#### **ORIGINAL PAPER**



# The role of scavengers in the CdS film corrosion

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#### Abstract

The cadmium sulfide (CdS) is one of the most applied chalcogenides to the water splitting (WS) on photoelectrochemical cells (PEC). The CdS bandgap is 2.4 eV which suits very well with the solar spectrum. Unfortunately, because to the high overpotential and sluggish kinetic of the oxygen evolution reaction (OER) on the anodic compartment of the PEC, the pristine form of CdS film undergoes photocorrosion. In this work, using a series of small organic molecules as alternatives for the OER on the CdS/FTO photoanodes under visible light irradiation in  $Na_2SO_4$  0.5 mol L<sup>-1</sup> solution, we explored the dependence of the CdS photocorrosion and the oxidation reaction of small organic molecules. Comparing the methanol, ethylene glycol, and glycerol oxidation reactions, the glycerol showed higher photocurrents in both linear potential voltammetry and chronopotentiostatic experiments. We also employed cross-sectional SEM images to investigate pristine and aged CdS films revealing distinct photocorrosion patterns depending on organic molecule employed. The results suggest the CdS film degradation relates to the low ability of water in scavenge the photogenerated holes and the presence of high reactive species, such as glycerol, capture these holes decreasing the CdS corrosion.

 $\label{eq:cds} \mbox{Keywords } CdS \cdot Photocorrosion \cdot Photoelectrocatalysis \cdot Glycerol oxidation \cdot Ethylene glycol oxidation \cdot Methanol oxidation$ 

## Introduction

The cadmium sulfide (CdS) is an active visible-light sensitive photocatalyst material with 2.4 eV bandgap that is suitable to application under visible light irradiation [1]. It is well known that light absorption with energy higher than bandgap leads to the formation of electrons ( $e^-$ ) and holes ( $h^+$ ) [2, 3] in aqueous solutions. While photogenerated electrons reduce protons, yielding H<sub>2</sub>, holes capture electrons from species in the electrolyte, for instance, water, organic and inorganic compounds, and even the elements from the photoanode, e.g., sulfur from CdS film structure [4], explaining the photocorrosion process by the irreversible hole-driven oxidation of the photoanode surface [5]. Specifically, CdS undergoes photocorrosion in presence of oxygen as follows (Eq. 1 and Eq. 2) [6, 7]:

$$\operatorname{CdS} \xrightarrow{h\nu} e_{\operatorname{CB}}^{-}(\operatorname{CdS}) + h_{\operatorname{VB}}^{+}(\operatorname{CdS}) \tag{1}$$

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$$CdS + 4h^{+} + 2H_2O + O_2 \rightarrow Cd^{2+} + SO_4^{2-} + 4H^{+}$$
 (2)

In the absence of oxygen, the photocorrosion also takes place, but it is caused by two photogenerated holes yielding sulfur (Eq. 3) [6], conversely, CdS is partially dissolved due to the chemical equilibrium with  $Cd^{2+}$  and  $S^{2-}$  (Eq. 4) and their secondary dependences with solution pH (Eqs. 5 and 6).

$$CdS + 2h^+ \to Cd^{2+} + S \tag{3}$$

$$CdS \approx Cd^{2+} + S2^{2-} \tag{4}$$

$$Cd^{2+} + 2OH^{-} \rightleftharpoons Cd(OH)_2$$
 (5)

$$S^{2-} + 2H^+ \rightleftharpoons H_2 S \tag{6}$$

Several strategies to inhibit photocorrosion have been described on the literature, including the use of protective layers [8, 9], heterojunctions [10, 11], or adding a sacrificial agent to the solution to capture the photogenerated holes of the valence band of the semiconductor [12–14]. The sacrificial agents scavenge the holes of the valence band easier than

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sulfur anions from CdS film, because its oxidation potential is lower than that for sulfur oxidation [15]. Oxygen evolution reaction (OER) can provide electrons to the holes; however, the standard potential for the OER is relatively high (1.23 V vs SHE) and the four-electron transfer to produce  $O_2$ has kinetic limitations. Furthermore, the oxygen evolution induces the photocorrosion as described by Eq. (2).

Other hole scavengers such as hydrogen peroxide, sulfite, and oxalic acid are largely employed to measure the maximum activity of material i.e., the maximum amount of hole generated in the valence band without kinetic limitations. Interestingly, more than provide an activity limit value, the oxidation of organic molecules may produce high-added value products, which is very attractive from an economic point of view [16], once while the cathode is devoted to produce  $H_2$ , the products of anode reaction can be employed on the industry.

Methanol [17–19], ethanol [20], and oxalic acid [21, 22] are examples of small organic molecules that have been applied in photoelectrochemical cell (PEC) anodes to increase the hole scavenge coupled to the production of added-value products or even CO<sub>2</sub>. Glycerol, one of the main biodiesel by-products, is also an important and abundant biomass-related compound [23] with the possibility to yield commercially valuable oxidation products as dihydroxyacetone, glyceraldehyde, and glyceric acid [24]. Therefore, glycerol oxidation seems to be a good substitute for OER on the photoanode compartment of the PEC improving the system overall efficiency. In this work, we compare the use of water, methanol, ethylene glycol, and glycerol on the anode compartments under visible light illumination composed by CdS thin film deposited onto a fluorine-doped tin oxide (FTO) substrate. The scavengers are compared in terms of photocurrents depicted on both linear potential voltammogram and chronoamperometric curves at 0.6 V vs RHE and photocorrosion inhibition as probed by scanning electron micrography (SEM).

# **Experimental procedure**

## Synthesis of CdS photoanode

All chemicals were purchased from Sigma-Aldrich and used without further purification. An Anton Paar Microwave Reactor Monowave 400 was employed to synthesize CdS thin films by the chemical bath deposition. Firstly, the FTO substrate was immersed for 5 min. in a G-30 vial containing 10 mL of a NH<sub>4</sub>OH 3.0 mol L<sup>-1</sup> solution, followed by the addition of 5 mL of a CdSO<sub>4</sub> 0.1 mol L<sup>-1</sup> solution and 5 mL CH<sub>4</sub>N<sub>2</sub>S 0.15 mol L<sup>-1</sup> solution. The vial was put into the microwave and heat up to 80 °C for 2 min under constant magnetic

stirring. After this, the CdS/FTO was washed in pure water to remove any non-deposited CdS excess and dried in air.

#### Characterization

XRD measurements were performed using a Shimadzu 6000 diffractometer operating at 30 kV/40 mA, with a copper tube:  $\lambda k\alpha = 1.5418$  Å, in the range of 20–70° (2 $\theta$ ) and a sweep rate of  $2^{\circ}$  min<sup>-1</sup>. Raman spectra were obtained using a Witec Raman Microscopy (Ulm, Germany) coupled with a high linear stage and Nikon lens (100X/0.90). Raman signals were excited with a laser Nd: YAG (514 nm; 10 mW). UV-Vis measurements were carried out in a Varian Cary 5E UV-Vis-NIR spectrophotometer, the films were analyzed in the 300-700 nm range. Transmittance mode was employed for this measurement, with a clean FTO substrate as reference. The cross-sectional SEM images were taken in a FEI inspect F50 microscopy. Mott-Schottky plot were obtained by AC techniques with ±10 mV perturbation in three frequencies: 10, 100, and 1000 kHz from 0.2 to 1.2 V vs SHE, with each point being collected at every 50 mV with current stabilization for 15 s.

## **Photoelectrochemical experiments**

A three-electrode electrochemical cell equipped with a quartz window was employed on photoelectrochemical experiments. The system was composed of CdS/FTO (1×1 cm<sup>2</sup>) as working electrode; a Pt plate (2×2 cm<sup>2</sup>) as a counter electrode; and a Ag/AgCl/KCl<sub>(sat)</sub> as reference electrode; however, all the potentials will be expressed in the Reversible Hydrogen Electrode (RHE) scale according to Eq. (7), in which the  $E_{appl}$  and  $E_{Ag/AgCl/KCl_{(sat)}}$  represent the applied potential and the standard potential for Ag/AgCl/KCl<sub>sat</sub> electrode (0.197 V vs SHE), respectively.

$$E_{\rm RHE} = E_{\rm appl} + 0.0592 \rm pH + E_{\rm Ag/AgCl/KCl_{(sat)}}$$
(7)

 $Na_2SO_4 0.5 \text{ mol } L^{-1} (pH=6.5)$  served as supporting electrolyte with further direct addition of methanol, ethylene glycol, glycerol, or sodium sulfite ( $Na_2SO_3$ ) in sufficient amounts to yield 0.5 mol  $L^{-1}$ . The pH was corrected back to 6.5 after  $Na_2SO_3$  due to the shift caused by the  $SO_3^{-2}$ /HSO<sub>3</sub><sup>-</sup> equilibrium as discussed on the text. The solar spectrum was simulated with a Newport LCS-100 Class ABB small area solar simulator with 100 mW cm<sup>-2</sup>. Linear potential voltammetry measurements were taken in the range of 0.0 to 1.0 V vs RHE under dark and simulated solar light regimes at 5 mV s<sup>-1</sup>. Further light on–off linear potential voltammetry experiments were carried out in the same conditions. Chronoamperometric profiles were also analyzed under and without solar light illumination at a fixed potential (0.6 V vs RHE) for 1 h.

## **Results and discussion**

#### Characterization

The CdS thin film deposited onto FTO substrates were examined by X-ray diffraction technique (XRD) as shown in Fig. 1a. The peaks highlighted by red circles and black squares correspond to the CdS film and FTO substrate, respectively. The Raman spectra (Fig. 1b) shows two well defined peaks centered at 305 cm<sup>-1</sup> (1LO) and 605 cm<sup>-1</sup> (2LO), both related to the longitudinal optical phonons. Figure 1c shows the optical transmittance spectra for the CdS film from 700 to 300 nm (inset) and the  $(ahv)^2$  vs hv plot, the so-called Tauc plot, in which was estimated the linear behavior at the beginning of adsorption region as indicates by the red solid line. The capacitances (*C*) of the semiconductor-electrolyte interface for several potentials

allowed to build the  $1/C^2$  vs *E* plot, the Mott-Schottky plot (Fig. 1d) and the linear trend observed at low potential was used to obtain the potential value at  $1/C^2 = 0$ , as shown by the red solid line.

Figure 2 presents the cross-sectional scanning electron microscopy (SEM) image of the synthetized CdS film in pristine form. Clearly, three strictly different regions can be identified: the glass substrate; the ordered crystalline FTO layer with approximately 590 nm; the CdS films with approximately 900 nm thick, showing a globular packed morphology.

#### **Electrochemical measurements and photocorrosion**

Figure 3 compares the activity of CdS for water (only supporting electrolyte), glycerol, and sulfite oxidation reactions in the Na<sub>2</sub>SO<sub>4</sub> 0.5 mol L<sup>-1</sup> solution with pH=6.5. The plate



**Fig.1** a X-ray diffraction patterns  $\bullet$  represent CdS and  $\blacksquare$  represent FTO diffraction peaks. b Raman spectra. c Tauc plot with n = 1/2, inset: UV–Vis spectra. d Mott-Schottky plot

Fig. 2 Cross-section SEM image for CdS thin films deposited into FTO



(a) shows the linear potential voltammogram at 5 mV s<sup>-1</sup> from 0.0 to 1.0 V (*vs* RHE) in the dark (dotted lines) and simulated solar light (continuous lines) and the plate (b) the same experiment but in intermittent light conditions i.e., on/ off regime. Finally, the plate (c) brings the currents observed in the presence of light along 1 h with constant potential of 0.6 V (chronoamperometry experiment). Regardless the method employed, the activity, in terms of photocurrents, follows the order:  $j_{sulfite} > j_{glycerol} > j_{water}$ .

Glycerol was also replaced by ethylene glycol and methanol to compare the CdS activity for the oxidation of distinct organic molecules. Figure 4(a) and (b) shows the linear potential voltammograms under intermittent solar illumination and the chronoamperograms at 0.6 V for the three organic molecules studied in this work, respectively. The error bar represents the standard deviation along the timeseries for the three distinct experiments with three distinct electrodes (prepared with identical conditions) for each organic molecule. The plate (c) brings an overview of the currents registered for all the systems after 1 h at 0.6 V.

The CdS films employed for water, glycerol, methanol, and sulfite chronoamperometry experiments were cut and analyzed after the electrochemical tests. The SEM imagens (cross-sections) of these films are shown in Fig. 5.

# Discussion

The obtained XRD analysis (Fig. 1a) has a good match with patterns for cubic CdS and tetragonal  $SiO_2$  from the glass substrate, JCPDS-ICDD Card No. 01-075-0581 and JCPDS-ICDD No. 41-1445, respectively. The CdS diffraction

peak located at  $2\theta = 26.9^{\circ}$  corresponds to the (0 0 2) hexagonal plane or the (1 1 1) cubic plane of CdS, the peak at  $2\theta = 44.5^{\circ}$  can be assigned to the (1 1 0) hexagonal plane or (2 2 0) cubic plane [25] and the peak at  $2\theta = 54.7^{\circ}$  assigns to the (3 1 1) plane of the cubic CdS [26]. Thus, the presence of the peak at  $54.7^{\circ}$  and the absence of the peaks relative to the hexagonal diffraction for the (1 0 0) and (1 0 1) planes [27] indicate the cubic crystal phase was obtained. The two bands observed on the Raman spectra (Fig. 1b) also agree with the CdS film structure. Both XRD and Raman spectra agree with the observed patterns for the cubic phase found in other papers where CdS films were prepared in similar conditions [25, 28].

The as-prepared film absorption band at wavelengths lower than 520 nm (Fig. 1c) is in good agreement with the value for the CdS band gap [29]. The band gap energy ( $E_g$ ) was obtained by the Tauc plot [30] according to Eq. (8):

$$(\alpha h\nu)^{1/n} = A(h\nu - E_g) \tag{8}$$

. ,

where  $\alpha$  is the absorption coefficient;  $h\nu$  is the energy of the incident photons; *n* is the transition coefficient; and *A* is a constant. The Tauc plot revealed a  $E_g = 2.36$  eV, in good agreement with CdS thin film in the literature [31]. The coefficient *n* is dependent on the transition type, and was set as  $n = \frac{1}{2}$  i.e., a direct allowed transition.

The Mott-Schottky plot (Fig. 1d) was based in the potential region employed for the photoelectrochemical measurements, which was from 0.0 to 1.0 V vs RHE. The positive slope indicates a n-type semiconductor and the interception of linear region with the x-axis  $(1/C^2=0)$  gives a -0.96 V vs SHE flat-band potential value ( $E_{\rm fb}$ ) in good agreement



**Fig. 3** CdS/FTO photoelectrochemical measurements in a  $Na_2SO_4$  0.5 mol  $L^{-1}$  solution for water oxidation (only supporting electrolyte) and after addition of glycerol (1.0 mol  $L^{-1}$ ) or sulfite anions (1.0 mol  $L^{-1}$ ). **a** Linear potential voltammograms at 5 mV s<sup>-1</sup> in the

with the literature [32]. The  $E_{\rm fb}$  is related with the conduction band (CB) for a n-type semiconductor and, taking the bang gap value obtained from Tauc plot into account, it is possible to determine the valence band (VB) energy as 1.40 V vs SHE. The Mott-Schottky plot slope also allowed us to estimate the number of charge majority carriers [33] as  $4.65 \times 10^{18}$  electrons and the same value for holes, once the CdS is an intrinsic semiconductor, i.e. once it has no impurities, the number of its charge carriers is determined by the material properties itself and the number of electrons and holes are the same.

The globular structure of the film agrees with the literature of CdS chemical bath deposition [34], as also the structural, optical, and microscopy characterizations of the CdS films, which confirms the synthesis of CdS film deposited into FTO substrates.

presence (solid) or absence (dotted) simulated solar illumination; **b** linear potential voltammograms under intermittent solar illumination; and **c** chronoamperometry under simulated solar illumination at 0.6 V vs RHE

The energy values estimated for conduction band (CB), valence band (VB), and band gap (Eg) were used to build the scheme shown in Fig. 6.

The electrochemical experiments showed photocurrents for water oxidation reaction in all potential region of the linear potential voltammogram in Fig. 3b (higher currents in the presence of simulated light than in the dark conditions); however, these photocurrents do not exceed more than 10  $\mu$ Acm<sup>-2</sup>. Glycerol is a better hole scavenger than water, as can be observed by the photocurrents increase in the region from 0.0 to 0.2 V vs RHE and are quite stable from 0.2 to 1.0 V vs RHE reaching ca 70  $\mu$ Acm<sup>-2</sup>. Likewise, ethylene glycol has higher photocurrent values than water and methanol in the region from 0.0 to 1.0 V vs RHE. When compared to glycerol its values are similar in the higher potential region (Fig. 4a). Despite the three of them being



**Fig. 4** CdS/FTO photoelectrochemical measurements in a  $Na_2SO_4$  0.5 mol L<sup>-1</sup> solution for water oxidation (only supporting electrolyte) and after addition of glycerol, ethylene glycol, or methanol in sufficient amount to yield 1.0 mol L<sup>-1</sup>. **a** Linear potential voltammograms

at 5 mV s<sup>-1</sup> under intermittent simulated solar illumination and **b** chronoamperometry under continuous simulated solar illumination at 0.6 V vs RHE. **c** Each studied molecule currents registered under simulated solar illumination at 0.6 V vs RHE after 1 h

**Fig. 5** Cross-section SEM images for CdS thin films deposited on FTO after chronoamperometric experiment for 1 h at 0.6 V vs ERH in Na<sub>2</sub>SO<sub>4</sub> 0.5 mol L<sup>-1</sup> under sunlightsimulated illumination. with addition of **a** Na<sub>2</sub>SO<sub>3</sub> 0.5 mol L<sup>-1</sup>; **b** glycerol 1.0 mol L<sup>-1</sup>; **c** methanol 1.0 mol L<sup>-1</sup>; **c** methanol 1.0 mol L<sup>-1</sup>; and **d** only in supporting electrolyte





Fig.6 Band energy scheme based on the energy values measured for synthetized CdS film. Me=methanol, EG=ethylene glycol, and Gly=glycerol

small alcohols, methanol has a lower activity than ethylene glycol and glycerol in all the potential range studied. Sulfite oxidation provides photocurrents up to 0.6 mAcm<sup>-2</sup> indicating that while the photocurrents for organic molecules are higher than that observed to OER, the photogenerated holes are not completely consumed.

About the sulfite oxidation, it is important highlight that at pH 6.5, both  $SO_3^{2-}$  and  $HSO_3^{-}$  (hydrogen sulfite anion) exists in solution. The sulfite oxidation to sulfate (Eq. 13) involves only two electrons and has a lower standard potential than water (Eq. 9). The sulfite oxidation can also yield  $S_2O_6^{2-}$  or  $SO_3^{\bullet-}$  (radical); however, these species can be further oxidized yielding sulfate as product. Herein, it is important to mention that the direct dissolution of Na<sub>2</sub>SO<sub>3</sub> into unbuffered solution such as Na<sub>2</sub>SO<sub>4</sub> raises the pH; for instance, the addition of Na<sub>2</sub>SO<sub>3</sub> in sufficient amount to yields 1.0 mol L<sup>-1</sup> modified the pH from 6.5 to 9.0. Figure 7 shows the linear potential voltammograms under intermittent



Fig. 7 Linear potential voltammogram under intermittent solar irradiation using CdS/FTO photoanode in a  $Na_2SO_4$  0.5 mol  $L^{-1}$  supporting electrolyte after addition of  $Na_2SO_3$  1.0 mol  $L^{-1}$ 

solar irradiation using the CdS/FTO photoanode in  $Na_2SO_4$ 0.5 mol L<sup>-1</sup> solution after addition of  $Na_2SO_3$  (mol L<sup>-1</sup>) with pH correction to 6.5 (red) and without pH correction (black line). Clearly, higher photocurrents are obtained in pH 6.5 than in pH 9.0, highlighting the importance of maintain the same experimental condition to compare distinct analytes.

Based on thermodynamic data, all the reactions studied can donate electrons to CdS holes  $(h^+)$ , as inferred by the standard potentials ( $E^0$ /vs SHE) to alcohol/CO<sub>2</sub> and sulfite/ sulfate equilibriums (Eqs. 9–13). These potentials were used to build the scheme shown in Fig. 6.

$$O_2 + 4H^+ + 4e^- \rightleftharpoons 2H_2O \ E^0 = 1.229$$
 (9)

$$CO_2 + 6H^+ + 6e^- \Rightarrow CH_3OH + H_2O E^0 = 0.025$$
 (10)

$$2\text{CO}_2 + 10\text{H}^+ + 10\text{e}^- \rightleftharpoons (\text{CH}_2\text{OH})_2 + 2\text{H}_2\text{O}\ E^0 = 0.013$$
(11)

$$3CO_2 + 14H^+ + 14e^- \Rightarrow C_3H_8O_3 + 3H_2O E^0 = 0.005$$
(12)

$$SO_4^{2-} + 2H^+ + 2e^- \rightleftharpoons SO_3^{2-} + H_2O \ E^0 = -0.104$$
 (13)

Sulfite can capture the holes with the best efficiency when compared to the other species, even with chemical equilibrium converting part of sulfite to hydrogen sulfite as above mentioned. Although the alcohols do not set a chemical equilibrium in aqueous solution, the methanol, ethylene glycol, and glycerol oxidation mechanisms are more complex than sulfite, especially by the possibility to exchange a large number of electrons per molecule, for example, glycerol oxidation can start by losing 2e<sup>-</sup>, yielding aldehyde or ketone, or going to higher oxidation states loosing up to  $14e^-$  to yield CO<sub>2</sub>. It is also important to considerer that organic molecules can adsorb on the semiconductor which prevent new molecules to arrive on the photoanode surface, lowering the conversion efficiency.

Although one could relate the standard potentials to alcohol/CO<sub>2</sub> equilibriums with the photocurrents values, these potentials are very close for methanol, ethylene glycol, and glycerol when compared to the water oxidation; however, considering the photocurrent after 1 h at 0.6 V for water oxidation, this value is increased by two and ten times in the presence of methanol and glycerol/ ethylene glycol, respectively which allow us to conclude that the reactivity is not straightforwardly related to the standard potential even for species with similar function groups. Moreover, the methanol oxidation to partially oxidized products such as formal-dehyde (Eq. 14) or formic acid (Eq. 15) also depict standard potentials [35] that allow the spontaneous reaction with  $h^+$ .

$$CH_2O + 2H^+ + 2e^- \rightleftharpoons CH_3OH \ E^0 = 0.319$$
 (14)

# $\text{HCOOH} + 4\text{H}^+ + 4\text{e}^- \rightleftharpoons \text{CH}_3\text{OH} + \text{H}_2\text{O} \ E^0 = 0.161 \ (15)$

Conventionally sacrificial agents containing hydroxyl groups are essential to efficient H<sub>2</sub> evolution in photoelectrochemical cells (PEC) [36, 37]. However, it is still not clear if the increase in the structure complexity can enhance the system performance. Some works show that increasing sacrificial agent complexity decreases the H<sub>2</sub> production rates [38, 39], while other ones show the opposite effect [40, 41]. This contradiction may be explained by the electronic and physical characteristics of the semiconductor catalysts that rules the photoactivity, as well as other factors such as the nature of the electrolyte, local pH, temperature, sacrificial agent concentration, and its interaction with the catalyst surface. Fu et al. reported that in a Pt/TiO<sub>2</sub> photocatalyst, higher H<sub>2</sub> and CO<sub>2</sub> evolution were observed when the number of hydroxyl groups was increased in the molecules in the order: isopropanol < propylene glycol < glycerol [41], these findings could be attributed to a dual characteristic of the hydroxyl groups: effective  $h^+$  scavengers and good anchors for adsorption on TiO<sub>2</sub>, as observed here for CdS.

We have mapped the products of electrochemical glycerol photooxidation by hematite observing a low selectivity and near zero activation energy [42]. The results were interpreted as mechanism involving an adsorbed or near to surface radical that will provides distinct products depending on the formation of the hole on the surface. In the present work, the polyalcohol molecules have higher activity because their hydroxyl groups are more susceptible to interact with the CdS surface, once it has two or more hydroxyl groups. Contrary, methanol has only one hydroxyl group, which makes weaker interactions with the CdS.

Finally, the CdS is not stable in pH below 8. The oxidation process in aqueous media consumes OH<sup>-</sup> (see Eq. 5) or releases H<sup>+</sup> leading to the pH decrease and contribute to the stabilization of the cation Cd<sup>2+</sup> in solution [43], and the consumption of S<sup>2-</sup> as shown in Eq. 6, which highlights the CdS dissolution. This, in fact, is enhanced when the CdS is directly illuminated, because regardless the presence of oxygen, the photogenerated holes oxidize CdS which contribute to the film corrosion (Eqs. 1–3).

The morphological analysis of the films after the electrochemical aging tests (Fig. 5) clearly shows that the decreasing in thickness depends on the hole scavenger in the electrolyte. The OER (Fig. 5d) was the reaction that almost completely vanished the film due to the photocorrosion reaction. In the case of sulfide oxidation reaction, the film thickness decreases almost a half, but keeping its globular packed morphology as shown in the image of pristine film (Fig. 2).

Interesting, despite the decrease of pH due to the oxidation processes increase the CdS chemical degradation, the presence of alcohols on the solution seems to prevent the CdS film corrosion. The CdS film thickness after the glycerol experiment seems to keep as high as the initial pristine thickness (see Figs. 5b and 2). This can be interesting, because even though glycerol does not present the same photocurrent as sulfite, it can remain stable for longer periods, being more efficient in long-term analysis than sulfite which suffers a loss of 50% of the initial thickness value of the film. In the presence of alcohol, the morphology seems to be changed and a floccular structure grows over the compact globular CdS. The mechanism of the morphological/structural changes is under investigation.

## Conclusion

In this work, we reported the fabrication of CdS photoanode films with a globular morphology by a chemical method deposition, which is easily photocorroded due to the reaction between the photogenerated holes on the valence band of the semiconductor material and the sulfur anions, as part of the film composition. A strong relationship between the photocurrent and the presence of a hole scavenge present in the electrolyte was found for all investigated small organic molecules, which shows the reactivity depends on the number of aliphatic alcohols groups presented in the molecule: methanol, ethylene glycol, and glycerol. Based on these results, we can expect promoting the small organic molecules oxidation, in particular, glycerol, over the photocorrosion reaction on CdS surface, on the basis of the linear potential voltammetry and chronoamperometric experiments under simulated sunlight conditions. Glycerol oxidation was found as an ideal hole scavenger having the compromise to get the photogenerated roles on CdS valence band and decreases the film dissolution in comparison with other organic and inorganic compounds. Finally, although the photocurrents observed for organic molecules are lower than those observed for  $SO_3^{2-}$  oxidation, the organic molecules open the possibility to produce added-value compounds on PEC anodes concomitant to  $H_2$  production on the cathode.

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