in the absence of surfactants, capping agents, or special treatments tend to favor the synthesis of rods with larger dimensions. This was the case for the elongated rods obtained by Sreedev et al., with average diameters of  $0.30\ \mu\text{m}$  and lengths of up to  $5\ \mu\text{m}$ . They performed a simple CP reaction from equimolecular quantities of  $\text{Na}_2\text{WO}_4$  and  $\text{AgNO}_3$  at room temperature and  $\text{pH}=7$ , followed by calcination treatments between  $500$  and  $600\ ^\circ\text{C}$  for 3 hours, resulting in the dark to light violet precipitates [102-104]. Although the obtained rods were above the nanoscale, the authors considered the materials nanophase, due to the good aspect ratio.

Some studies observed the effect of pH on the formation of different nanostructures. For example, a deep interpretation is given by the work of Yu et al. [10, 78], in which the influence of pH on the morphology of the as-synthesized samples was analyzed, and  $\text{Ag}_2\text{WO}_4$  nanorods and  $\text{Ag}_2\text{W}_2\text{O}_7$  nanowires (approximately 50 nm of diameter and length up to micrometers) at pH=9 and 2 were obtained, respectively. The authors suggest that the growth process of nanoparticles occurs through a typical hydrothermal ripening process, with the direct mixing of two solutions leading to a larger number of amorphous particles. The larger particles grow at the cost of the smaller ones, with the adjustment of pH significantly affecting the composition of other species. Summarily, the authors cite references [105] and Gibbs-Thomson's theory to establish a relation between the chemical potential of crystals and the one-dimension growth.

Zhang et al. [15] obtained rods with sizes from 200 to 600 nm, with a diameter range of 60-100 nm, and some plate-like particles, verifying the key role of basic pH. They also argue that the formation of the rods occurs through the formation of tiny crystalline nuclei that takes place first, followed by the growth of larger particles and posterior aggregation of some rods, explaining why the plate-like particles are also observed [78, 106].

In 2005, George et al. [75] synthesized  $\alpha\text{-Ag}_2\text{WO}_4$  fibers of 0.5 $\mu\text{m}$  in a diameter and approximately 100  $\mu\text{m}$  in length, while the rods present diameter of about 30 nm. They used diluted reactant and precursor concentrations between 0.003-0.001 M for silver nitrate and 0.0015-0.0005 M for sodium tungstate and with a slightly basic pH, staying in suspension overnight. According to the authors, below this concentration, the fiber growth



is not visible, while above, the fiber growth is inhibited by the precipitation of  $\text{Ag}_2\text{WO}_4$ , which hinders long fiber growth.

Ng and Fan reported the synthesis of highly uniform one-dimensional (1D)  $\alpha$ - $\text{Ag}_2\text{WO}_4$  nanostructures with controllable aspect ratio, and they also investigated the growth mechanism of this material. According to the authors, the aspect ratio of the 1D materials is tuned by varying the duration of stirring during the initial synthesis period, and the results of the reaction intermediates bring to light a three-stage growth mechanism involving the directed aggregation of spherical building blocks [107].

Surfactants can be employed to modulate morphology and, consequently, the functionality of a material. Sodium dodecyl sulfate was used in MAH synthesis at different temperatures (100 °C – 160 °C) to tune the morphology of  $\alpha$ - $\text{Ag}_2\text{WO}_4$  and find a relationship between the photocatalytic/antibacterial activity and morphology. The results show that the lower MAH temperature (100 °C) generated an eight-facet hexagonal rod-like shape elongated along the [010] direction, while within higher microwave temperatures, new facets appear, resulting in crystals with 12 facets (120 °C – 140 °C), and crystals with 14 facets (160 °C). Photocatalytic activity measurements on the RhB dye and Rhodamine 6G (Rh6G) dye degradation show that this 14-facet crystal was the most efficient photocatalyst under UVC-light, which the authors relate to the presence of the higher values of surface energy ( $E_{\text{surf}} = 0.83 \text{ J m}^{-2}$ ) of the (110) plane. However, the antibacterial activity tests (*Escherichia coli* (*E. coli*) inactivation) indicated that the 8-facet crystal was the most efficient, which they attribute to a different mechanism involving direct interaction between specific (010) and (0 $\bar{1}$ 0) surfaces and cell surfaces [24]. This

surfactant was used to obtain, for the first time, a morphology very close to a previously calculated cubic-like morphology [108, 109]. The as-synthesized crystals have a cuboid shape formed by the junction of two or more cubes in one direction, resulting in elongated cubes with average sizes of 200 nm [109]. Although the importance of the study for the relation with previous theoretical works, this morphology showed to be more inactive as a photocatalyst when compared to the most obtained eight-facet hexagonal micro-rods. However, in 2019, a sequential study was published showing that this cuboid morphology, when irradiated by an fs laser, provokes the formation of Ag/ $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> composite 64 times more efficient against *Methicillin-susceptible Staphylococcus aureus* – MRSA, in comparison to the non-irradiated morphology, and 2 times higher than the rod-like irradiated morphology [110]. Very recently, the toxicity of this morphology and also rod  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> samples were investigated to freshwater microalga *Raphidocelis subcapitata* at the cellular and population levels [111,112]. Both caused population growth inhibition, being the rod morphology 1.7 times more toxic than the cubic.

By using different solvents (water, alcoholic solution, and ammoniacal solution) [8] two morphologies could be observed, micro-rods in the alcoholic solution and flowerlike structures in water and the ammoniacal solution. A considerable number of three-dimensional (3D) flowerlike  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> microcrystals were observed in the aqueous solution. The observed structures were essentially composed of aggregated nanorod particles exhibiting partial orientation and attachment on the irregular facets. The same 3D flowerlike  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> structures are obtained using the ammoniacal solution. However, in this case, the crystals exhibit three surfaces, with the (001), (101), and (010)

crystallographic planes, which are more defined than the crystals obtained using water as solvent. With the synthesis using the alcoholic solution,  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> nanorods with three surfaces were observed, and the same crystallographic planes as those observed in the case of the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> structures synthesized using the ammoniacal solution.

Dimethyl sulfoxide was employed as a solvent during the synthesis by using the CP method, and the injection of the reactants was carefully controlled. As a result, rectangular rods with the exposed (010), (001), and (100) facets were synthesized with sizes from 0.27 to 0.67  $\mu\text{m}$  in height and 0.07 to 0.15  $\mu\text{m}$  in width [113]. In this work, the effect of an SEM and TEM electron beam interaction was also investigated. It was observed, for the first time, besides the Ag<sup>0</sup>, the presence of silver oxides (Ag<sub>2</sub>O and Ag<sub>3</sub>O<sub>4</sub>) on the extruded material.

In 2020, the potential of ethylenediamine complexing agent to modulate the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> morphology and structural properties was demonstrated, obtaining microflowers with favorable optical properties [114]. Additionally, the morphological evolution and antibacterial (against *Staphylococcus aureus*) and photocatalytic activity for degradation of RhB dye with  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> as catalyst obtained by the CP method followed by microwave irradiation for different times were also analyzed by our research group in 2020 [115]. We found that the local coordination of superficial Ag and W atoms (i.e., incomplete [AgO<sub>y</sub>·zV<sup>x</sup>O] and distorted [WO<sub>6</sub>]<sub>d</sub> and [AgO<sub>4</sub>]<sub>d</sub> clusters) on each exposed surface of the corresponding morphology dictated the antibacterial and photocatalytic activities of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub>.

Although  $\alpha$ - $\text{Ag}_2\text{WO}_4$  crystals are the most investigated and stable phase of  $\text{Ag}_2\text{WO}_4$  material, it might exist in more than one structure, as previously mentioned [48, 49, 68]. Subtle differences in pH and the heating of the solution appear to determine which of the polymorphs is obtained, but accurate powder data were not provided for the different forms, and polymorphic behavior was not investigated in detail. The inferior number of studies referring to the metastable phases is due to the difficulties associated with the isolation of phases that are non-thermodynamically favored, such as the  $\beta$ - or  $\gamma$ -phases of  $\text{Ag}_2\text{WO}_4$ , which are also unsuitable for practical applications, given the possibility of the transition to the  $\alpha$  phase over time.

The hexagonal  $\text{Ag}_2\text{WO}_4$  phase with multilevel sphere clusters of 2.5–2.8  $\mu\text{m}$  in diameter composed of several second-level spheres (approximately 0.8  $\mu\text{m}$  in diameter) was synthesized for the first time by Zhang et al. [33] with the help of an eggshell membrane and acrylamide as templates. The authors showed the importance of the acrylamide template using the same synthesis method, with and without the polymer. Thus, in the presence of the polymer, symmetric and concentric sphere clusters were built. In the absence of the polymer, but in the presence of an eggshell membrane, sphere-like structures, but not clusters, were the products. However, only dendritic crystals were formed in the absence of both membrane and polymer. The monomers acted, not only as an auxiliary template but also as an absorber of photogenerated electrons, forming PAM through a polymerization reaction.

The  $\gamma$ - $\text{Ag}_2\text{WO}_4$  crystals with cubic phase were obtained by our research group in 2017 *via* a CP reaction at low temperatures (0-5  $^\circ\text{C}$ ) [116]. In this study, first-principles

calculations and Wulff construction were performed to display the available morphologies of this polymorph. The experimental morphologies observed by scanning electron microscopy (SEM) were nearly spherical, with small nanocrystals with a cuboid structure on the large microcrystal surfaces, for the 10 min reaction, while the samples corresponding to a 20 min reaction time had the most microparticles with irregular or approximately spherical structures, although some faceted particles were also found. These faceted NPs were observed to be rod-shaped [116]. In addition, this metastable  $\gamma$ - $\text{Ag}_2\text{WO}_4$  phase has recently been successfully synthesized and stabilized by the CP method assisted by PVP at room temperature [117]. Particle formation with octahedral morphology and preferential growth in the (111) plane is favored in the presence of this surfactant.

The promissory antimicrobial properties of  $\text{Ag}_2\text{WO}_4$  microcrystals were studied by Bastos et al. against several pathogenic bacteria stains (gram-positive and gram-negative bacteria). The results showed a minimum inhibitory concentration less than the positive control [80]. Due to the current COVID-19 pandemic, our research group has developed the next generation of innovative materials with enhanced anti-SARS-CoV-2 activity to prevent the spread of this virus within the community. In 2022, we reported the synthesis of chitosan/ $\alpha$ - $\text{Ag}_2\text{WO}_4$  composites generated by fs laser irradiation. This material is very efficient in eliminating bacteria (*E. coli*, *MRSA*, and the yeast strain *Candida albicans* – *C. albicans*) and SARS-CoV-2 by contact [118].

A fundamental framework, based on the combination of first-principles calculations and advanced electron microscopy techniques, was developed to obtain the available morphologies of different semiconductors [108, 119-126]. The evolution of the

morphology has been interpreted through equilibrium considerations by using the Wulff method [127], in which the increases/decreases of the surface energy values are the driving force to accompany the appearance of a given exposed surface at the morphology. We showed that the predicted theoretical morphology could match the experimental ones, thereby rationalizing the reaction pathway along the synthesis process. This strategy opens the door for the rational design of semiconductors to enhance the performance of a desired technological application. The high degree of coincidence between theory and experiments suggests that this model has a more general scope of application in different metal oxides, as reported by different research groups [128-138].

**Fig. 4** illustrates some morphologies that the  $\alpha$ -,  $\beta$ -,  $\gamma$ - $\text{Ag}_2\text{WO}_4$  polymorphs can exhibit, and the complete map of morphologies can be found in the literature [31, 108, 116].

<Insert Fig. 4>

Based on the detailed analysis of the atomic coordination environment of surface atoms, we can find a relationship between the material properties (photocatalytic and biocide activities) and the exposed surfaces at the morphology. This is an excellent strategy not only to rationalize and explain the behaviors and properties of the materials but also to explain the corresponding action mechanism [139].

Several strategies can be utilized to improve the activity of  $\text{Ag}_2\text{WO}_4$  polymorphs, such as solid solutions and doping, composites, and EB and fs irradiation (**Fig. 5**). The next sections will address these issues.

<Insert Fig. 5>

## 5. Solid solutions and doping

The aim of the research is to improve some properties of materials or even create new applications for them. This proposal can be achieved through the modification of the synthetic route, such as using different temperatures, a change of solvent, and adjusting the methodology for obtaining the material. Another way to achieve this goal can be through the addition of different atoms from the precursors to the reaction medium to generate a solid solution or a doping process.

Doping and solid solutions are important and well-demonstrated strategies for maintaining the integrity of the material's crystal structure. At the same time, physicochemical properties are effectively tuned by changing the crystal parameters, morphology, and electronic structure, among others. Therefore, the rich chemistry of  $\text{Ag}_2\text{WO}_4$  is often modified, intentionally or unintentionally, through the inclusion of defects and dopants. Kumar et al. [99] have proven that the doping process involving  $\text{Eu}^{3+}$  cations is an interesting strategy for the stabilization of the hexagonal metastable  $\beta\text{-Ag}_2\text{WO}_4$  using the CP method at room temperature. In that report, no changes were observed in the  $\alpha\text{-Ag}_2\text{WO}_4$  crystal structure until 1.0 mol%  $\text{Eu}^{3+}$  doping, and for 2.0 mol%, the  $\beta\text{-Ag}_2\text{WO}_4$  crystals were stabilized, inhibiting the hexagonal-to-orthorhombic phase transition. Optical characterization, time-resolved photoluminescence, and temperature-dependent studies

supported the origin of the spectral findings. For both  $\alpha$ - and  $\beta$ - $\text{Ag}_2\text{WO}_4$  phases, the peak emission has been attributed to the charge transfer transition within tungsten octahedra and tetrahedra in  $\alpha$ - and  $\beta$ - $\text{Ag}_2\text{WO}_4$ , respectively.

Our early studies showed that the  $\text{Eu}^{3+}$  cations are doping  $\text{Ag}_{2-3x-y}\text{Eu}_x\text{Li}_y\text{WO}_4$  (where  $x = 0.01$ ,  $y = 0.01, 0.02$ , and  $0.03$  mol) [37] and  $\text{Ag}_{2-3x}\text{Eu}_x\text{WO}_4$  (with  $x = 0.01, 0.02, 0.04$ , and  $0.08$  mol) [140], and they were efficiently synthesized by the CP method at  $90^\circ\text{C}$  for 30 min. For all samples, the XRD patterns can be indexed to the pure orthorhombic  $\alpha$ - $\text{Ag}_2\text{WO}_4$  phase, and the FE-SEM analysis revealed nanorod-like microcrystals. The structural refinement data and the emission spectra analysis (lifetime decays) suggest that  $\text{Eu}^{3+}$  cations occupy at least two crystallographic sites (probably Ag sites on clusters coordinated by seven O atoms,  $[\text{AgO}_7]$ ). In the latest report,  $\text{Li}^+$  cation addition facilitated  $\text{Eu}^{3+}$  cation incorporation and the emission of red light much more efficiently. The *Commission Internationale de l'Eclairage* coordinates [141] confirmed the classification of this material as a potential phosphor in the visible range.

Similar studies have been extended for other rare earth dopant ions. These works describe the structural and luminescence properties of  $\alpha$ - $\text{Ag}_{1.97}\text{RE}_{0.01}\text{WO}_4$  for  $\text{RE} = \text{Pr}, \text{Sm}, \text{Eu}, \text{Tb}, \text{Dy}$ , and  $\text{Tm}$  [142] and for  $\text{RE} = \text{Eu}, \text{Dy}, \text{Tm}$ , and  $\text{Er}$  [143] synthesized using a simple CP method. The X-ray photoelectron spectroscopy properties of these phosphors were reported for the first time [142] and efficient energy transfer from the  $\text{WO}_4^{2-}$  group to  $\text{RE}^{3+}$  cations was observed. Additionally, these systems showed the possibility of color tuning by simple RE dopant selection. These pieces of evidence also contribute to the development of new materials for temperature sensing (by the emission intensity



dependence on temperature) and their use for optical applications, including light-emitting diodes and lasers. The fluorescence property of  $\text{Eu}^{3+}$  ions in the  $\alpha$ - and  $\beta$ - $\text{Ag}_2\text{WO}_4$  was investigated by Huangfu et al. The results have shown that the metastable  $\beta$ - $\text{Ag}_2\text{WO}_4$  structure is more favorable as a host material for  $\text{Eu}^{3+}$  than  $\alpha$ - $\text{Ag}_2\text{WO}_4$ , and the  $\beta$ - $\text{Ag}_2\text{WO}_4:\text{Eu}$  presented about 60 times superior fluorescence intensity [144].

The effects of solid solution formation due to the incorporation of transition-metal (TM) cations have been investigated from a combination of experimental and theoretical studies [37, 143, 145-149]. From  $\alpha$ - $\text{Ag}_{2-2x}\text{TM}_x\text{WO}_4$ , cations such as Ni ( $0 \leq x \leq 0.08$ ) [119], Zn ( $0 \leq x \leq 0.25$ ) [148] and Cu ( $0 \leq x \leq 0.12$ ) [145, 146] were used for substitution of Ag atoms. In the recently published article, Pereira et al. showed that the increase in the  $\text{Cu}^{2+}$  cations content in the  $\alpha$ - $\text{Ag}_{2-2x}\text{Cu}_x\text{WO}_4$  solid solutions provoked a change in the morphology to a nanoplate-like morphology formed only of (010) and (101) surfaces enhancing the antibacterial and antifungal activities against MRSA and *C. albicans*, respectively [150]. In another publication, Marzieh et al. investigated the behavior of  $\text{Ag}_{1.8}\text{Zn}_{0.1}\text{WO}_4$  nanoparticles in the production of singlet oxygen,  $^1\text{O}_2$ , and hydroxyl free radical,  $\bullet\text{OH}$ , to improve photodynamic therapy, and the results have shown that this material can be used as an effective drug for the treatment of cancer due its properties such as photoluminescence, cheapness, and stability [151].

Additionally, a W chemical substitution in  $\alpha$ - $\text{Ag}_2\text{W}_{0.75}\text{Mo}_{0.25}\text{O}_4$  has been described [147]. These solid solutions were prepared *via* the MAH [145, 147, 149] and CP [148] methods. Recently, the  $\alpha$ - $\text{Ag}_2\text{W}_{0.75}\text{Mo}_{0.25}\text{O}_4$  solid solution has been synthesized in two steps: initially, the precipitated sample corresponds to the  $\alpha$ - $\text{Ag}_2\text{WO}_4/\beta$ - $\text{Ag}_2\text{MoO}_4$

heterostructure, and then, the posterior microwave irradiation favors the substitution of  $W^{+5}$  by  $Mo^{5+}$  cations and subsequent formation of the solid solution [147]. The results showed changes in the morphology and optical properties, and by tuning the TM concentration, simultaneous control of the optical and electronic properties can be achieved. This behavior enhances the photodegradation of methylene blue (MB) dye and tetracycline (TC) antibiotic [145] and RhB dye [147]. When the MAH [145, 147] method is used, in the samples obtained at the longest microwave irradiation time, the number of structural defects was increased. As an example, **Fig. 6** illustrates the  $\alpha$ - $Ag_2WO_4$  structure with doping atoms at the W and Ag sites.

<Insert Fig. 6>

## 6. $Ag_2WO_4$ composites

### 6.1. C-based compounds

Coupling C-based materials, graphitic carbon nitride (g- $C_3N_4$ ), reduced graphene oxide (rGO), or transition metal carbides/nitrides (MXenes), with other semiconductor materials, have received considerable attention due to their excellent chemical stability and unique electronic structures. Particularly, these C-based materials with two-dimensional thin sheet structures retain this shape with wrinkled regions after hybridizing with  $Ag_2WO_4$ . This morphology might provide enough contact surface area between them and the  $Ag_2WO_4$  particles and can facilitate charge-carrier transport. The  $Ag_2WO_4/g-C_3N_4$

composites can typically be prepared by a simple deposition-precipitation method (or *in-situ*  $\text{Ag}_2\text{WO}_4$  growth on the  $\text{g-C}_3\text{N}_4$  surface) at room temperature [152-156]. The most interesting result of this procedure is that the orthorhombic  $\alpha\text{-Ag}_2\text{WO}_4$ , the thermodynamically most stable phase, is not observed, and diffraction peaks indexed according to the hexagonal  $\beta\text{-Ag}_2\text{WO}_4$  phase, thermodynamically less stable, are obtained. The addition of  $\text{g-C}_3\text{N}_4$  nanosheets plays an important role in controlling the phase structure of  $\text{Ag}_2\text{WO}_4$ . After adsorbing the precursor ions, the  $\text{g-C}_3\text{N}_4$  nanosheets can provide active sites for the nucleation and growth of  $\beta\text{-Ag}_2\text{WO}_4$  NPs, and the phase transformation from  $\beta\text{-Ag}_2\text{WO}_4$  to stable  $\alpha\text{-Ag}_2\text{WO}_4$  can effectively be inhibited because precursor ions are tightly tiled onto the  $\text{g-C}_3\text{N}_4$  nanosheets [154, 155]. These results are in correspondence with previous reports where  $\beta\text{-Ag}_2\text{WO}_4$  crystals, synthesized using a simple precipitation method, are generally obtained using templates, such as polymethacrylic acid [93], polyvinyl pyrrolidone [98], and acrylamide [33]. When  $\text{Ag}_2\text{WO}_4$  supported on  $\text{g-C}_3\text{N}_4$  is synthesized by the same precipitation route, but additional SC impregnation is used, the  $\beta\text{-Ag}_2\text{WO}_4$  is obtained, but extra diffraction peaks appear, which are assigned to  $\alpha\text{-Ag}_2\text{WO}_4$  [157]. Similarly, when at the final step of  $\text{Ag}_2\text{WO}_4/\text{g-C}_3\text{N}_4$  synthesis, the precursor mixture is transferred into a Teflon-lined autoclave and subsequently hydrothermally heated (at 180 °C for 12 h) [158] or warmed at 300 °C for 2 h [159], pure  $\alpha\text{-Ag}_2\text{WO}_4$ , as the only  $\text{Ag}_2\text{WO}_4$  phase in the composite, is obtained. These synthetic routes, assisted by additional energy, sonochemical or thermal, provide adequate conditions for partial or total phase transformation to the thermodynamically more stable  $\alpha\text{-Ag}_2\text{WO}_4$ . Furthermore, this procedure can achieve the growth of Ag NPs on the  $\text{Ag}_2\text{WO}_4$  surface and the ternary composite  $\text{Ag}/\text{Ag}_2\text{WO}_4/\text{g-C}_3\text{N}_4$  [159]. In addition, quaternary

$\text{Ag}_2\text{WO}_4$ -based g- $\text{C}_3\text{N}_4$ -containing composites have been reported recently [160], and the corresponding results proved that multicomponent heterostructures are a promising strategy to promote the photocatalytic performance of  $\text{Ag}_2\text{WO}_4$ .

A ternary Fe-doped g- $\text{C}_3\text{N}_4/\text{Ag}_2\text{WO}_4$  visible light active photocatalyst with exceptional RhB degradation was fabricated by Alosaimi et al. through an ultrasonic-assisted facile hydrothermal method [161]. By using the solvothermal method, Vinoth et al. have obtained a novel  $\text{Ag}_2\text{WO}_4$  nanorod encapsulated sulfur-doped graphitic carbon nitride ( $\text{Ag}_2\text{WO}_4@\text{SGCN}$ ). According to the authors, this material was successfully applied for the electrochemical detection of nitrofurantoin, revealing its practical applicability for sensing [162].

Recently, graphene oxide (GO) and  $\text{Ti}_3\text{C}_2$ , with a two-dimensional sheet structure, were successfully used on  $\text{GO}/\text{Ag}_2\text{WO}_4$  [163],  $\text{GO}/\text{Ag}_2\text{O}/\text{Ag}_2\text{WO}_4$  [164], and  $\text{Ag}_2\text{WO}_4/\text{Ti}_3\text{C}_2$  [165] composites. The diffraction peaks of the  $\text{Ag}_2\text{WO}_4$  pattern can be assigned to crystal planes of the  $\alpha$ -phase of  $\text{Ag}_2\text{WO}_4$  crystals. Again, the hydrothermally assisted route provides favorable conditions for producing the most stable  $\text{Ag}_2\text{WO}_4$  phase. On the other hand, as expected by the CP method, the combination of two-dimensional (2D)  $\text{Ti}_3\text{C}_2$  with  $\text{Ag}_2\text{WO}_4$  produced an  $\text{Ag}_2\text{WO}_4$  sample in good agreement with the hexagonal phase  $\beta$ - $\text{Ag}_2\text{WO}_4$  [165]. In addition, an  $\text{Ag}_2\text{WO}_4/\text{GO}$ -based nanocomposite has recently been SC synthesized with a suitable electrochemical performance as a potential material for supercapacitor electrodes [166]. By analysis of the images from TEM for the C-based compounds with  $\text{Ag}_2\text{WO}_4$  composites, several morphologies, such as irregular sphere-like [152, 154, 165], nanorod [157, 158], spindle-like [153], and rod-like [163, 167]

morphologies, can be observed. All these composites have been successfully applied to several photocatalytic degradation processes of organics (dyes, antibiotics, and so on) for environmental remediation. In addition, other interesting abilities such as antibacterial disinfection [153, 155, 168] have also been reported.

## 6.2. Chalcogenides

Some chalcogenide compounds, such as  $\text{Ag}_2\text{S}$ ,  $\text{WS}_2$ ,  $\text{MoS}_2$ ,  $\text{VS}_4$ ,  $\text{ZnS}$ ,  $\text{MnS}$ , and  $\text{Sb}_2\text{S}_3$ , have also been used in composites with  $\text{Ag}_2\text{WO}_4$ . These compounds behave as semiconductors with a band gap between 1-2 eV, exhibiting excellent performance in charge separation, accelerating electron transfer, and high redox ability.

A series of  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$  composites were synthesized with different  $\text{Ag}_2\text{S}$  compositions via a facile successive precipitation route [169]. For the  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$  composites, the main characteristic diffraction peaks of  $\text{Ag}_2\text{WO}_4$  can be indexed according to pure  $\alpha\text{-Ag}_2\text{WO}_4$ . From a morphological point of view,  $\alpha\text{-Ag}_2\text{WO}_4$  appears with a regular microrod, and on their surface, regular NP- $\text{Ag}_2\text{S}$  are deposited. On the other hand, for a facile SC method,  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$  (denoted as U- $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$ ) was also obtained [170] and compared with  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$  composites fabricated by the precipitation method. A similar orthorhombic  $\alpha\text{-Ag}_2\text{WO}_4$  phase with more evident changes in morphology can be observed. Ultrasonic irradiation yields an evident improvement in surface area in connection with the decrease in the average crystallite size of the samples, which could indicate that ultrasonic irradiation improved the dispersion and decreased the crystallite sizes. The attached NP- $\text{Ag}_2\text{S}$  are fused into fluffy particles. In addition, a rod-like (typical

hexagonal submicron-rod)  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystal with Sb<sub>2</sub>S<sub>3</sub> chalcogenide nanorods on its surface was prepared by a simple wet impregnation method [171]. This heterostructure exhibits prolonged visible light absorption when compared to pure  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub>.

2D chalcogenides, such as MoS<sub>2</sub> and WS<sub>2</sub>, have also been used to join Ag<sub>2</sub>WO<sub>4</sub> in composites. A MoS<sub>2</sub>/Ag<sub>2</sub>WO<sub>4</sub> nanohybrid was prepared *via* the hydrothermal method on a Teflon-lined autoclave heated at 180 °C for 24 h [172]. Instead, the Ag<sub>2</sub>WO<sub>4</sub>/WS<sub>2</sub> composites [167] were fabricated by using a deposition-precipitation method in a similar procedure to composites of Ag<sub>2</sub>WO<sub>4</sub> with 2D g-C<sub>3</sub>N<sub>4</sub> [152-154]. The XRD patterns of MoS<sub>2</sub>/Ag<sub>2</sub>WO<sub>4</sub> and Ag<sub>2</sub>WO<sub>4</sub>/WS<sub>2</sub> were consistent with all the diffraction peaks of the orthorhombic  $\alpha$ -phase and hexagonal  $\beta$ -phase of Ag<sub>2</sub>WO<sub>4</sub>, respectively, in agreement with the observed report for 2D g-C<sub>3</sub>N<sub>4</sub>. By a simple *in-situ* deposition-precipitation method, the sheet morphology can provide active sites for the nucleation and growth of metastable  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> in which precursor ions are tightly tiled onto the nanosheets, and the phase transformation from  $\beta$ - to stable  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> can effectively be inhibited. On the other hand, the synthesis routes assisted by additional energy, SC and hydrothermal, are thermodynamically favorable for the synthesis of the most stable  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> polymorph. Recently, Sudheer et al. developed VS<sub>4</sub>/Ag<sub>2</sub>WO<sub>4</sub> [173], ZnS/Ag<sub>2</sub>WO<sub>4</sub> [174], CoS/Ag<sub>2</sub>WO<sub>4</sub> [175], and MnS/Ag<sub>2</sub>WO<sub>4</sub> [176] composites synthesized through the CP method, and the orthorhombic phase of Ag<sub>2</sub>WO<sub>4</sub> was obtained. These composites presented highly efficient photocatalysis activity for the removal of pollutants (MB degradation) from water and as excellent antimicrobial agents (for *E. coli* and *Bacillus subtilis* – *B. subtilis*).

The overall morphology for these composites of the  $\text{Ag}_2\text{WO}_4$  crystals with 2D materials prepared by *in-situ* deposition-precipitation clearly shows that the  $\text{Ag}_2\text{WO}_4$  particles were grown over the 2D nanosheet. Initially, the Ag precursor is introduced into the 2D-material suspension, and  $\text{Ag}^+$  cations can be tightly adsorbed onto the surface of the nanosheets through electrostatic interactions. Upon the addition of the W precursor,  $\text{Ag}^+$  cations react with  $\text{WO}_4^{2-}$  complex anions and  $\text{Ag}_2\text{WO}_4$  nuclei are generated. Finally, the  $\text{Ag}_2\text{WO}_4$  particles grow directly on the sheet surfaces. This growth strategy is schematically illustrated in **Fig. 7**.

<Insert Fig. 7>

Regarding the optical behavior of  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$ ,  $\text{U-Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$ , and  $\text{Ag}_2\text{WO}_4/\text{WS}_2$  composites, the absorption edge of the ultraviolet-visible (UV-Vis) spectra shows a slight redshift and an increase in the visible light absorption when compared to pure  $\text{Ag}_2\text{WO}_4$  crystals. Additionally, from photoluminescence (PL) measurements, the peak intensity in the composites is much weaker than that of pure  $\text{Ag}_2\text{WO}_4$  crystals, indicating the importance of the efficiency of charge carrier separation for the composites.

### 6.3. Halogens

The  $\text{AgX}$  ( $X = \text{Cl}, \text{Br}, \text{I}$ ) family is one of the most important Ag-based efficient visible-light-driven photocatalysts, attracting much attention. It is well known that these compounds are easily decomposed to generate, after light irradiation, hybrids with metallic Ag, which have been used as highly efficient plasmonic photocatalysts owing to the surface

plasmon resonance (SPR) effect [177-180]. Consequently, the production of AgX/Ag<sub>2</sub>WO<sub>4</sub> [27, 40, 61, 98, 181-186] binary composites is an effective approach to photocatalyst design for practical applications in solving existing environmental problems. Initially, by a facile precipitation process, Ag<sub>2</sub>WO<sub>4</sub> particles can be obtained with a high yield at room temperature. Subsequently, AgX crystals are grown on the surface of Ag<sub>2</sub>WO<sub>4</sub> to form Ag<sub>2</sub>WO<sub>4</sub>/AgX composites by a facile *in-situ* anion exchange reaction with different X<sup>-</sup> ion solutions at room temperature. The results revealed that the Ag<sub>2</sub>WO<sub>4</sub> rod-like structure acted as the template of 1D AgX/Ag<sub>2</sub>WO<sub>4</sub> heterocrystals. These 1D semiconductor structures demonstrated substantial catalytic efficiency due to the high aspect ratio, which enhanced light absorption, rapid electronic transport along the long axis, and efficient utilization of electron-hole pairs.

In the Ag<sub>2</sub>WO<sub>4</sub>/AgX hybrid nanorods, all the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals with well-indexed diffraction peaks according to the orthorhombic structure joined to typical peaks of the face-centered cubic phases of AgCl and AgBr and the hexagonal AgI phase, indicating the successful formation of the composite structures. Interestingly, the  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> crystals are stabilized in AgBr/ $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> heterostructures when cetyltrimethylammonium bromide is used as the AgBr precursor [185]. The FE-SEM image of the as-prepared composite consisted of uniform rod-like morphologies for Ag<sub>2</sub>WO<sub>4</sub>, while the AgX nanocrystals were uniformly dispersed on the surface of Ag<sub>2</sub>WO<sub>4</sub>. The formation of the Ag<sub>2</sub>WO<sub>4</sub>@AgX core@shell or heterostructure is evident for some composites [26, 27, 40, 182]. A schematic procedure for the synthesis and final microstructure is shown in **Fig. 8**.

<Insert Fig. 8>



For all  $\text{Ag}_2\text{WO}_4/\text{AgX}$  hybrid nanorods, a pronounced enhancement in the visible-light absorption range and an absorptive capability stronger than that of  $\text{Ag}_2\text{WO}_4$  are clearly observed in the UV–vis diffuse reflection spectra [40, 98, 182]. This enhancement in the optical property was achieved when the AgX content was progressively increased. Notably, among all AgX hybrids, the  $\text{Ag}_2\text{WO}_4/\text{AgBr}$  composite shows a much stronger response in the visible region. The AgX/ $\text{Ag}_2\text{WO}_4$  heterojunctions can improve the visible light response of  $\text{Ag}_2\text{WO}_4$  crystals, and thus, they can be expected to be efficient visible-light-driven photocatalysts. Through the appropriate procedure (mostly by photoirradiation), some of these binary composites are transformed into Ag/AgX/ $\text{Ag}_2\text{WO}_4$  ternary composites [27, 61, 181, 183]. In addition to the advantages already mentioned for similar binary systems, the highly enhanced performance of the Ag-modified AgX/ $\text{Ag}_2\text{WO}_4$  hybrid can be attributed to the SPR effects of metallic Ag NPs.  $\text{Ag}^0$  growth from AgX/ $\text{Ag}_2\text{WO}_4$  in the nanometer range increases the total surface area and improves the synergistic effects, which can effectively suppress the recombination of photo-induced electrons and holes.

Additionally, these Ag-based binary systems were used to improve the structural/optical performance of several semiconductor oxides, producing multicomponent composites for enhancing the photocatalytic behavior of these oxides [184, 187, 188].  $\text{Ag}_2\text{WO}_4$ -AgX (X = Cl, Br, I)-sensitized  $\text{TiO}_2$  nanotubes (NTs) arrays ( $\text{TiO}_2$  NTs were prepared by two-step anodization of Ti foils) were prepared *in-situ* by successive ionic layer adsorption and reaction deposition of  $\text{Ag}_2\text{WO}_4$  NPs on  $\text{TiO}_2$  NTs [187]. The as-prepared  $\text{TiO}_2$  NTs/ $\text{Ag}_2\text{WO}_4$  was immersed into  $\text{X}^-$  solution, and by ion-exchange

hydrothermally assisted (180 °C for 3 h), TiO<sub>2</sub> NTs/Ag<sub>2</sub>WO<sub>4</sub>-AgX (X = Cl, Br, I) heterojunctions were formed. Similarly, TiO<sub>2</sub> NPs were used in this method (*in-situ* preparation by successive ionic layer adsorption and reaction deposition) to prepare TiO<sub>2</sub>/Ag<sub>2</sub>WO<sub>4</sub>/AgBr nanocomposites by the ionic exchange in a suspension refluxed at 96 °C for 15 min [188]. Also, by this multi-step procedure, the ZnO/Ag/Ag<sub>2</sub>WO<sub>4</sub>/AgI, with additional photoirradiation [189] and TGCN/Ag/Ag<sub>2</sub>WO<sub>4</sub>/AgI (TGCN=tubular g-C<sub>3</sub>N<sub>4</sub>) [190], quaternary composites were obtained and used for photocatalytic removal of MB, methyl orange (MO), and fuchsine dyes; and antibiotics such as TC and amoxicillin, under visible-light illumination using Ag<sub>2</sub>WO<sub>4</sub> for environmental applications.

#### **6.4. Ag<sub>2</sub>WO<sub>4</sub> composites with metal oxides and metal-organic compounds**

A particularly interesting strategy for preparing Ag<sub>2</sub>WO<sub>4</sub>-based composites is mixing with magnetic oxides. It should be noted that after coupling with Ag<sub>2</sub>WO<sub>4</sub>, the magnetic behavior of these oxides is still retained, which suggests a potential aptness of the composites for magnetic separation and recovery using a magnet.

Based on this strategy, binary (Ag<sub>2</sub>WO<sub>4</sub>/Fe<sub>3</sub>O<sub>4</sub>) [191], ternary (Ag/β-Ag<sub>2</sub>WO<sub>4</sub>/CoFe<sub>2</sub>O<sub>4</sub>, Ag<sub>2</sub>WO<sub>4</sub>@ZnO@Fe<sub>3</sub>O<sub>4</sub> and SiO<sub>2</sub>/Fe<sub>3</sub>O<sub>4</sub>/Ag<sub>2</sub>WO<sub>4</sub>) [192-194] and quaternary (g-C<sub>3</sub>N<sub>4</sub>/Fe<sub>3</sub>O<sub>4</sub>/Ag<sub>2</sub>WO<sub>4</sub>/AgBr and Fe<sub>3</sub>O<sub>4</sub>@SiO<sub>2</sub>@Ag<sub>2</sub>WO<sub>4</sub>@Ag<sub>2</sub>S) [195, 196] systems have been studied. The Fe<sub>3</sub>O<sub>4</sub>/Ag<sub>2</sub>WO<sub>4</sub> nanocomposites were obtained through facile routes with cubic Fe<sub>3</sub>O<sub>4</sub> and β- and α-Ag<sub>2</sub>WO<sub>4</sub>. Images clearly show that Fe<sub>3</sub>O<sub>4</sub> particles are randomly attached to the surface of Ag<sub>2</sub>WO<sub>4</sub> nanorods. For the ternary systems, CoFe<sub>2</sub>O<sub>4</sub> and ZnO/Fe<sub>3</sub>O<sub>4</sub> solutions were previously dispersed, and then the

solutions of  $\text{Ag}_2\text{WO}_4$  precursors were subsequently mixed. The  $\text{Ag}_2\text{WO}_4/\text{CoFe}_2\text{O}_4$  composite (a flower-like morphology) was finally irradiated under a halogen lamp to obtain the  $\text{Ag}/\text{Ag}_2\text{WO}_4/\text{CoFe}_2\text{O}_4$  ternary composite. At the same time, the  $\text{Ag}_2\text{WO}_4/\text{ZnO}/\text{Fe}_3\text{O}_4$  was transferred into a Teflon-lined autoclave and heated at  $140\text{ }^\circ\text{C}$  for 2 h. As expected, by the simple deposition/precipitation route (to obtain  $\text{Ag}/\text{Ag}_2\text{WO}_4/\text{CoFe}_2\text{O}_4$ ) and by the hydrothermal route (to obtain  $\text{Ag}_2\text{WO}_4/\text{ZnO}/\text{Fe}_3\text{O}_4$ ),  $\beta\text{-Ag}_2\text{WO}_4$  and  $\alpha\text{-Ag}_2\text{WO}_4$ , respectively, were revealed by XRD. The  $\text{g-C}_3\text{N}_4/\text{Fe}_3\text{O}_4/\text{Ag}_2\text{WO}_4/\text{AgBr}$  quaternary composite was prepared by the refluxing method. For this route,  $\beta\text{-Ag}_2\text{WO}_4$  with a hexagonal structure appeared in the nanocomposite. The magnetic properties were measured, and it should be noted that the magnetic behavior of the oxides is still retained after coupling in the composite. However, due to the presence of a nonmagnetic  $\text{Ag}_2\text{WO}_4$  counterpart in the composite, the magnetic parameter values are much lower than that of pure magnetic oxide. These results confirm that the composites can be a recoverable material, magnetically collected and reused.

On the other hand, nonmagnetic semiconductor oxides have also been used in composites joined to  $\text{Ag}_2\text{WO}_4$ .  $\text{TiO}_2$ ,  $\text{WO}_3$ , and  $\text{ZnO}$  are among the common oxide materials used due to their good physical and chemical stability, non-toxicity, and excellent optics, biology, and photochemistry properties. However, these pure oxides are problematic due to their relatively wide band gap and high recombination rate for photoexcited charge carriers. Hence, it is necessary to tune this characteristic to enhance efficiency in diverse applications, and the oxide/ $\text{Ag}_2\text{WO}_4$  composite has been used for this purpose. Composite systems with simple oxides such as  $\text{Ag}/(\text{ZnO}, \text{Mn}_3\text{O}_4, \text{WO}_3)/\text{Ag}_2\text{WO}_4$  [197-204],  $\beta\text{-}$

$\text{Bi}_2\text{O}_3/\text{Ag}_2\text{WO}_4$  [205],  $\text{MnWO}_4/\text{Ag}_2\text{WO}_4$  [206],  $\text{Ag}_2\text{WO}_4/\text{CeO}_2$  [207],  $\alpha\text{-Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$  [208],  $\text{CoO-Ag}_2\text{WO}_4$  [209, 210],  $\text{ZnBi}_2\text{O}_4/\text{Ag}_2\text{WO}_4$  [211], and  $\text{Ag}_2\text{WO}_4/(\text{TiO}_2, \text{ZnO})/\text{AgX}$  ( $\text{X} = \text{Cl}, \text{Br}, \text{I}$ ) [187-189] have been obtained for diverse synthetic routes. These last three heterojunctions have been described in the present work in the section on composites with  $\text{Ag}_2\text{WO}_4$  crystals and halogen-based materials. Other more complex oxides have also been used [212-220]. In short, some results on preparation methods, phases, and morphologies reported in previous works are summarized in **Table 1**. Most of these reports explore applications of photocatalytic properties (as seen in **Table 1**) due to the fact that the fabrication of  $\text{Ag}_2\text{WO}_4$ -based composites has proven to be an important research strategy for further enhancement of the photodegradation of organic pollutants.

Similarly, metal-organic framework (MOF) compounds have emerged as potential candidates to form composites with  $\text{Ag}_2\text{WO}_4$  crystals. These metal complexes have interesting electron transfer abilities and a large number of derivatives, among other promising characteristics for several electronic and optoelectronic applications. Some reports of  $\text{Ag}_2\text{WO}_4$  composites with these compounds [103, 221-223] are listed in **Table 1**.

<Insert Table 1>

In summary, the composite catalyst systems exhibit higher photoactivity than their respective single counterpart catalysts, which can be attributed to the excellent synergy effects. Among others, higher photoexcited electron-hole pair separation, ease of interfacial charge transfer, higher surface sensitivity, and excellent stability are essential characteristics for the successful application of  $\text{Ag}_2\text{WO}_4/\text{oxide}$  and  $\text{Ag}_2\text{WO}_4/\text{metal-organic}$  composites.

This formation of heterojunctions proves that it is a suitable strategy to improve several  $\text{Ag}_2\text{WO}_4$  applications, such as the photocatalytic degradation of organic dyes, biocide activity, and others. The adequate energy bands match between  $\text{Ag}_2\text{WO}_4$  and the other semiconductor reduced the free charges recombination in the composite, ascribed to the rapid and effective separation and transportation of the photogenerated electron-hole pairs, and also enhanced the photocorrosion inhibition with a small  $\text{Ag}_2\text{WO}_4$  self-consumption. Further, it is worth pointing out that the addition of another semiconductor with a narrow band gap overcomes the low-efficiency  $\text{Ag}_2\text{WO}_4$  in the utilization in visible light regions. When  $\text{Ag}_2\text{WO}_4/\text{gC}_3\text{N}_4$  photocatalysts, for example, were exposed under visible light for degradation of RhB, a typical organic dyes pollutant, the photocatalytic degradation efficiency achieves optimal values that are >90% within 20-50 min under irradiation [152, 159, 224]. Similar efficient performances were obtained when  $\text{Ag}_2\text{WO}_4$  is hybridized with silver halide,  $\text{AgX}$  ( $X = \text{Cl}, \text{Br}, \text{I}$ ). The photocatalysis for the RhB dye showed that the  $\text{AgBr}/\text{Ag}_2\text{WO}_4$  composite had the most notable photoactivity among them. The complete RhB degradation time is 0.5 h, while  $\text{AgCl}/\text{Ag}_2\text{WO}_4$ ,  $\text{AgI}/\text{Ag}_2\text{WO}_4$ , and pure  $\text{Ag}_2\text{WO}_4$  catalysts took about 2, 1.5, and 4 h, respectively [182]. Also, 100% of degradation efficiency at 60 min for  $\text{AgBr}/\text{Ag}_2\text{WO}_4$  had been reported [185]. Other organics non-azo dyes have also been effectively degraded under visible light such as: MO with  $\text{gC}_3\text{N}_4/\text{Ag}_2\text{WO}_4$  [40, 152, 154],  $\text{Ag}_2\text{S}/\text{Ag}_2\text{WO}_4$  [169] and  $\text{Ag}_2\text{WO}_4/\text{MoS}_2$  [172]; and methyl blue (MB) with  $\text{Ag}_2\text{WO}_4/\text{VS}_4$  [173],  $\text{Ag}_2\text{WO}_4\text{-MnS}$  [176],  $\text{CoS}/\text{Ag}_2\text{WO}_4$  [175] and  $\text{AgCl}/\text{Ag}_2\text{WO}_4$  [61]. Similarly, contaminating drugs such as indomethacin in  $\text{Ag}_2\text{WO}_4/\text{gC}_3\text{N}_4$  [156]; para-chlorophenol [40], and ceftriaxone, in  $\text{AgI}/\text{Ag}_2\text{WO}_4$  [186], among others, underwent photochemical degradation under visible light.

In several works, photocatalytic disinfection efficiency of the excited photogenerated electron-hole pairs in the composite has strong oxidative capacities that could attack, inactivate and break the semipermeable cellular membrane and eventually result in cell death. This enhanced visible light antibacterial activity has been reported against *E. coli* in g-C<sub>3</sub>N<sub>4</sub>/Ag<sub>2</sub>WO<sub>4</sub> [153, 155]; and for *E. coli* and *B. subtilis* for several composites of Ag<sub>2</sub>WO<sub>4</sub> hybridized with chalcogenides based-compounds [153, 173-175].

In terms of the successful heterojunction design, several aspects must be taken into consideration according to the suitable bands (VB and CB) alignment of the adjacent semiconductors [225, 226]. In particular, for the Ag<sub>2</sub>WO<sub>4</sub> composites, the Z-scheme heterojunction [154-156, 167, 185, 199, 227-230] and conventional type II heterojunction [169, 205, 207, 211, 213, 224] are the most employed. A schematic representation of both types of Ag<sub>2</sub>WO<sub>4</sub> heterojunctions is displayed in **Fig. 9**.

<Insert Fig. 9>

## 7. Interaction with electrons and laser

Electrons and electromagnetic radiation interact with matter and can change the physical and chemical properties of a material. Therefore, techniques such as EB and fs laser irradiation have emerged as modern, clean, and efficient tools for tuning material properties. FE-SEM and TEM microscopes have recently been employed as EB sources to produce metal NPs on the surfaces of semiconductor materials. At the same time, it is

possible to observe the *in-situ* growth of these NPs and the phenomena involved. Different metal NPs, such as Au, Cu, Li, Co, and others, have already been generated on the surfaces of semiconductor materials by this technique [231-236]. Ag NPs are well known for their attractive properties, and their growth on the surface of Ag-based semiconductors can improve some properties, such as antimicrobial activity, gas sensing, and photocatalysis. Thus, several studies have been published related to the growth of Ag NPs stimulated by EB irradiation on the surfaces of Ag-based ternary oxides, such as  $\text{Ag}_2\text{WO}_4$ ,  $\beta\text{-Ag}_2\text{MoO}_4$ ,  $\text{Ag}_3\text{PO}_4$ ,  $\beta\text{-AgVO}_3$ , and  $\text{Ag}_2\text{CrO}_4$  [237-241].

Irradiation of solids with energetic particles or light, such as electrons or lasers, typically gives rise to the formation of atomic defects in the target and spoils the material properties. The necessity to understand the irradiation-induced processes was the initial driving force for studying the effects of irradiation on solids. Recently, great advances in nanoscience and nanotechnology fields have been made due to the expansion and diffusion observed employing TEM and high resolution (HR)-TEM images, which is possible using TEM heating holders for *in-situ* electron microscopy [242-246]. However, new *in-situ* observations of the crystal growth of different materials from TEM [247, 248] and *in-situ* liquid TEM have shed light on understanding the growth process of copper and lead sulfide nanostructures [249-251]. Moreover, the growth mechanisms of some nanomaterials can be observed simply through the watchful eyes of the researcher and microscopist [252], without the use of extra accessories under the microscope [253, 254]. This point is of great relevance to disseminating this new method of investigating the stages of the crystal growth process due to the possibility of observing all evolution step-by-step at the nanoscale [255,

256]. In general, several works have been reported using TEM heating holders to monitor the growth process of different nanomaterials, such as bismuth [257], germanium [258], indium-arsenide [259], and vanadium oxide [260]. Moreover, for the first time, our research group [18, 20, 31, 113, 261] observed the *in-situ* growth of metallic Ag nanorods from  $\alpha$ - and  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> crystals with unstable surfaces through FE-SEM images without a heating holder, as shown in **Figs. 10(a-c)**.

<Insert Figs. 10(a-c)>

A series of *in-situ* FE-SEM images were acquired during Ag NP growth from the unstable surface of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals at 30 kV without a heating holder. **Fig. 10(a)** illustrates an initial FE-SEM image of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals, which was acquired soon after rapid approach and focus adjustment. These crystals have a hexagonal rod-like elongated shape (**Fig. 10(a)**) and were modeled from the crystallographic data presented. These crystals are formed rapidly by mixing Ag<sup>+</sup> cations and WO<sub>4</sub><sup>2-</sup> ions into a hot aqueous solution (90 °C). This synthesis method promotes fast kinetics of nucleation and the growth of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals. **Fig. 10(b)** shows that when these crystals under a high vacuum ( $1 \times 10^{-5}$  Pa) are exposed to electrons provided by a microscope after 6 min, it is possible to observe the growth process of metallic Ag NPs on the hexagonal rod-like  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals. After the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals receive a reasonable amount of electrons, the appearance of several surface defects can be noted due to the mobility of metallic Ag NPs to the surface of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals, as illustrated in **Fig. 10(b)**. Several FE-SEM experiments and the corresponding images demonstrate that the growth process of metallic Ag nanorods is highly reproducible. In **Fig. 10(b)**, it can be observed a large amount of



metallic Ag nanorods formed on the surface of the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals after 10 min of exposure to electrons from the microscope. Due to this EB process, a structural transformation or symmetry break of the orthorhombic structure occurs, resulting in the *in-situ* formation of Ag<sup>0</sup> nanorods and defects in the  $\alpha$ -Ag<sub>2-x</sub>WO<sub>4</sub> crystals. We have modeled this EB process and the growth of these metallic Ag<sup>0</sup> nanorods with the formation of disordered  $\alpha$ -Ag<sub>2-x</sub>WO<sub>4</sub> crystals based on FE-SEM images (**Fig. 10(b)**). Tsuji et al. [262] monitored the rapid transformation of various Ag nanostructures using the chemical reduction method from Ag<sup>+</sup> cations to Ag<sup>0</sup>. However, the EB process uses only electrons from the microscope to promote the reduction of angular [AgO<sub>2</sub>] clusters present in  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals to metallic dodecahedral [AgAg<sub>12</sub>] clusters. Finally, in **Fig. 10(c)**, a significant change in the  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> surface utilizing energy-dispersive X-ray (EDX) imaging can be noted by mapping Ag-L3, which is due to the formation of Ag<sup>0</sup> NPs at other regions of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals through an absorption process of some Ag<sup>0</sup> NPs and amorphization of part of the orthorhombic structure of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub>.

### **7.1. The growth of Ag NPs on Ag<sub>2</sub>WO<sub>4</sub> crystals by electron beam irradiation**

The *in-situ* growth of Ag NPs on the surface of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> under an accelerated EB from a high *vacuum* electron microscope was reported for the first time by our group in 2013. A sample of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> was illuminated inside a TEM operated at 200 KV, and images of different moments of exposition, with intervals of 5 seconds, were registered. They observed Ag filaments growing on the semiconductor surface, with posterior reabsorption within time. They also observed these phenomena with an SEM microscope,

but the process was slower and less intense in this case. However, in SEM and TEM, the NPs grow relatively quickly, with just a tiny electron dose used to focus the image [12]. Several subsequent studies from our group investigated the growth of Ag NPs on the  $\text{Ag}_2\text{WO}_4$  surface from the EB irradiation via *in-situ* analysis using SEM and TEM, combined with theoretical calculations. One of them reported the influence of the synthesis method on growth characteristics, suggesting that order-disorder structural parameters influence nanocrystal growth [17].  $\text{Ag}_2\text{WO}_4$  is an *n*-type semiconductor, and with EB irradiation, randomly distributed Ag vacancies form in the crystal. This phenomenon produces *p*-type defects in the structure, creating a new intrinsic *n/p*-type semiconductor with novel properties. This fact was confirmed when an exponential increase in the bactericidal property was observed with the *n*-type semiconductor, as shown in another publication [19].

The early events related to the nucleation of Ag filaments on  $\alpha\text{-Ag}_2\text{WO}_4$  crystals, driven by the EB microscope under a high *vacuum*, were studied in detail by our research group using a combination of theory, computation, and experiments. These studies demonstrated that the Ag nucleation and formation on  $\alpha\text{-Ag}_2\text{WO}_4$  crystals is a result of order/disorder effects generated in the crystal by EB irradiation, associated with structural and electronic changes of the angular  $[\text{AgO}_2]$  clusters and distorted tetrahedral  $[\text{AgO}_4]$  clusters and, to a minor extent, to the distorted octahedral  $[\text{WO}_6]$  clusters, which collectively represent the constituent building blocks of the  $\alpha\text{-Ag}_2\text{WO}_4$  [18]. To further investigate and understand the phenomena involved in the growth of Ag NPs on  $\alpha\text{-Ag}_2\text{WO}_4$  crystals, another combined experimental and theoretical study was performed using *in-situ*

SEM and TEM images and density functional theory. The EB irradiation by using SEM was generated at four different electron voltages (5, 10, 15, and 20 kV), whereas TEM images were obtained at 200 kV. The main contributions of that work were that first, the start of Ag NP growth occurs in a few seconds of exposure to the EB independent of the voltage, although there is an increase in the number of NPs for higher voltages; and second, the Ag centers at the (100) surface suffer a decrease in the activation barrier for diffusion processes with the injection of electrons [20]. Other mechanistic aspects of the formation of Ag NPs by EB irradiation were discussed by our research group [263] and a review on the topic was also published later [264].

Despite all these studies related to the growth of Ag NPs on  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub>, the description of the atomic and electronic structure of the semiconductor irradiated by electrons remained unclear until our recent work [261]. This study concluded that there is a predominant growth of Ag on the (100) surface of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub>, where Ag<sup>0</sup> species (with different coordination numbers) can diffuse out of the surface with a very low energy barrier (less than 0.1 eV). Thus, they initiate the growth of metallic Ag nanostructures and leave Ag vacancies in the bulk material, which increases the structural disorder of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> crystals and their electrical resistance [261]. The morphological evolution of Ag NPs in response to the EB passage was investigated in detail [265]. Different mechanisms were observed, such as mass transport, site-selective coalescence, structural rearrangement during and after coalescence or dissolution, and the evolution of the growth mechanism with surface faceting leading to a more stable configuration.

Additionally, the nature of the sintering process (the phenomena involved in the junction of the grain boundaries of two initially isolated particles that bond together to form larger NPs) of Ag NPs was unraveled at an atomic resolution for the first time [266]. This study concluded that the EB of TEM produces SPR in the Ag, forming neutral nanoelectric dipoles responsible for the coalescence processes, which start with a disorder-driven pathway followed by a structural disorder-to-order transition *via* an oriented attachment mechanism. The resulting larger clusters are thus a consequence of these sintering processes controlled by dipole interactions [267].

Moreover, the dose-dependent effect (0, 2, 4, and 6 kGy from a variable energy accelerator) of EB irradiation on the structural and optical properties of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> NPs was studied by Sreedev *et al.* [268]. The findings reveal that EB irradiation induces particle size variation, increases strain, surface defects, optical band gap deviations, and PL and Raman spectra modifications, changing the optical and photoluminescent properties of  $\alpha$ -Ag<sub>2</sub>WO<sub>4</sub> [268]. Recently, for the first time, our research group [113] showed the presence of some silver oxides (Ag<sub>2</sub>O and Ag<sub>3</sub>O<sub>4</sub>) close to the interface of the mother Ag<sub>2</sub>WO<sub>4</sub> crystal, in addition to the predominance of metallic Ag in the extruded material from the electron-irradiated Ag<sub>2</sub>WO<sub>4</sub>, which suggests that the EB induced atomic diffusion of O and Ag species in the microcrystals. The *in-situ* formation, growth, and subsequent reabsorption of Ag NPs on the surface of the  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> metastable phase was reported, for the first time, in 2016 [269]. After that, we examined the structural properties and electronic behavior of  $\beta$ -Ag<sub>2</sub>WO<sub>4</sub> irradiated by an EB by joint an experimental and theoretical investigation. Based on the analysis of the evolution of geometry and electronic structure,

the Ag nucleation process takes place via the reduction of the  $\text{Ag}^+$  of the  $[\text{AgO}_6]$  and  $[\text{AgO}_5]$  clusters [31].

## **7.2. Ag NP growth on $\text{Ag}_2\text{WO}_4$ by femtosecond laser irradiation**

The laser ablation of solids is a material processing technique that has received much attention from the scientific community over the last few years. The interactions of ultrafast pulses from a laser source with matter can also be an efficient tool for generating metal NPs, with the advantage of a more straightforward scale-up compared to the TEM or SEM electron beams. Although a few articles refer to laser ablation in liquid to modulate and enhance some properties of  $\text{Ag}_2\text{WO}_4$  crystals, very recently, Lin et al. [30], report the use of laser ablation to generate Ag NPs on solid  $\text{Ag}_2\text{WO}_4$ .

The first paper on the production of Ag NPs on  $\text{Ag}_2\text{WO}_4$  surfaces using fs laser irradiation was published in 2018 [3]. This study reported the novel formation of Ag NPs from  $\alpha\text{-Ag}_2\text{WO}_4$  by laser-assisted irradiation. The Ag NPs generated were spherical, isolated, or agglomerated, and they did not grow with increasing exposure time under the electron microscope, unlike what was observed in EB, as previously discussed. The authors attributed the short-pulse laser interaction with matter to a mechanism not fully understood but based mainly on the laser pulse energy absorption by the electrons on the sample's surface in an instantaneous practical process. Thus, the excitation of electrons contrasts with the basal state of the electrons in the lattice, which leads to an equilibrium that is reached through rapid electron-phonon interactions and electron diffusion out of the excited region. The ablation occurs when the excitation is high enough, which is different from the

EB irradiation process, which occurs in sequential steps. The fs laser-irradiated samples presented an improved bactericidal activity compared to the non-irradiated  $\text{Ag}_2\text{WO}_4$  material.

The peculiar phenomena caused by the interactions of ultra-short laser pulses and matter generates unexpected or even non-equilibrium metastable materials. As an example, we have reported [270] the formation of a nanoalloy formed by Ag and Bi, which are immiscible in bulk. Still, the fs laser irradiation of a simple mixture of the precursors  $\text{Ag}_2\text{WO}_4$  and  $\text{NaBiO}_3$  obtained a random mixture of up to  $6\pm 1$  atom % of Bi into the face-centered cubic structure of Ag. This partial Ag-Bi miscibility was also discussed recently from *ab initio* DFT calculations [271]. Another study reports a theoretical strategy based also on DFT and molecular dynamics simulations to investigate the effect of laser irradiation on semiconductor materials [272]. This methodology provides insights into the mechanisms of local disordering induced by the laser and conducting to the formation of metallic Ag clusters [273]. Irradiation of fs laser on  $\alpha\text{-Ag}_2\text{WO}_4$  provokes a 64-fold enhancement in the bactericidal activity, mainly due to the generation of metastable hexagonal  $\text{Ag}^0$  on the surface of the material [110].

In summary, the efficiency of both novel techniques in the generation of new or enhanced materials was shown very recently by the application of (micro, nano)composites of  $\text{Ag}_2\text{WO}_4$  and Ag NPs synthesized by EB and fs irradiation against tumor cells, both having enhanced effects against these tumor cells *via in vitro* assays, reassuring the importance of both tools in the superior material production [4]. Moreover, a very recent study demonstrated unconventional magnetic effects on  $\text{Ag}_2\text{WO}_4$  by these irradiation

techniques, which opens the possibility for further explorations [274]. Also, the influence of morphology, surface band gap, and microstructure on the photoluminescence emissions of  $\alpha$ - $\text{Ag}_2\text{WO}_4$  samples obtained by the CP method, followed by treatment with MAH at different synthesis times and two different irradiation processes, EB and fs laser, has been experimentally and theoretically investigated [275].

**Fig. 11** presents a timeline (from 1973 until 2022) highlighting the key milestones and advances in the synthesis and applications of  $\alpha$ -,  $\beta$ -, and  $\gamma$ - $\text{Ag}_2\text{WO}_4$  polymorphs.

<Insert Fig. 11>

## 8. Summary and future work

$\text{Ag}_2\text{WO}_4$ -based materials have seen rapid growth in the last few years, with new examples of the different types and new applications being reported more frequently. As the field progresses, it can also be envisioned that structure and electronic properties are responsible for a wide range of technological applications. To enable their widespread practical usage, significant efforts have been devoted to finding relationships among the nature of chemical bonds, geometrical arrangements, and their unique properties. This review summarizes their rich history and highlights the contemporary relevance for assessing the most important properties and applications of  $\text{Ag}_2\text{WO}_4$ -based materials, as well as the fundamentals and recent progress of  $\text{Ag}_2\text{WO}_4$  as multifunctional materials. To this end, new concepts and innovative strategies are discussed in detail. Special emphasis is devoted

to synthesis methods. Overall, this review article aims to help materials science chemists and physicists select suitable methods and techniques for the synthesis and characterization of semiconductor  $\text{Ag}_2\text{WO}_4$ -based materials and potentially other relevant materials. However, the changes in the local coordination of Ag and W cations and their electronic structure are difficult to observe directly by experimental methods. The combination of the results derived from experimental and first-principles calculations is a feasible strategy for studying these materials and should be efficiently transformed into a potent tool to explain and rationalize the properties and affordable applications. Widely spread crystal lattices of  $\text{Ag}_2\text{WO}_4$  polymorphs represent a natural flexible platform for the chemical design of various advanced functional materials with unique features.

The photocatalytic efficiency could be improved by morphology control to obtain superior performance, which provides more active sites for the corresponding reaction. In this strategy, various additives, solvents, and surfactants are used to fabricate  $\text{Ag}_2\text{WO}_4$ -based materials with controlled morphology. The synthetic methods and the photocatalytic behavior of numerous morphologies were introduced systematically. Our thorough analysis of the geometric structure, morphology and electronic and optical properties of  $\text{Ag}_2\text{WO}_4$  polymorphs reveals fundamental insights into their chemistry and shows high variability that can be used to justify and guide additional experimental investigation into this complex material. The interaction between electrons/lasers and matter promotes major changes, such as the growth of Ag NPs in  $\text{Ag}_2\text{WO}_4$ -based materials. This work reviews the ultimate applications of  $\text{Ag}_2\text{WO}_4$ -based materials when exposed to EB and *fs* laser irradiation, leading to the enhancement of photocatalytic and bactericidal activities. This review serves



the purpose of informing researchers in this interdisciplinary field about a unique approach based on the combination of theory, simulation, and experiments. We hope that this study will provide valuable information and inspiration for the study of  $\text{Ag}_2\text{WO}_4$ -based materials and that future advances will drive rapid development in this field. This roadmap will prove to be valuable both as a collective self-analysis of the materials science community, providing an updated picture, and as a dissemination tool, helpful to whoever is willing to spread knowledge and thus draw both new young scientists and further public and private resources to this exciting field of research.

From our viewpoint, future research needs to focus on the exploration of the formation of reactive oxygen species (ROS) over semiconductor surfaces is a topic of tremendous scientific attention owing to their essential role in photocatalytic and biological processes. In particular, the activation of both molecular oxygen,  $\text{O}_2$ , and water,  $\text{H}_2\text{O}$ , molecules is a fundamental step in almost all oxidation reactions involving ROS formation, including water treatment and photodegradation of pollutants due to their high chemical reactivity. Since ROS generation occurs on the surface of semiconductors, its efficiency is related to surface electronic and structural features. Specific chemical composition and crystal arrangements at the atomic level may contribute to the different performances. Therefore, precise design to balance the advantage of active sites at the morphology remains challenging. A comprehensive understanding of the mechanisms governing the nature of the active sites at the morphology of the materials thus entails a description of the defective states at the exposed surfaces in terms of their concentration, local structure, and spatial distribution since all of these affect the chemistry that the material is able to drive.

Inspired by the present results, high-throughput screening, machine learning, and computational materials design can guide for developing synthesis methods of  $\text{Ag}_2\text{WO}_4$ -based materials, thereby shortening the experimental period, reducing the workload, and saving experimental costs. In other fundamental studies, combining experiment and theory represents a promising research field for understanding the principles of activity, screening alternative high-performance and to develop advanced techniques for large-scale production.

Finally, the benefits of these materials for utilization in the fields of energy, environment, catalysis, and biological sciences are consistently analyzed, and several perspectives are presented. Nonetheless, since the biological properties of  $\text{Ag}_2\text{WO}_4$ -based nanomaterials nanoparticles are still in their infancy, future detailed *in vitro* and *in vivo* studies to assess the detailed biocompatibility of these nanoparticles are warranted. Hence, more examinations are expected to investigate their tunable properties and performance in interdisciplinary science. Moreover, a future perspective is also provided to shed light on the development of highly efficient  $\text{Ag}_2\text{WO}_4$ -based material. At the same time, their revealed architecture, chemistry, and morphological requirements allow us to believe that these materials will come into commercial use; at least, it is a prediction to optimistically believe. Most of the studies on the technological applications of  $\text{Ag}_2\text{WO}_4$ -based materials are limited to laboratory research, and more efforts are still needed to advance their practical use. Indeed, these and related materials are analyzed, and several possible research directions for overcoming the challenges are proposed to facilitate further research and development. Further efforts are expected to propel this exciting field forward. We hope the

discussions are of general interest and will stimulate additional developments for the next generation of advanced materials related to  $\text{Ag}_2\text{WO}_4$  materials to help push the field forward in meaningful directions.

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## Figure Captions

**Fig. 1.** Schematic representation of the unit cell corresponding to (a) orthorhombic  $\alpha$ - $\text{Ag}_2\text{WO}_4$ , (b) hexagonal  $\beta$ - $\text{Ag}_2\text{WO}_4$ , and (c) cubic  $\gamma$ - $\text{Ag}_2\text{WO}_4$ . The cluster coordination numbers are highlighted.

**Fig. 2.** Schematic representation synthesis of  $\alpha$ - $\text{Ag}_2\text{WO}_4$  ceramics by the solid-state reaction method. (a) 1:1 stoichiometric mixture of the respective oxides ( $\text{Ag}_2\text{O}$  and  $\text{WO}_3$ ), (b) several grinding cycles for oxides homogenization and surface contact improvement, (c) the final stage with calcination between (350—950 °C) at different times (2—24 h) of the homogenized oxides after several grinding cycles, respectively.

**Fig. 3.** Some procedures applied in the synthesis of the different polymorphs of  $\text{Ag}_2\text{WO}_4$ -based materials.

**Fig. 4.** A schematic representation of some available morphologies for  $\alpha$ - $\text{Ag}_2\text{WO}_4$ ,  $\beta$ - $\text{Ag}_2\text{WO}_4$ , and  $\gamma$ - $\text{Ag}_2\text{WO}_4$  polymorphs. The exposed surfaces at the morphology are highlighted.

**Fig. 5.** Various strategies used to improve the activity of  $\text{Ag}_2\text{WO}_4$  polymorphs.

**Fig. 6.**  $\alpha$ - $\text{Ag}_2\text{WO}_4$  structure with doping atoms at W and Ag sites.

**Fig. 7.** Schematic representation of the  $\text{Ag}_2\text{WO}_4$  crystals grown on g- $\text{C}_3\text{N}_4$  sheet surfaces. The formation of g- $\text{C}_3\text{N}_4/\alpha$ - $\text{Ag}_2\text{WO}_4$  is energetically assisted by light, microwave (MW), and high temperature (T), while g- $\text{C}_3\text{N}_4/\beta$ - $\text{Ag}_2\text{WO}_4$  is favored under room temperature (T).

**Fig. 8.** Schematic synthetic procedure for the obtention of  $\text{Ag}_2\text{WO}_4/\text{AgX}$  core-shell or heterostructure.

**Fig. 9.** Schematic representation of the VB, CB, and Fermi Level for the semiconductors (p- and n-type) and for the  $\text{Ag}_2\text{WO}_4$  before and after the formation of the heterojunction.

**Fig. 10. (a)** Production of FE-SEM images from interacting electrons with  $\alpha\text{-Ag}_2\text{WO}_4$  structure, **(b)** Real-time FE-SEM images of Ag growth from  $\alpha\text{-Ag}_2\text{WO}_4$  structure: FE-SEM images taken with intervals of 2 min. FE-SEM of the same region where Ag filaments grow from the  $\alpha\text{-Ag}_2\text{WO}_4$  crystal surface. Ag filaments (blue arrows) grow while other Ag NPs are absorbed by the  $\alpha\text{-Ag}_2\text{WO}_4$  crystal surface (red arrows), indicating that there is a balance of clusters with the incidence of the EB, and **(c)** EDX image mapping Ag-L<sub>3</sub> from the  $\alpha\text{-Ag}_2\text{WO}_4$  crystal surface.

**Fig. 11.** Timeline showing some of the key milestones and advances in the field from 1973 until 2022.

**Table**

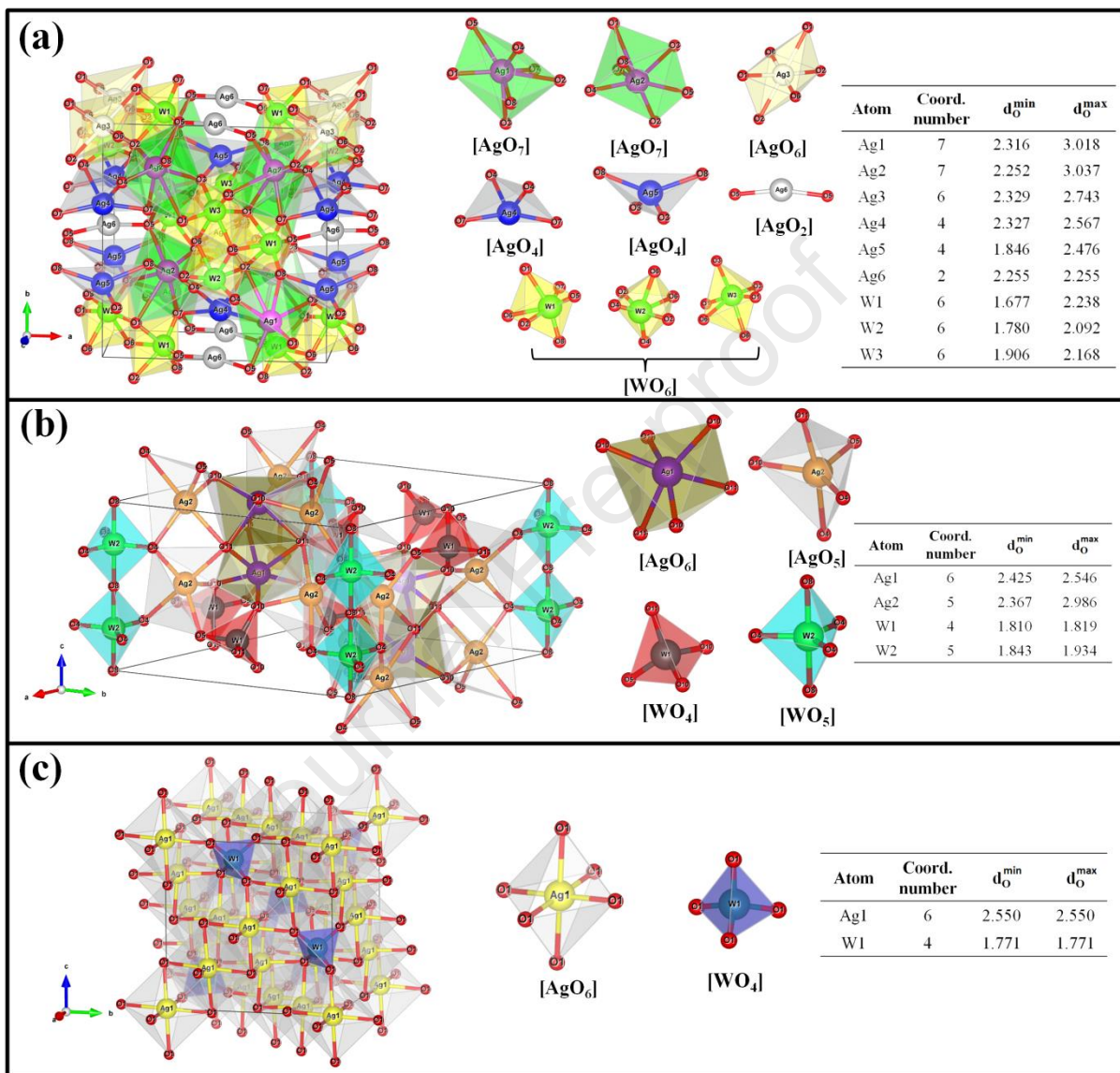
**Table 1.** The synthesis method, phase, and morphologies of  $\text{Ag}_2\text{WO}_4/\text{oxide}$  and  $\text{Ag}_2\text{WO}_4/\text{metal-organic}$  composites and their applications (RhB: Rhodamine B; TC-HCl: tetracycline hydrochloride; ThP: thiophene; MB: methylene blue; MO: methyl orange; VLI: visible-light irradiation).

Composite	Synthesis method	Phase	Morphology	Application	Ref.
<b>Binary Oxides</b>					
$\text{Ag}_2\text{WO}_4/\text{WO}_3$	Calcining method	$\alpha\text{-Ag}_2\text{WO}_4$	Hierarchical flower-like hollow	RhB decolorization under VLI	[197]
$\text{Ag}_2\text{WO}_4/\text{c-ZnO}$	Hydrothermal	$\alpha\text{-Ag}_2\text{WO}_4$	Nanorods with different average size distribution	Degrading the TC-HCl antibiotics	[199]
$\text{Ag}/\text{ZnO}/\text{Ag}_2\text{WO}_4$	Ultrasonic-irradiation	$\beta\text{-Ag}_2\text{WO}_4$	Micron Rods with different size	Photodegradation of RhB, MB, MO, and fuchsine under VLI	[200]
$\text{Mn}_3\text{O}_4/\text{Ag}_2\text{WO}_4$	Microwave-assisted precipitation	$\alpha\text{-Ag}_2\text{WO}_4$	Nanocomposites	Photooxidation of ThP under VLI	[201]
$\text{Ag}/\text{WO}_3/\text{Ag}_2\text{WO}_4$	Simple precipitation	$\beta\text{-Ag}_2\text{WO}_4$	$\text{Ag}/\text{Ag}_2\text{WO}_4$ decorated on $\text{WO}_3$ nanoplates	Degradation of RhB, MB, and MO under VLI	[276]
<b>Ternary Oxides</b>					
$\text{CaTiO}_3/\text{Ag}_2\text{WO}_4$	Coprecipitation	$\alpha\text{-Ag}_2\text{WO}_4$	Rod-shaped	Photoluminescence	[212]
$\text{CaWO}_4/\text{Ag}_2\text{WO}_4$	Hydrothermal	$\alpha\text{-Ag}_2\text{WO}_4$	Rod-like	MB degradation under UV-VLI.	[213]
$\text{Ag}_2\text{WO}_4/\text{Ag}_3\text{PO}_4$	Chemical precipitation method	not detected (low concent.)	Regular NPs	Mineralization of Bisphenol A under simulated solar light irradiation	[214]
$\text{Ag}/\beta\text{-Ag}_2\text{WO}_4/\text{BiVO}_4$	Hydrothermal and ultrasonic-	$\beta\text{-Ag}_2\text{WO}_4$	Spherically shaped	Selective oxidation of alcohols to	[215]

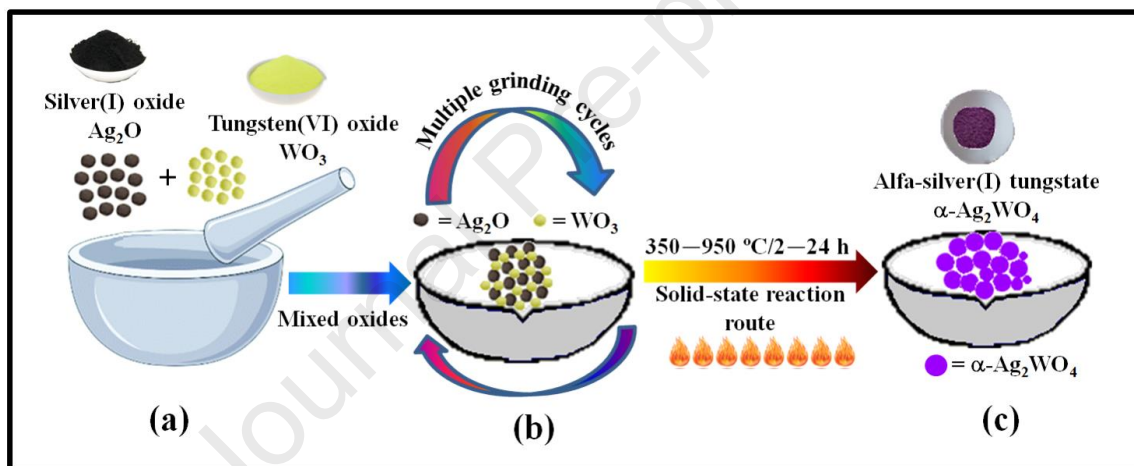


		irradiation	aldehydes.	
Ag <sub>2</sub> WO <sub>4</sub> /Ag /Bi <sub>2</sub> MoO <sub>6</sub>	Hydrothermal/ photochemical process	β-Ag <sub>2</sub> WO <sub>4</sub>	Ag <sub>2</sub> WO <sub>4</sub> /Ag NPs on Bi <sub>2</sub> MoO <sub>6</sub> nanosheets surface	Degradation of MB under 410 nm LED [216]
Ag <sub>2</sub> WO <sub>4</sub> /Bi <sub>2</sub> Fe <sub>4</sub> O <sub>9</sub>	Hydrothermal/ precipitation method	α-Ag <sub>2</sub> WO <sub>4</sub>	Ag <sub>2</sub> WO <sub>4</sub> NPs on Bi <sub>2</sub> Fe <sub>4</sub> O <sub>9</sub> surface	RhB degradation under visible light [217]
Ag <sub>2</sub> WO <sub>4</sub> /Bi <sub>2</sub> WO <sub>6</sub>	Hydrothermal- precipitation	β-Ag <sub>2</sub> WO <sub>4</sub>	Ag <sub>2</sub> WO <sub>4</sub> NPs on the Bi <sub>2</sub> WO <sub>6</sub> nanosheets	Degradation of RhB [227]
<b>Metal-Organic Compounds</b>				
Ti-based MOF (MIL-125-NH <sub>2</sub> )	Simple <i>in-situ</i> precipitation	α-Ag <sub>2</sub> WO <sub>4</sub>	spherical Ag <sub>2</sub> WO <sub>4</sub> NPs on micro MIL-125-NH <sub>2</sub> .	MB and RhB photodegradation under UV and VLI. [221]
Zr-based MOF (UiO-66)	Simple <i>in-situ</i> precipitation		Ag/Ag <sub>2</sub> WO <sub>4</sub> on UiO-66 surface	Photodegradation of PBS (azo dye) and RhB under VLI [222]
Co- phthalocyanine (CoPc)	Solvent evaporation	α-Ag <sub>2</sub> WO <sub>4</sub>	Irregular (rod/block like + spherical)	[223]
Fe- phthalocyanine (FePc)	Solvent evaporation	α-Ag <sub>2</sub> WO <sub>4</sub>	Irregular (rod/block like + spherical)	[103]

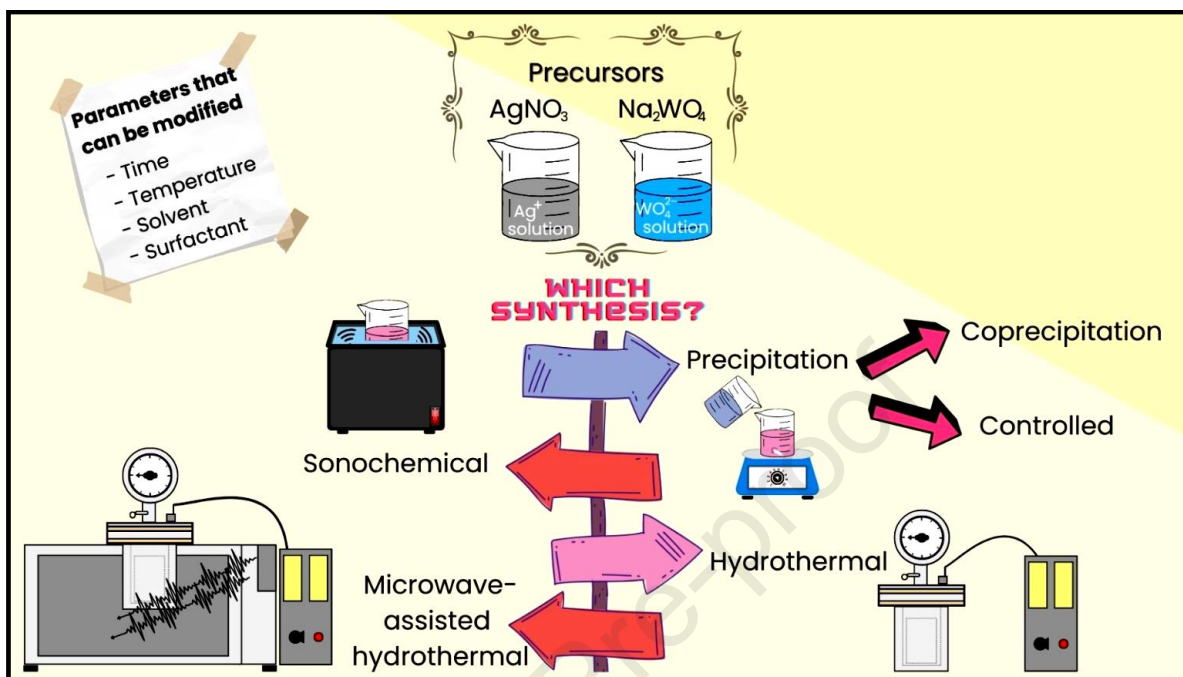
## Figures:



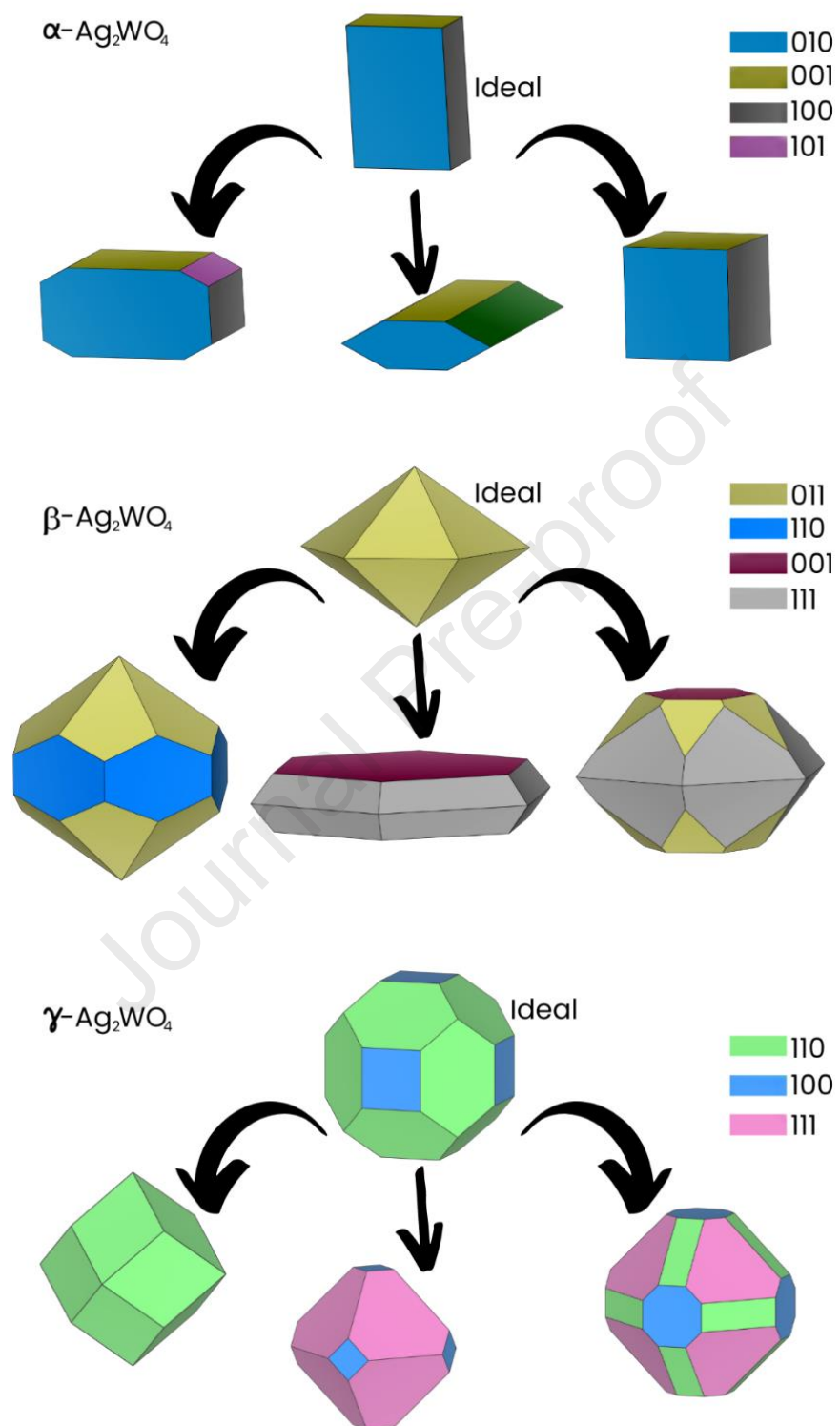
&lt;Figs. 1(a-c)&gt;



&lt;Fig. 2(a-c)&gt;



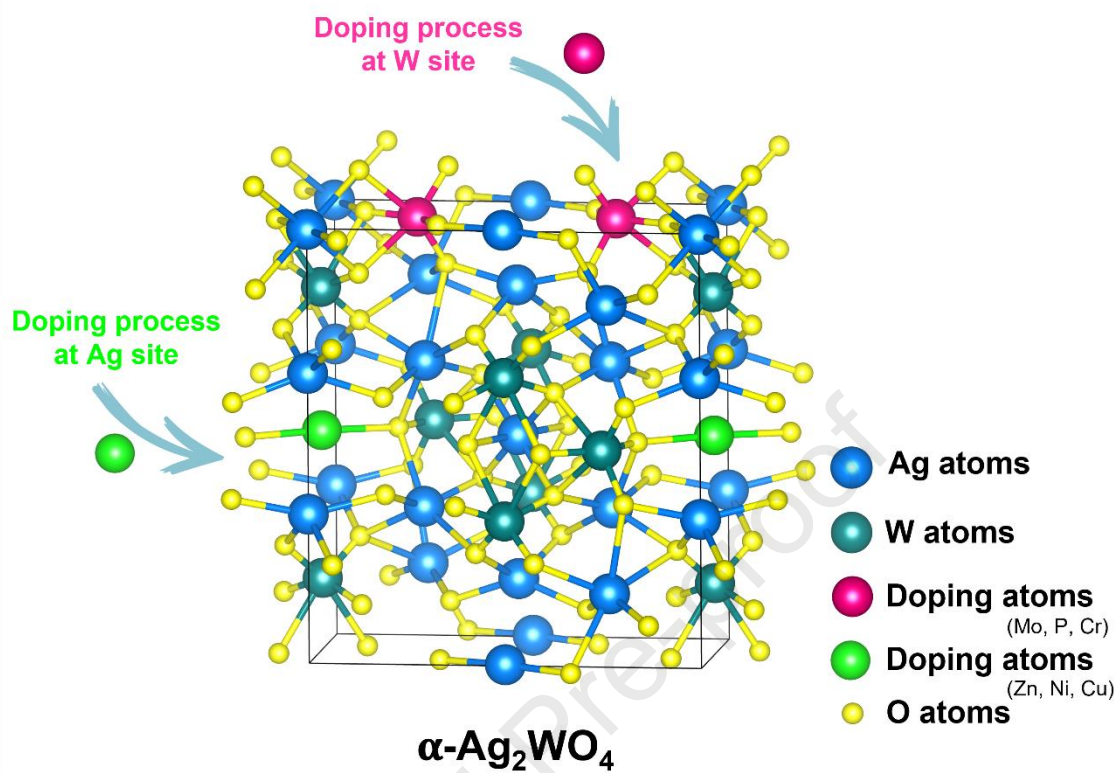
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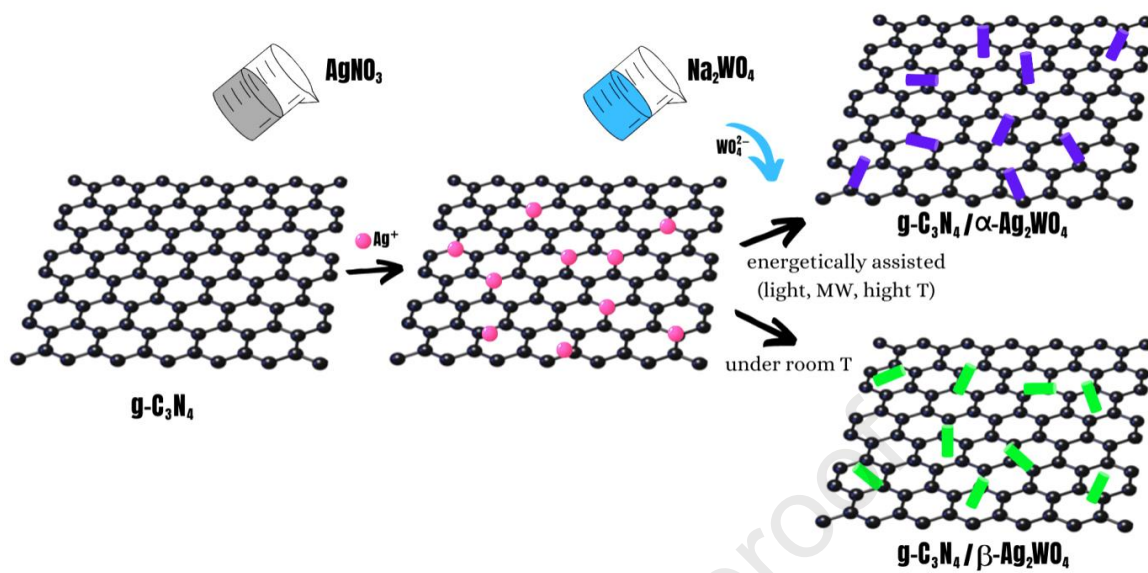
&lt;Fig. 4&gt;



<Fig. 5>

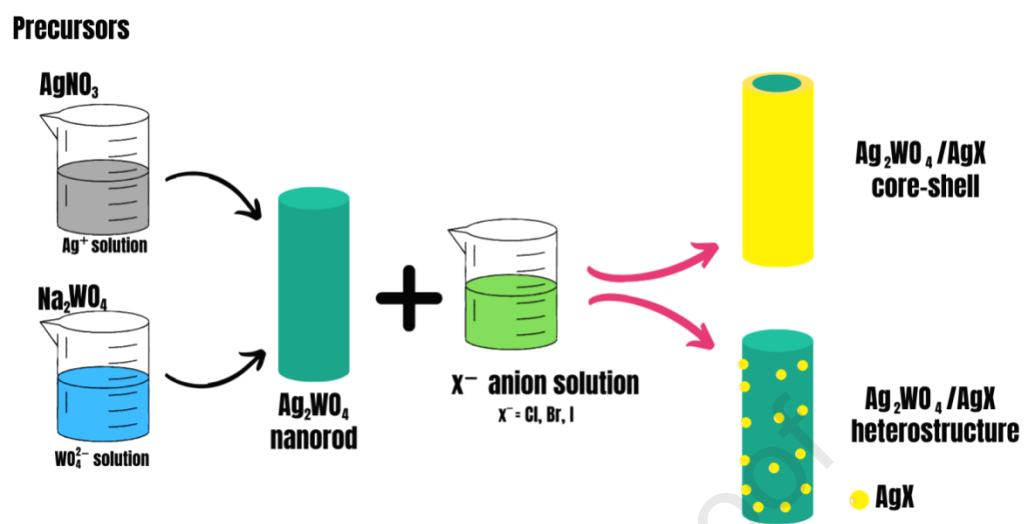


&lt;Fig. 6&gt;

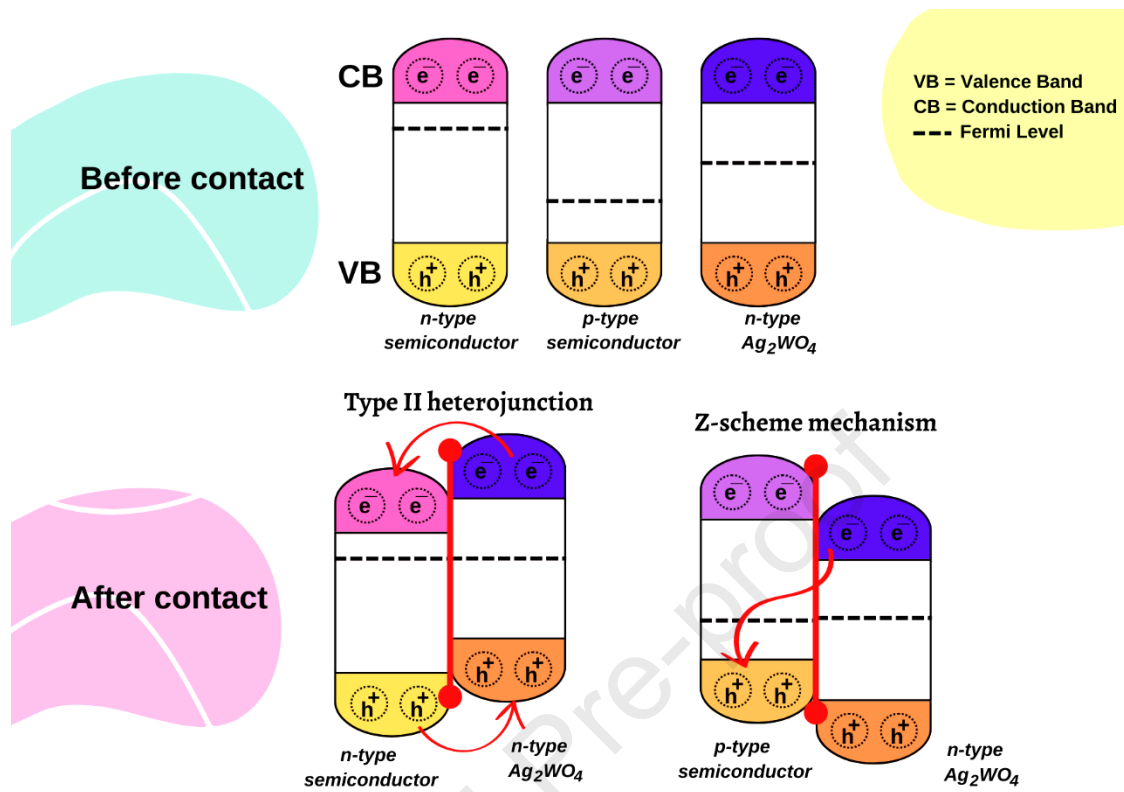


&lt;Fig. 7&gt;

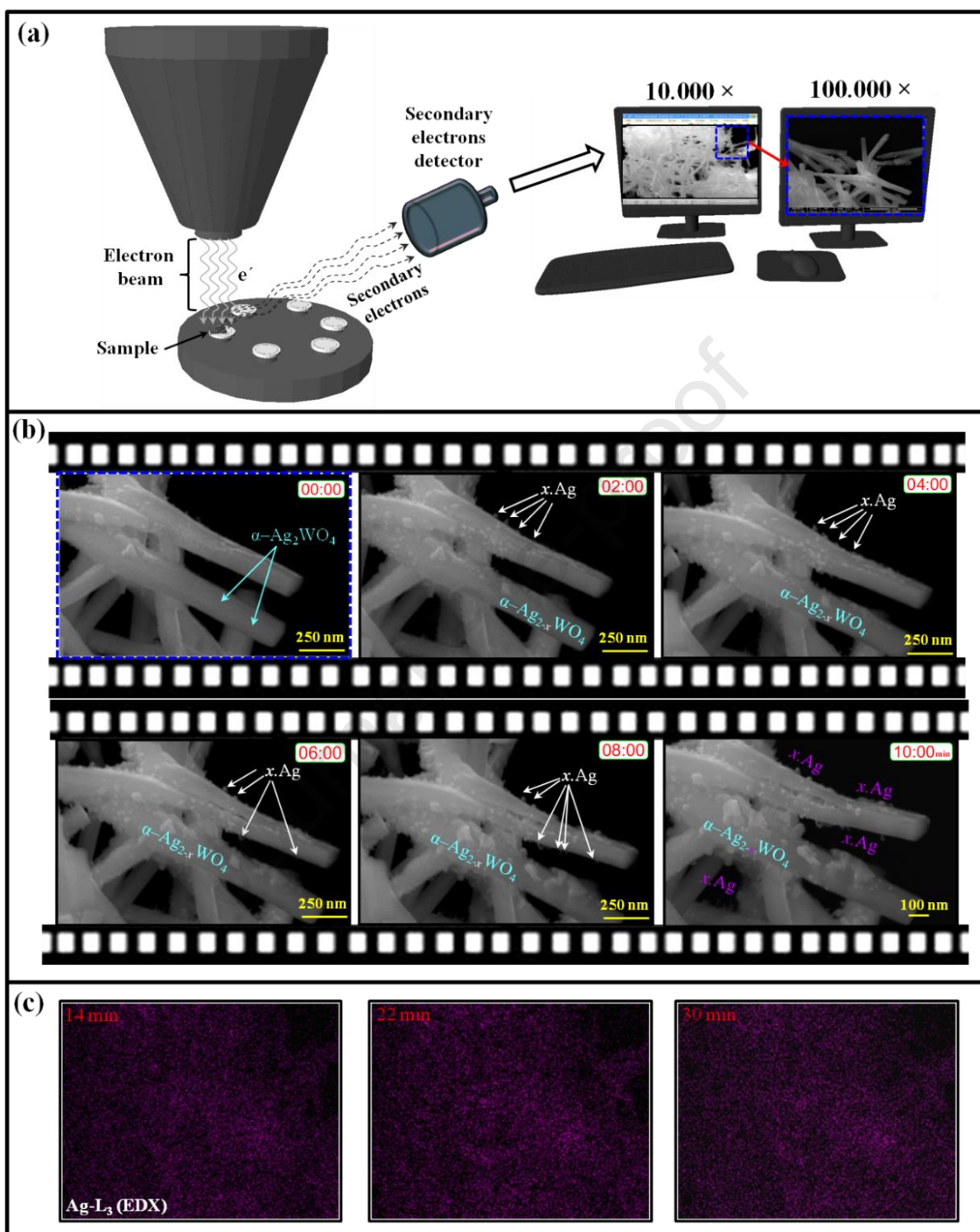




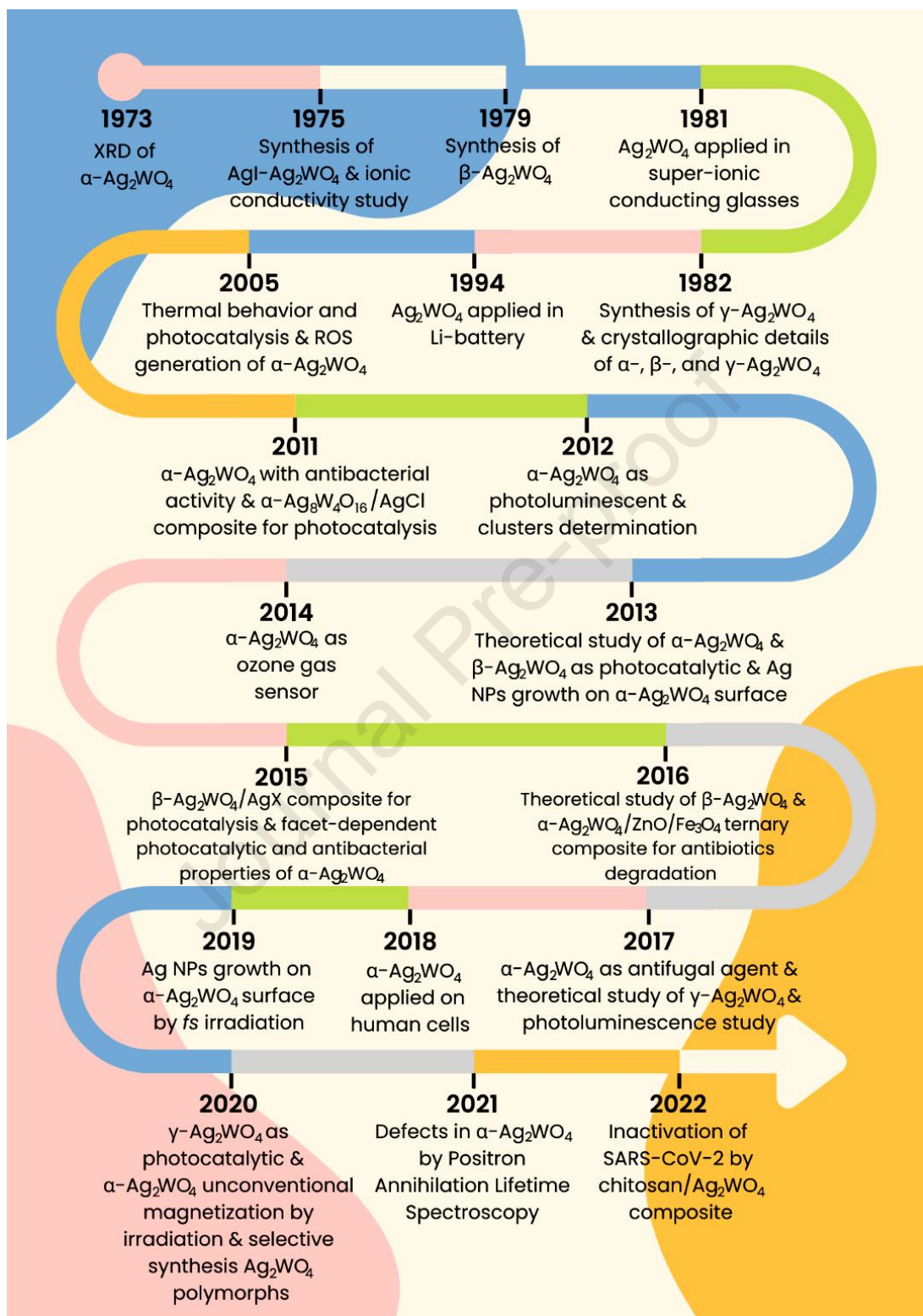
&lt;Fig. 8&gt;



<Fig. 9>



&lt;Figs. 10(a-c)&gt;



&lt;Figs. 11&gt;

## Highlights

- Synthesis of  $\text{Ag}_2\text{WO}_4$  polymorphs and their structures.
- Multifunctional properties of  $\text{Ag}_2\text{WO}_4$ -based materials.
- Improving the property and functionality of  $\text{Ag}_2\text{WO}_4$ -based materials.
- A timeline of the key milestones.

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**Declaration of interests**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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